Photoelastic evaluation of the effect of composite formulation on polymerization shrinkage stress

Abstract: We compared polymerization stress in two commercial composites and three experimental composites made using camphorquinone (CQ) and/or phenylpropanedione (PPD) as photoinitiators. The internal surfaces of photoelastic resin discs with cylindrical cavities were roughened and treated with adhesive. Composites were divided into five groups: two commercial composites (Filtek Silorane and Filtek Z250) and three experimental composites with CQ/amine, CQ/PPD/amine, and PPD/amine. Composites were photopolymerized inside cavities, and subjected to photoelastic analysis immediately and at 24 hours and 7 days later using a plane polariscope. Stress created by Silorane (3.08 ± 0.09 MPa) was similar to that of Z250 (3.19 ± 0.13 MPa) immediately after photopolymerization (p > 0.05). After 24 hours and 7 days, Z250 (3.53 ± 0.15 and 3.69 ± 0.10 MPa, respectively) showed higher stress than Silorane (3.19 ± 0.10 and 3.16 ± 0.10 MPa, respectively). Qualitative analysis immediately after photopolymerization showed composite/CQ promoted higher stress than PPD, but stress levels at other evaluated times were statistically similar, varying between 3.45 ± 0.11 MPa and 3.92 ± 0.13 MPa. At 24 hours and 7 days, Silorane created the lowest stress. All photoinitiators created comparable tensions during polymerization.

Descriptors: Composite Resins; Polymerization; Photoinitiators, Dental; Dental Stress Analysis.

Introduction

Dental composites are polymer-based materials that shrink during polymerization. Stress occurs when shrinkage is limited and the material is rigid and flow resistant. These stresses may cause imperfections at the bond interface, allowing marginal leakage and bacterial contamination. Many factors—including some that are related to material composition such as organic matrix composition, initiation system, and amount of inorganic filler—affect the degree of polymerization shrinkage, influencing contraction and strain.

The organic matrices of composites contain different monomers. Recently, a new silorane-based monomer composite was introduced for use in dentistry. The silorane polymer chain is formed by cationic ring opening and involves the action of three initiator components. Ring opening results in molecular expansion that compensates for polymerization shrinkage. Thus, stresses generated by polymerization processes associated with silorane are lower than those associated with methacrylate-based composites.

Most composites include camphorquinone (CQ) as a photoinitiator agent
and an amine acting as co-initiator. 1-Phenyl-1,2-propanedione (PPD) has been suggested as a possible CQ substitute, with the potential to reduce discoloration and improve the aesthetic properties of composites. Generally, PPD-containing composites attain maximum polymerization values more slowly than those with CQ, because the polymerization rate and degree of conversion depend on the absorbed power density.

The aim of this study was to use photoelastic analysis to evaluate polymerization shrinkage stress. Photoelastic analysis is a visual measurement based on the property of some transparent materials to exhibit interference fringes when stressed in a polarized light field. This technique has been used previously to evaluate polymerization shrinkage stress in dental composites. These fringes have mathematical relationships and order numbers that are dependent on the load intensity applied to the photoelastic material. In this study, stress induced in three experimental and two commercial composites was assessed immediately after photopolymerization and 24 hours and 7 days later.

Methodology

Composites composition

Commercial composites were Filtek Silorane (Silorane) and Filtek Z250 (Z250) (3M ESPE, St. Paul, USA) prepared using Adper Single Bond 2 and Silorane adhesive systems (3M ESPE).

For experimental composites, the organic matrix was composed of the dimethacrylate monomers bisphenol A glycidyl dimethacrylate (Bis-GMA, Aldrich Chemistry, São Paulo, Brazil), ethoxylated bisphenol A dimethacrylate (Bis-EMA, Aldrich), urethane dimethacrylate (UDMA, Aldrich), and triethyleneglycol dimethacrylate (TEGDMA, Aldrich). The inorganic matrix was composed of Bario Aluminium Silicate (BaAlSi, 0.5 µm, Biodinâmica, Ibiporã, Brazil) and 20 wt.% silicon dioxide (SiO₂, 0.04 µm, Biodinâmica). Both were treated with 3% methacryloxipropyl trimethoxysililan. The initiator agents were CQ (Aldrich) and PPD (Aldrich), the reducing agent was dimethyl amino ethyl methacrylate (DMAEMA) (Aldrich), and the inhibitor was butylated hydroxytoluene (0.1 wt.%) (Aldrich). The composition of the experimental composites is presented in Table 1. Experimental composites were prepared at room temperature under low illumination by first preparing separate organic and inorganic matrices and then mixing these phases.

