Crystallization of Anatase TiO$_2$ in Niobium Potassium Phosphate Glasses

Carolina Dakuzaku Freschi$^a$, José Tadeu Gouveia$^a$, Lia Marcondes$^a$, Jefferson Luis Ferrari$^b$, Fábia Castro Cassanjes$^a$, Gael Poirier$^{a*}$

$^a$ Research Group of Materials Chemistry, Universidade Federal de Alfenas, Campus de Poços de Caldas, Poços de Caldas, MG, Brazil
$^b$ Departamento de Ciências Naturais, Universidade Federal de São João Del Rei, Campus Dom Bosco, Praça Dom Helvécio, 74, CEP 36301-160, São João Del Rei, MG, Brazil

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In this work, the glass forming ability was studied in potassium phosphate glasses with increasing amounts of TiO$_2$ in order to obtain a glass-ceramic with photocatalytic properties. The first studied series has been the binary system (100-x) KPO$_3$-xTiO$_2$. Homogeneous and transparent glasses could be obtained with x varying from 10 to 30 mole%. Since the photocatalytic anatase phase could not be precipitated in this system, the ternary system KPO$_3$-TiO$_2$-Nb$_2$O$_5$ was investigated in order to incorporate higher TiO$_2$ contents without spontaneous crystallization under cooling. Thermal properties of all glass samples were investigated by DSC and allowed identifying an increase of glass transition temperatures with increasing TiO$_2$. For all compositions, exothermic events related with crystallization were also observed and suitable heat-treatments resulted in specific crystalline phases identified by X-ray diffraction. Selective precipitation of the anatase titanium oxide was successfully obtained from the glass composition 35KPO$_3$-25Nb$_2$O$_5$-40TiO$_2$ (KN25T40) after heat treatment at 720°C for 2 h, suggesting the possibility of obtaining a glass-ceramic for photocatalytic applications. Structural investigations by Raman were also performed on glasses and glass-ceramics and allowed to point out the glass intermediary behavior of TiO$_2$ in the phosphate vitreous network where TiO$_4$ and TiO$_6$ octahedra are inserted inside the phosphate network with TiO$_6$ clusters identified at higher TiO$_2$ contents. Raman analysis also identified anatase TiO$_2$ in the KN25T40 glass-ceramic.

Keywords: glass; phosphate; titanium oxide; anatase

1. Introduction

Titanium dioxide is considered one of the most important semiconductor oxide with interesting chemical, electrical and optical properties. Since its discovery as photocatalytic agent observed by Fujishima and Honda in 1972, research on TiO$_2$ applications has attracting considerable interest in several fields of science due to its excellent properties, including chemical stability, photostability and appropriate electronic band structure$^{1,2}$.

Titanium dioxide thin films have been formed on glass, steel and other surfaces by a wide range of techniques, especially by sol-gel and chemical vapour deposition$^{3,4}$. However, the properties of these materials produced by deposition or coating techniques can change over time by surface damage and thus a recoating process can be necessary. On the other hand, TiO$_2$ crystallites precipitated in the glass matrix can exhibit stable physical and chemical properties even with surface polishing$^5$.

There are three allotropic forms of TiO$_2$: anatase (tetragonal), brookite (orthorhombic) and rutile (tetragonal) among which rutile is the most thermodynamically stable phase for bulk TiO$_2$ under most conditions. However, the rutile activity as a photocatalytic compound is generally poor. Anatase TiO$_2$ is considered to be the active photocatalytic component based on charge carrier dynamics, chemical properties and the activity of photocatalytic degradation of organic compounds$^{1,3}$.

When a glass-ceramic is obtained by heat-treatment of the mother glass, both the disordered glass regions and ordered crystalline regions are presents. The final material can exhibit not only the benefits of the glass material but also the unique physical properties originated from the crystalline phase. However, only a few works reported phosphate glass-ceramics containing anatase TiO$_2$ in crystal volume fraction by heat treatment and this lack is in part attributed to the difficulty of promoting controlled and selective crystallization of anatase in phosphate-based glasses with high TiO$_2$ contents since titania is known to act as a nucleating agent of other crystalline phases$^6$.

