Effect of calcium phosphate addition on sintering of El-Oued sand quartz raw materials

(Efeito da adição de fosfato de cálcio na sinterização de matérias-primas de quartzo da areia de El-Oued)

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Abstract

This work addresses the development of an eventual low cost ceramic insulating or microfiltration membrane supports from inexpensive raw materials such Eloued quartz sand (EQS) and calcium phosphate (CP) using uniaxially dry compaction method. The prepared samples were sintered at different temperatures ranging between 1200 and 1550 °C. Subsequently, the effects of sintering temperature and amount of CP on samples proprieties were investigated. It is observed that X-ray diffraction measurements confirmed that EQS was transformed into cristobalite and tridymite phases when both sintering temperature and holding time were increased. This transformation is favored with increasing amount of CP; it is observed also formation of CaSiO3 and Ca(Fe)(Al)PO4. The SEM images of the samples sintered at different temperatures illustrate that silica grains, CP grains, intergranular phase contents (Si, Ca, P, and O) and an important open porosity depend on the sintering temperature. The porosity ratio changes in the range between 37% and 34% according to sintering temperature and to CP content; these values are in good agreement with SEM images. The thermal expansion behavior shows a weak expansion in the range of temperature between 600 and 1000 °C which is situated between 1.27% and 1.33% (variation of 0.05% at 400 °C).

Keywords: quartz sand, calcium phosphate, sintering, porosity, cristobalite.

INTRODUCTION

Silica is a mineral used in many domains. The traditional view is that there are three distinct families of silica structures which are stable at ambient pressure: quartz, tridymite and cristobalite. Although quartz is believed to be the stable phase at temperatures below 870 °C, both tridymite and cristobalite can be cooled to low temperatures, where they are metastable. The conversion of quartz into tridymite can be occurred at temperatures more than 870 °C with mineralizer. It seems besides that the temperature of transition depends on nature and concentrations of the mineralizer. The transformation of pure quartz, under the only action of the temperature, always leads to cristobalite formation. Theoretically, the temperature must exceed 870 °C; practically it is generally necessary to reach 1000 °C to put into evidence the presence of cristobalite; and the speed of transformation increases with the sintering temperature. However, the cristobalite maintained at a temperature ranging between 870 and 1470 °C is transformed into stable tridymite in this temperature range; the transformation rate is always much smaller than that of quartz to cristobalite [1]. The
use of silica in the form of bricks involves a sintering and a transformation of quartz to cristobalite and tridymite at higher temperatures. Foreign oxides additions may be used in order to promote this phase transformation [2]. This work deals with the effect of calcium phosphate (CP) raw material on the sintering and transformation of El-Oued sand quartz (ESQ) in order to prepare an eventual low cost ceramic microfiltration membrane supports or a thermal insulation material from inexpensive raw materials. In this way, Algeria is one of the countries in the world that have abundantly available raw materials. Mainly: calcite \((\text{CaCO}_3)\), dolomite \((\text{CaCO}_3 \cdot \text{MgCO}_3)\), bones (natural derived hydroxyapatite: \(\text{HA} (\text{Ca}_{10} (\text{PO}_4)_6 (\text{OH})_2)\)), kaolin, feldspar and quartz. Many works have already been published for valorizing these native raw materials. These topics concern advanced ceramics [3-9], ceramic membranes [10-16] and bioceramics [17-24].

**EXPERIMENTAL PROCEDURES**

The used raw materials are very rich sand crystalline silica as a quartz (Eloued sand, Algeria) and a calcium phosphate mineral (Bir Elater, Algeria) CP. Their chemical compositions in wt% are given in Table I.