Specimen preparation

Araldite GY 279 photoelastic base resin and Aradur 2963 catalyst (Araltec Produtos Químicos, São Paulo, Brazil) were used to fabricate discs 20 mm in diameter, 2 mm thick, with a central perforation 6 mm in diameter. Inner preparation surfaces of Araldite resin discs were roughened using a bench drill to improve photoelastic resin–composite bonding. Discs were stored at 37°C for 24 hours to allow relief of photoelastic resin stress. Before specimen preparation, each disc was placed in the photoelastic device to assess residual stress. If any was detected, the disc was stored in an oven at 37°C for 24 hours until stress was completely released.

Photoelastic discs were separated into five groups (n = 5 each):
- Silorane,
- Z250,
- Experimental I (CQ),
- Experimental II (CQ/PPD), and
- Experimental III (PPD).

The inner surfaces of Araldite discs were treated with the appropriate adhesive system for each composite.

<table>
<thead>
<tr>
<th>Composites</th>
<th>Organic matrix (40 wt.%)</th>
<th>Filler (60 wt.%)</th>
</tr>
</thead>
<tbody>
<tr>
<td>Experimental I</td>
<td>Bis-GMA (29%)</td>
<td>BaAlSi, 0.5 µm (80%)</td>
</tr>
<tr>
<td>Experimental II</td>
<td>UDMA (31%)</td>
<td>SiO₂, 0.04 µm (20%)</td>
</tr>
<tr>
<td>Experimental III</td>
<td>Bis-EMA (31%)</td>
<td>(CQ/amine (0.4/0.8%))</td>
</tr>
<tr>
<td></td>
<td>TEGDMA (7%)</td>
<td>(CQ/PPD/amine (0.2/0.2/0.8%))</td>
</tr>
<tr>
<td></td>
<td>Initiation system</td>
<td>(PPD/amine (0.4/0.8%))</td>
</tr>
</tbody>
</table>

Table 1 - Composition of experimental composites (in wt.%).

Bis-GMA, bisphenol A glycidyl dimethacrylate; BIS-EMA, ethoxylated bisphenol A dimethacrylate; UDMA, urethane dimethacrylate; TEGDMA, triethyleneglycol dimethacrylate; CQ, camphorquinone; PPD, 1-phenyl-1,2-
following the manufacturer’s instructions. A bulk increment of composite was placed in the central perforation of the photoelastic disc and polymerized using an LED-based dental curing unit (Ultra-Lume LED 5, Ultradent Products Inc., South Jordan, USA) set at 890 mW/cm² in standard exposure mode for 60 seconds to induce maximum stress in the composite.

Optical calibration

To obtain the photoelastic constant ($f_\sigma$) of Araldite resin, optical calibration was performed. Briefly, a beam of photoelastic resin was loaded using the four-points bending method, which produces stress patterns of pure bending, generating horizontal isochromatic fringes from a neutral axis on the center of the beam. According to a plot of 3 Pa versus $h^2N$, the mean coefficient value was $f_\sigma = 11271$ N/m.

Photoelastic analysis

Photoelastic images were recorded with a 2.1-mega-pixel digital camera (FD Mavica, Sony model MVC-FD97, Tokyo, Japan) at three time points (immediately after light activation and 24 hours and 7 days post-polymerization) in a plane photoelastic device with exact filter angulations (90 degrees) to obtain constant colored fringes. Next, a red filter was placed between the light source and the first polarized lens to obtain constant light and dark fields on images. Fringes formed on the photoelastic resin were qualitatively analyzed based on the Stress-Optical Law.