Selective crystallization of the glass CaO-Bi$_2$O$_3$-B$_2$O$_3$-Al$_2$O$_3$-TiO$_2$ containing 20% TiO$_2$ under heat-treatment using an infrared halogen lamp is reported but allowed only precipitation of the rutile phase$^7$. Recently, selective crystallization of TiO$_2$ by controlled heat-treatment and SiO$_2$
addition in the glass \(14\text{TiO}_2\cdot23\text{ZnO}\cdot45\text{B}_2\text{O}_3\cdot18\text{Al}_2\text{O}_3\) was also reported but XRD results indicate that TiO\(_2\) was not the unique crystalline phase after heat treatment and Al_2B_2O_7 was precipitated as a subcrystalline phase.

In this work, TiO\(_2\)-containing glass compositions were investigated for preferential anatase crystallization. First, the binary system \((100-x)\text{KPO}_3\cdot x\text{TiO}_2\) was investigated and glass samples were obtained from \(x=10\) to \(x=30\). These glasses were characterized by DSC and Raman spectroscopy and specific heat-treatments were performed but anatase could not be precipitated under heat-treatments. Adding Nb_2O_5 to this binary system allowed incorporating higher TiO\(_2\) contents (40%) and anatase TiO\(_2\) could be successfully precipitated in the mother glass.

2. Experimental Part

Glass samples were synthesized from the precursors: titanium dioxide 99.9% (TiO\(_2\)) from Vetec, potassium phosphate monobasic 99.9% (KH\(_2\)PO\(_4\)) from Synth and niobium oxide 99.8% (Nb_2O_5) from Sigma-Aldrich by conventional melt-quenching: The starting powders were weighted using an analytical balance and grinded in an agate mortar. The batches were melted for 2 hours in a platinum crucible between 1100 and 1450°C depending on the TiO\(_2\) content. Finally, the melts were quenched in a steel mold preheated 20°C below the glass transition temperature and kept at this temperature for 8 hours before slow cooling inside the furnace. First, samples in the binary system \((100-x)\text{KPO}_3\cdot x\text{TiO}_2\) with \(x\) varying from 10 to 30 mole% were prepared using the methodology described above. In a second step, a ternary system was investigated with the introduction of niobium oxide as glass intermediary in order to increase the TiO\(_2\) content in the final glass using the same methodology. Molar compositions, melting temperature and visual aspect of the glass samples are presented in Table 1. DSC curves were performed on bulk glass samples of 30 mg in Pt/Rh covered crucibles between 200°C and 1100°C at 10°C/min under N\(_2\) atmosphere. These thermal analyzes were obtained using a DSC/TG calorimeter STA 449 F3 Jupiter from Netzsch. X-ray diffraction measurements were performed on powder samples using a Rigaku ultima IV diffractometer working at 40KV and 30mA between 10° and 70° in continuous mode of 0.02°/s. The crystalline phases were identified according to X-ray powder diffraction patterns (PDF file). Finally, Raman spectra were collected on bulk glass samples between 100cm\(^{-1}\) and 1200cm\(^{-1}\) using a LabRam Micro-Raman from Horiba Jobin-Yvon operating at 632,8 nm with a He-Ne laser.