Fig. 1 shows a schematic drawing of the main steps of the samples preparation route, used in this work. Firstly, three mixtures were prepared; the first contains 90 wt.% EQS and 10 wt.% CP whereas the second one contains 80 wt.% EQS and 20 wt.% CP and the third contains 70 wt.% EQS and 30 wt.% CP. The materials were then ground with water in a planetary mill for 1 h and dried at 100 °C. The mixtures are compacted in the form of discs (13 mm in diameter: and 5-6 mm in thickness) at 140 MPa. Subsequently, the compacted samples were fired for 2 and 4 h at temperatures ranged between 1200 and 1550 °C at a heating rate of about 5 °C/min.

The phase composition of samples was determined on a derived configuration equipments (INEL CPS120 curved position sensitive detector). CuKα radiation was used at the operating condition of 40 kV and 20 mA. The XRD data was collected over the 2θ range of 15-70° with a step size of 0.02 degree/min. Identification of phases was achieved by comparing the diffraction patterns of the composites with the JCPDS files. The morphology and the microstructure of sintered samples were observed on polished cross sections using a scanning electron microscope (SEM) equipped with EDAX analyses system (SEM, Cambridge Instr.). Before observation and EDAX analyses, the samples were carbon coated. The starting materials were characterized by thermal analyses (DTA/TGA, Setaram 1750 Setsys) during heating from room temperature to 1500 °C at a heating rate of 5 °C/min in air. The expansion of bars (5 mm x 5 mm x 5 mm) taken from compacted cylinders was characterized during heating from room temperature to 1500 °C at a heating rate of 5 °C/min under air, using a DI24 dilatometer. The open porosity of sintered samples was determined using the water absorption method (ISO/TS 17892-2:2004). There were calculated using the following equation

\[ e = (M3-M1/M3-M2) \times L \]  

\[ (A) \]

\( e \) total porosity, \( M1 \) sample weight m, \( M2 \) mass of the sample immersed in water, \( M3 \) mass of the sample impregnated, \( L \) density of water at the specified test temperature.

**RESULTS AND DISCUSSION**

Fig. 2 presents the evolution of the transformation of phases in samples containing 10 wt.% CP as a function of the sintering temperature. It is observed that firstly there is a transformation of quartz from cristobalite at 1200 °C and wollastonite is formed. From 1250 °C, tridymite appears, it is may be due to the transformation of cristobalite and the

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**Table I - The chemical composition of EQS, using ICP spectroscopy.**

<table>
<thead>
<tr>
<th>Composition</th>
<th>SiO₂</th>
<th>Al₂O₃</th>
<th>Fe₂O₃</th>
<th>CaO</th>
<th>MgO</th>
<th>Na₂O</th>
<th>K₂O</th>
<th>P₂O₅</th>
</tr>
</thead>
<tbody>
<tr>
<td>EQS</td>
<td>91.24</td>
<td>1.74</td>
<td>2.74</td>
<td>3.15</td>
<td>0.26</td>
<td>0.09</td>
<td>0.63</td>
<td>0</td>
</tr>
<tr>
<td>CP</td>
<td>7.27</td>
<td>1.13</td>
<td>0.31</td>
<td>48.04</td>
<td>0.23</td>
<td>1.05</td>
<td>0.78</td>
<td>40.85</td>
</tr>
</tbody>
</table>
rate of the transformation increases with the increase in the sintering temperature. The transformation will be completed in a same time. This observation is in a good agreement with the literature. Indeed, using these raw materials may enable to develop tridymite as a major phase component and becoming one of certain procedures adopted industrially for silica refractories [25].

Fig. 3 shows the effect of CP additions on phase transformation of samples sintered at 1300 °C for 4 h. It is observed that the transformation is more important in the case of 30 wt.%, tridymite appears in more quantity. So, there is a new phase Ca(Fe)(Al)PO$_4$ which is formed by reaction between tricalcium phosphate (TCP) and oxides. This phase is detectable in the case of 30 wt.% due to elevated CP amounts. In a previous work using pure TCP with quartz, it has been found that the transformation of quartz into cristobalite was only observed without any tridymite appearance. These observations may allow believing that the TCP plays a very important role in the transformation into cristobalite and the transformation into tridymite is may be caused by impurities. This effect was observed in a work carried out on refractories containing zirconia (ZrO$_2$) [26]. A similar effect of the phosphate addition on transformation from µ to α cordierite was studied [27].