Photoelastic measurements

Polymerization shrinkage stress data (in MPa) were calculated based on the methodology of Ernst et al.\(^4\) Basically, three measurements of the first-order photoelastic fringe obtained from Araldite discs were obtained from each specimen ($n = 5$) on monochromatic images using UTHSCSA ImageTool software (University of Texas, San Antonio, TX, USA; available without cost from ftp://maxrad6.uthscsa.edu). Shrinkage stress values ($\sigma_0$) were obtained from the isochromatic ring diameters using the following equation:

$$\sigma_0 = (N_x D_x^2 f_\sigma)/(D_i^2 2d)$$

where $N_x$ is the isochromatic ring order, $D_x$ is the isochromatic ring diameter (mm), $D_i$ is the disc preparation diameter (mm), $f_\sigma$ is the photoelastic constant (N/mm), and $d$ is the disc thickness (mm).

Homogeneity and homoscedasticity tests were performed to confirm normality of the data. The mean of the shrinkage stress values was obtained, and the data were statistically analyzed considering the five groups and the three different times for each group. The same data sets were analyzed for the different times using Repeated Measures ANOVA followed by the Bonferroni test. $P$ values < 0.05 were considered significant.

Results

Each rainbow band corresponded to the first-order isochromatic fringe and, when monochromatic light was used, each line represented a fringe. The fringe appeared when a fixed amount of deformation occurred on the photoelastic disc. Thus, the fringes are directly related to the deformations created in the photoelastic resin by the polymerization shrinkage process.

Qualitative analysis

Images taken under white light showed colored fringe formation around the composite. Greater color variation indicates more stress in the photoelastic material. Photoelastic resin from the group with Silorane showed the lowest stress, as characterized by a yellow fringe formation that darkened over time (Figure 1). This yellow fringe was also observed with Z250 immediately after photoactivation. After 24 hours, fringes were thicker and had greater color variation. After 7 days, the color variation was even more evident. For the experimental composites, the CQ group promoted the highest stress in photoelastic resin (Figure 2). CQ images taken immediately after photoactivation showed fringe stresses with clear colors, indicating a high stress level in the photoelastic disc. Images taken after 24 hours showed an increase in stress, with formation of a new green fringe. This new fringe formed indicated an increasing stress pattern after 7 days. Images generated for the CQ/PPD group immediately after photoactivation showed a pattern indicating lower stress than in the CQ group. After 24 hours, fringes with greater color variation were observed, and stress was slightly increased after 7 days. For the PPD group, the photoelastic
resin showed an orange fringe with some darker areas immediately after photoactivation, characterizing lower stress than in the CQ and CQ/PPD groups. Completely blue fringes were seen at 24 hours, and the beginning of a new fringe formation was observed at 7 days.

Quantitative analysis

Table 2 shows the calculated stress values. No statistical difference between commercial composites was observed upon immediate evaluation (p > 0.05). After 24 hours and 7 days, the Z250 composite showed higher stress than Silorane (p < 0.05). The experimental composites were statistically equivalent at all evaluation times (p > 0.05). Stresses evaluated immediately after photoactivation were statistically lower than those after 24 hours and 7 days (p < 0.05), except for the silorane composite. Stresses observed after 24 hours and 7 days did not differ statistically for any composite tested (p > 0.05).

Discussion

Photoelastic images revealed composite polymerization shrinkage stress. According to the literature, the resulting volumetric contraction is directed towards the center of the material mass, and the distribution of stress is controlled by free surfaces and bond quality.\textsuperscript{11} In this study, the isochromatic fringe pattern was the result of traction forces that the composite induced on the photoelastic resin, suggesting that the resulting contraction occurred towards the material center, as previously suggested.\textsuperscript{11}

Commercial composites showed lower polymerization shrinkage stress than experimental composites. The percentage of organic matrix was higher in the experimental composites (40 wt.\%) than in the Z250 matrix (18 wt.\%, according to the manufacturer). Composites
with high filler content undergo low volumetric contraction because they have less resin matrix content. Composites with low filler content and low contraction may undergo a rapid increase in material stiffness, resulting in high stress values.

