INTRODUCTION

The solid electrolytes are materials that block the passage of liquids and gases, but allow the transport of ions through the network when there is a tendency to diffusion. The conductor’s oxygen ions are extensively studied and used for a variety of applications, such as oxygen sensors and fuel cells. Solid electrolytes, also known as fast ionic conductors, are a class of materials with significantly higher low-temperature conductivity than would be expected from a simple statistical assumption regarding the number of vacancies. The measured oxide ion conductivities were indeed quite high, and the transport number near unity [1-3]. Abraham’s report spurred a flurry of other papers reporting various partial substitutions of other elements for the Bi, V, or both metals in the structure, and the resulting modifications to the ionic conductivity [4].

The high oxide ion conductivity reported for the BiMeVOX family is believed to be due to the disorder of the oxygen vacancies that are associated with the vanadium atoms in the perovskitic layer. Mairesse [5] has summarized results for the substitutions of various metal cations onto the vanadium site.

This study was based in ceramic solid electrolytes BiMeVOX (Bi: bismuth, V: vanadium, OX: oxygen and Me: Cu) for conducting oxygen ions, using the method of fusion of oxides as technique of synthesis of the main phase whose compositions are based on the use of Cu²⁺ metal ions. BiCuVOX has potential applications as a fast ion conducting component in solid oxide fuel cells, oxygen gas sensors and electrochemical pumps for oxygen separation. This material is derived from Bi₄V₂O₁₁ where vanadium is partially substituted by copper. The gamma phase of Bi₄V₂O₁₁ is the ionic conducting phase. The gamma phase is unstable but can be stabilized by partial substitution with copper or cobalt [6].

MATERIALS AND METHODS

The materials used as basic reagents in the formulation of BiCuVOX sample were bismuth oxide (Bi₂O₃), vanadium oxide (V₂O₅), copper oxide [Cu(NO₃)₂], and titanium oxide (TiO₂), all from Aldrich. The powders of crystalline oxides were be prepared with compositions based on the addition of Cu²⁺ at BiMeVOX the main stage by fusion of oxides technique. These powders are milled in alcohol medium (balls of zirconia). The samples were compressed in the cylindrical pellets by
isostatic pressing with a size of approximately 10 mm in diameter and 3 mm thick. Before, the sample were submitted to thermal treatment carried out in air, for appropriated time, temperature and heating rate.

The potential barrier analysis was performed in an atomic force microscope Nanoscope IIIA (Digital Instr.) operating in a electric force microscopy mode (EFM) that was equipped with an extender electronic module (Veeco Instr.). Topographical measurements and electrical data were obtained by the two-pass technique (lift mode). In this configuration, during the first pass, the probe (operating in tapping mode) scans a topographical line. In the second scan, the cantilever is lifted to a predefined distance (75 nm) in order to minimize the effect of the van der Waals forces, during which it detects variations in the electrical force gradient over the same line and the influence of surface topography is ruled out. A NSC15 tip (MikroMasch) was used in all the experiments. Electrostatic force gradient images were obtained by monitoring the shifts in phase and frequency variations as a function of bias voltages applied to the cantilever. The initial EFM imaging conditions were: interleave frequency drive, 25 Hz; integral gain, 0.35; proportional gain, 2.5. The images of surface potential and barrier layer were obtained by applying 4, 8 and 12 V \textit{in situ} to the sample. Imaging was carried out at room temperature.

**RESULTS AND DISCUSSION**

The microstructural and electrical properties analysis of BiCuVOX sample was analyzed by scanning probe microscopy (SPM): AFM combined with EFM. The results were analyzed and interpreted using the software DI.

Fig. 1 shows images obtained via AFM/EFM of BiCuVOX when subjected to different voltages. Fig. 1a provides topographic information. The microstructure shows no pores. The grains appear to be homogeneous which indicate that the densification occurred completely.

The EFM results are presented in Fig. 1 (b, c, d, e) with application of external voltage from 4 to 12 V, showing details about the variations in the electric field gradient. The

![Figure 1: Analysis of BiCuVOX by AFM/EFM: (a) 3D topography image (10 x 10 x 1.2) \( \mu \text{m} \); 3D profiles EFM with application of external voltage (10 x 10) \( \mu \text{m} \): (b) 4 V (c) 8 V, (d) 12 V.](image-url)

(Figura 1: Análise de BiCuVOX por AFM/EFM: (a) imagem topográfica 3D (10 x 10 x 1.2) \( \mu \text{m} \); perfis 3D EFM com aplicação de voltagem externa (10 x 10) \( \mu \text{m} \): (b) 4 V (c) 8 V, (d) 12 V.)
The interface charge changes the Fermi level in the vicinity of the grain boundary, with band bending as a result. The electronic charges stored in an interface represent a repulsive potential for the majority carriers - the electrons in the case of an n-doped semiconductor - across the interface.

CONCLUSION

The use of atomic force microscopy techniques, electric force microscopy, shows a powerful tool in analyzing phenomena associated to grain boundary regions. The potential barriers formed at grain boundaries due segregation of dopants in these regions, as stated in the literature, could be imaged in situ. The potential barriers, due electrical forces developed at interfaces, detected by EFM, are in agreement with theoretical models proposed in the literature.

ACKNOWLEDGEMENTS

The authors gratefully acknowledge the Brazilian agencies FAPESP and CAPES for their financial support.

REFERENCES

(Rec. 27/09/2010, Ac. 22/01/2011)