3. Results

Transparent and homogeneous samples were obtained in the binary system \((100-x)\text{KPO}_3\cdot x\text{TiO}_2\) with \(x\) varying from 10 to 30 mole% by melt-quenching with an increasing brownish color for higher TiO\(_2\) contents. As expected, an increase of the melt viscosity with increasing TiO\(_2\) content was observed during synthesis and higher melting temperatures were used for a suitable casting of the melt. However, partial or total crystallization of the samples was observed for TiO\(_2\) contents above 30 mole% under these experimental conditions. After X-ray diffraction analysis of the crystalline phases obtained by heat-treatment in these glasses, Nb_2O_5 was added and samples containing 40 mole% of TiO\(_2\) could be prepared. In fact, several compositions were tested by varying the \(\text{KPO}_3:\text{Nb}_2\text{O}_5\) ratio (50:10; 40:20; 35:25; 30:30) with constant TiO\(_2\) content of 40%. Compositions 50\text{KPO}_3\cdot10\text{Nb}_2\text{O}_5\cdot40\text{TiO}_2\) and 30\text{KPO}_3\cdot30\text{Nb}_2\text{O}_5\cdot40\text{TiO}_2\) crystallized under quenching while compositions 40\text{KPO}_3\cdot20\text{Nb}_2\text{O}_5\cdot40\text{TiO}_2\) and 35\text{KPO}_3\cdot25\text{Nb}_2\text{O}_5\cdot40\text{TiO}_2\) successfully vitrified. Table 1 resumes all investigated compositions, required melting temperatures and the visual aspect of the final samples. DSC curves for all bulk glass samples are presented in Figure 1 and Table 2 resumes the characteristic temperatures T_g, T_x and T_f where T_g is the glass transition temperature, T_x the onset of crystallization and T_f the melting temperature as well as the thermal stability parameter T_x-T_g. These thermal data point out that the addition of TiO\(_2\) results in a significant increase in T_g values from 330°C to 510°C in the binary system KPO\(_3\)\cdot TiO\(_2\) with an increase of the thermal stability parameter from 10% to 20% and further decrease for higher contents. Nb_2O_5 incorporation also results in a clear increase of T_g identified around 650°C for 20 and 25 mole% of niobium oxide. Crystallization events were

### Table 1. Molar compositions, characteristic temperatures and visual aspect of the glass samples.

<table>
<thead>
<tr>
<th>Sample</th>
<th>Molar Compositions (% mol)</th>
<th>Melting Temperature (°C)</th>
<th>Picture</th>
</tr>
</thead>
<tbody>
<tr>
<td>KT10</td>
<td>(90) (\text{KPO}_3), (10) (\text{TiO}_2), ___</td>
<td>1100</td>
<td><img src="image" alt="KT10" /></td>
</tr>
<tr>
<td>KT20</td>
<td>(80) (\text{KPO}_3), (20) (\text{TiO}_2), ___</td>
<td>1200</td>
<td><img src="image" alt="KT20" /></td>
</tr>
<tr>
<td>KT30</td>
<td>(70) (\text{KPO}_3), (30) (\text{TiO}_2), ___</td>
<td>1300</td>
<td><img src="image" alt="KT30" /></td>
</tr>
<tr>
<td>KN25T40</td>
<td>(35) (\text{KPO}_3), (40) (\text{Nb}_2\text{O}_5), (25) (\text{TiO}_2)</td>
<td>1450</td>
<td><img src="image" alt="KN25T40" /></td>
</tr>
<tr>
<td>KN20T40</td>
<td>(40) (\text{KPO}_3), (40) (\text{TiO}_2), (20) (\text{Nb}_2\text{O}_5)</td>
<td>1500</td>
<td><img src="image" alt="KN20T40" /></td>
</tr>
</tbody>
</table>
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K₄P₂O₇ and cyclic potassium metaphosphate K₃P₃O₉. For sample KT30, phases KPO₃ and KTiOPO₄ were detected. Finally, for samples modified with niobium oxide, anatase TiO₂ could be identified as the first precipitated phase in sample KN25T40 heat-treated below the first crystallization peak whereas heat-treatments at higher temperatures induce crystallization of the mother glass with formation of K₂TiNb₂P₂O₁₃. For the other niobium titanium phosphate glass KN20T40, crystalline phases K₂TiNb₂P₂O₁₃, K₂Nb₆P₄O₂₆, and rutile TiO₂ were identified.