The results of this study confirmed that the composite resin matrix influences polymerization shrinkage stress. The Silorane composite showed the lowest stress at the three times evaluated. Statistical analysis demonstrated that Silorane and Z250 showed similar results immediately after photopolymerization, but Z250 promoted higher stress than Silorane after 24 hours and 7 days. These two composites mainly differ in the monomer used in the organic matrix. Silorane is composed of silorane monomers, whereas Z250 is composed of dimethacrylate monomers such as Bis-GMA, Bis-EMA, and UDMA. Dimethacrylate monomers are polymerized by C=C double bond reactions, resulting in decreased space between the chains and consequently an increase in volumetric contraction. In contrast, the cationic ring-opening reactions in silorane materials are responsible for the decreased polymerization stress, generating low stress shrinkage.

Although there were no statistically significant differences between experimental composites, qualitative analysis of images suggested that CQ composites generated more stress in photoelastic resin than CQ/PPD or PPD composites immediately after photopolymerization. This discrepancy between qualitative and quantitative results is probably due to the difficulty in measuring some images to obtain accurate quantitative results. Some monochromatic images did not present clear fringe edges, which may have compromised the operator’s ability to measure isochromatic fringe diameters.

The CQ absorption spectrum ranges from ~400 to 500 nm with an absorption peak at 468 nm, whereas PPD has a relatively wide absorption spectrum with an absorption peak at approximately 392 nm. The spectrum emitted by LED units produces more energy within the CQ absorption spectrum. Thus, CQ activation was probably more efficient, so that the CQ polymerization reaction was initiated more quickly than the PPD reaction, creating fringes with greater color variation or higher polymerization stress immediately after photopolymerization. Fringes formed at 24 hours and 7 days were similar for the different initiation systems. Photopolymerization was performed using a third-generation LED unit with secondary accessory LEDs that emit a second peak of energy at 402 nm. This probably contributed to enhanced PPD activation.

Moreover, PPD and CQ reactions are probably initiated using different mechanisms. CQ operates predominantly by proton abstraction of the amine hydrogen, whereas PPD can undergo photocleavage and proton abstraction, which forms free radicals more efficiently. Apparently, this had no effect on polymerization stress in this study, because PPD composites and CQ composites generated similar stress values. Thus, it appears that both photoinitiators were able to promote double link conversion; although PPD reduced the reaction velocity, this did not decrease total composite polymerization shrinkage.

Polymerization stress can be reduced without decreasing the degree of conversion when the flow capacity of the composite is increased. In the present study, this appears to be less apparent in composites with CQ alone. Aside from the slower polymerization rate, other factors such as lower elastic modulus and glass transition temperatures may reduce shrinkage stress.

The composite with CQ/PPD showed intermediate
stress fringes compared with composites with CQ or PPD alone, but the stress values were statistically similar. This probably occurred because the combination of CQ/PPD led to an intermediate polymerization rate at the beginning of the reaction, but to a rate similar to that of CQ and PPD alone at the end of the polymerization process. This result is inconsistent with a previous study reporting a synergistic effect between CQ and PPD, suggesting that one could expect more efficient use of photon energy.

Not all polymerization shrinkage occurs immediately after light activation. Our results showed that stress increased at 24 hours and remained constant at 7 days for all composites except Silorane. The contraction observed after the light source was removed may be attributed to progressive cross-linking reactions that occur after photoactivation was concluded (post-cure), and to thermal contraction due to loss of radiant heat. Considering that the light exposure time was similar for all groups, thermal contraction may have occurred for all composites.

Photoelastic analysis was appropriate for evaluating polymerization shrinkage stress. However, the limitations of this study include the fact that other relevant aspects of stress development could not be monitored (such as kinetic reactions and polymerization rates), and that the degree of conversion and hardness of experimental composites was not evaluated. Future studies that address these limitations will improve our understanding of polymerization shrinkage using different composites.

Conclusion

Within the limitations of this study, we conclude that polymerization shrinkage stress could be minimized by using silorane-based composites, which may cause less gap formation. Experimental composites containing PPD showed a slight potential for minimizing polymerization shrinkage stress immediately after photoactivation.

Acknowledgments

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References