These CP additions may allow the formation of more porous volume. A typical overview of the microstructure of samples containing 10 and 30 wt.% CP sintered at 1300 °C for 4 h is illustrated in Fig. 4. This micrograph shows well the distribution mod of the pore size within the matrix, the open and the communicating pores throughout the sintered samples.

A careful exam at the SEM microstructures shown in Fig 4, one can notice that there are three distinguishable phases or grains: white grains, small white grains and grey grains.
O with small amounts of Mg. They should correspond to CP. Since the small white grains are distributed on the grey grains, they contain Si, Ca, O and a small P. One can suggest that there is a decomposition of the same CP into CaO and P$_2$O$_5$. Consequently, CaO reacts with SiO$_2$ to form wollastonite and the volatilization of P$_2$O$_5$. That explains the decrease in P peak intensities shown in Fig. 5D. By increasing the temperature, this phenomenon continuous, thus samples sintered at 1500 °C for 4 h do not contain CP grains. They are formed by two types of grains. The first
type contains Si and O which constitute tridymite and/or cristobalite phases. However, the others contain Si, Ca, O and a few of P which may correspond to wollastonite and a composite of phosphate (Fig. 3).

Fig. 6 shows the expansion curve of samples sintered at 1200 °C for 2 h. It is observed that at 220 °C expansion results from transformation of α cristobalite to β cristobalite, whereas at 567 °C the expansions may be due to the transformation of residual α quartz to β quartz. Afterwards, the transformation of the residual quartz to cristobalite and tridymite continues between 1200 and 1400 °C, which explains the expansion in this range. During cooling, expansion is very low from 600 to 1400 °C in the case of 10 wt.%. In the case of 30 wt.%, the transformation is completed and there is no a reversible transformation from β to α and expansion is very low in the range between 800 and 1400 °C.

The thermal behavior of EQS + 30 wt.% in terms of DTA and TG are illustrated in Fig 7. There is no loss in weight observed until 1200 °C. Subsequently, a loss starts within a small amount between 1200 to 1400 °C and increases rapidly. This increase may be due to a decomposition of a small amount of CP and P$_2$O$_5$ volatilization. This result is also confirmed by EDX analyses shown in Fig. 5. In fact, this loss explains an increase in a total porosity in samples sintered at a temperature above 1400 °C. A similar observation is reported [28]. The DTA curve shows a sharp endothermic peak at 570 °C on heating step, attributed to the α –β quartz transformation and an exothermic peak at 220 °C on cooling step, attributed to the β-α cristobalite transformation. At 1240 and 1260 °C there are two peaks; one endothermic and other exothermic, respectively. These two peaks may be attributed to the transformation of tri-calcium phosphate (TCP) (β to α on heating step and α to β on cooling step.

The total porosity as function of sintering temperature is shown in Fig. 8. In these three cases, the total porosity decreases slightly in the range between 1200 and 1400 °C. Afterwards, it increases slightly up to 1550 °C.

For example, the porosity ratio values are ranged between 37 and 33% for the three sets of samples. This decrease is closely related to the sintering phenomena whereas the increase is may be due to decomposition of CP and the P$_2$O$_5$ vaporization. Moreover, the same porosity ratio was obtained [29] using quartz and zeolite without any pre-sintering.

**CONCLUSIONS**

From the X-ray diffraction analyses it can be concluded that the CP addition plays a very important role as a mineralizer that promotes transformation and stabilization phase. It acts as a binder and allows achieving acceptable sintered samples within a considerably higher porosity ratio. It is believed that according to the currently chemical mapping there is a reaction between the CP and silica to form a composite which contains Si, P, Ca and O.

**REFERENCES**


(Rec. 06/08/2014, Ac. 07/10/2014)