Raman spectra of the glass samples and crystalline references KPO₃, TiO₂, and Nb₂O₅ in the frequency region between 100 and 1500 cm⁻¹ were recorded to probe the structural evolution with TiO₂ addition and to compare the Raman features of precursor glass and glass-ceramic KN25T40 (Figure 4). In the binary system KPO₃-TiO₂, the first clear change in Raman spectra is the progressive vanishing of vibrational bands centered at 1150 cm⁻¹ and 670 cm⁻¹ and commonly attributed to symmetric stretching modes of P-O bonds in P-O-P linkages and Q₂ tetrahedra in metaphosphate compounds. On the other hand, TiO₂ addition results in new Raman signals centered around 1250 cm⁻¹ and 520 cm⁻¹ which shifts to lower wavenumbers. These signals are reported to be due to P=O terminal bonds in PO₄ tetrahedra wherein the other three oxygens are linked to another cation (P-O-X where X=P or Ti) and vibrational modes of distorted TiO₆ respectively. Other Raman features appear with TiO₂ addition at 290 cm⁻¹, 735 cm⁻¹, 863 cm⁻¹, 935 cm⁻¹, and 977 cm⁻¹ and were attributed to bending modes of TiO₆ octahedra, stretching modes of TiO₄, stretching modes of TiO₆, Ti-O stretching of the axial Ti-O bond in TiO₂ pyramidal units, and PO₄ orthophosphate units respectively. Precursor glass composition KN25T40 is dominated by broad Raman bands centered around 200 cm⁻¹, 650 cm⁻¹, and 825 cm⁻¹ as for crystalline Nb₂O₅ whereas the glass-ceramic exhibits sharper Raman signals at 240 cm⁻¹ and 640 cm⁻¹ as for crystalline anatase TiO₂.

**Table 2.** Characteristic temperatures, thermal stability and crystalline phases identified in heat-treated glass samples.

<table>
<thead>
<tr>
<th>Samples</th>
<th>Characteristic temperatures (°C)</th>
<th>Thermal stability parameter</th>
<th>Crystalline phase after heat-treatments</th>
</tr>
</thead>
<tbody>
<tr>
<td>KT10</td>
<td>330 475 620 - 780 145</td>
<td></td>
<td>KPO₃</td>
</tr>
<tr>
<td>KT20</td>
<td>440 705 740 - 770 140</td>
<td></td>
<td>K₂P₂O₇</td>
</tr>
<tr>
<td>KT30</td>
<td>510 650 660 - 770 140</td>
<td></td>
<td>KPO₃</td>
</tr>
<tr>
<td>KN20T40</td>
<td>670 730 - 770 140 60</td>
<td></td>
<td>Rutile TiO₂</td>
</tr>
<tr>
<td>KN25T40</td>
<td>650 720 790 870 70</td>
<td></td>
<td>Anatase TiO₂</td>
</tr>
</tbody>
</table>

- Event not observed.
4. Discussion

Since the main objective of this work was the obtaining of glass-ceramics containing anatase crystallites for photocatalytic applications, the glass forming ability of the binary system KPO$_3$-TiO$_2$ was investigated with increasing amounts of TiO$_2$. Homogeneous glass samples were obtained by melt-quenching from TiO$_2$ contents ranging from 10 to 30 mole% whereas higher contents lead to crystallized samples under these experimental conditions. As presented in Table 1, these glasses also exhibit an increasing brownish color attributed to partial titanium reduction from Ti$^{4+}$ to Ti$^{3+}$. It is suggested that the increasing melting temperatures can induce oxygen loss of the melt and titanium reduction as described for non-stochiometric crystalline titanium oxides. The strong absorption in the visible is due to both d-d internal electronic transition of Ti$^{3+}$ (d$^3$ electronic configuration) and electronic polaron transitions between reduced Ti$^{3+}$ and oxidized Ti$^{4+}$ species. These increasing melting temperatures and increasing viscosity of the resulting melt observed during synthesis bring a first indication of the effective insertion of titanium ions inside the phosphate chains and higher network connectivity and field strength. A detailed structural description can be extracted from a general analysis of thermal results, Raman data and crystalline phase identification obtained by X-ray diffraction. Since the results obtained from each technique point out the same structural evolution, these data are discussed simultaneously. The first important point is the clear increase of glass transition temperatures from 330ºC to 510ºC for TiO$_2$ contents ranging from 10 to 30 mole%. This behavior is commonly related with an increase in the glass network connectivity as well as in the network bond strength as reported in many structural studies concerning transition metal oxide-containing phosphate glasses. In fact, these metallic oxides usually exhibit strong metal-oxygen bonds and high coordination number. In addition, their intermediary behavior in phosphate network usually allow their partial or complete insertion inside the phosphate covalent chains, resulting in cross-linking bonds between these chains. In our case, the Tg increase with composition can be attributed to a progressive insertion of titanium oxide units such as TiO$_4$ and/or TiO$_6$ between PO$_4$ tetrahedra. These species were identified by the Raman bands observed at 863cm$^{-1}$.
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Centered at 863 cm$^{-1}$ and attributed to TiO$_4$ units enhances from 10% to 20% of TiO$_2$ but further decreases for 30% but a new band at 935 cm$^{-1}$ clearly appears for KT30. Since this Raman signal is known to be due to the axial Ti-O bond in pyramidal TiO$_5$ units, it can be suggested that titanium ions are preferentially four-fold and six-fold coordinated for lower TiO$_2$ contents whereas TiO$_6$ and TiO$_5$ units dominate for higher titanium oxide concentrations. The precipitation of titanium-containing crystalline phases could not be detected by X-ray diffraction of the heat-treated samples KT10 and KT20 but the thermal behavior of sample KT30 also gives important structural clues explaining the limited glass forming ability of this binary system. The Raman spectrum of this sample is dominated by signals centered at 290, 520, 935 and 1250 cm$^{-1}$ related with TiO$_6$, TiO$_5$ and PO$_4$ units whereas the DSC curve exhibits two crystallization peaks and a melting event. Heat-treatment of this sample resulted in precipitation of crystalline phases KPO$_3$ and KTiOPo$_4$. Since the Tg temperature of this glass is higher than compositions KT10 and KT20, it appears that titanium oxide polyhedra effectively modified the phosphate network through P-O-Ti bonds but the thermal treatment induces a spontaneous phase separation of a potassium metaphosphate structure and potassium titanium orthophosphate constituted of TiO$_6$ octahedra and orthophosphate PO$_4^{3-}$ ions. The KPO$_3$ formation has been detected by X-ray diffraction as well as melting event observed at 770ºC. This phase separation tendency for these higher TiO$_2$ contents is in agreement with the spontaneous crystallization of melts with higher TiO$_2$ contents by melt-quenching.

Anyway, these binary compositions were not able to precipitate TiO$_2$ anatase by heat-treatment and niobium oxide Nb$_2$O$_5$ was added in order to investigate the glass forming ability of compositions with 40 mole% of TiO$_2$. Samples with 10% and 30% of Nb$_2$O$_5$ crystallized under quenching while compositions with 20% and 25% successfully vitrified (KN20T40 and KN25T40). For sample KN20T40, only one crystallization peak was observed by DSC with $T_x$ at 730ºC and heat-treatments were performed at 750ºC and 1050ºC in order to identify the first crystalline phase as well as other possible precipitations not detected by thermal analysis. The first heat-treatment resulted in a diffraction pattern that could not be identified using the ICSD database whereas heat-treatment at higher temperatures allowed characterizing the potassium niobium phosphate K$_3$Nb$_6$P$_4$O$_{26}$, potassium niobium titanium phosphate K$_2$TiNb$_2$P$_2$O$_{13}$ and rutile TiO$_2$. For this composition, these results clearly point out that these crystalline phase precipitation are not related with distinct crystallization peaks and can hardly be selectively precipitated by suitable heat-treatments. The chemical composition of these phases suggests crystallization of the mother glass with rutile TiO$_2$. In any case, anatase TiO$_2$ could not be detected for this composition. On the other hand, sample KN25T40 exhibits three distinct exothermic peaks at 290, 520, and 935 cm$^{-1}$ respectively. The former Raman band is often attributed to Nb-O-P linkages$^{21}$ but is better attributed to TiO$_4$ units in these glasses since niobium free samples prepared in the binary system KPO$_3$-TiO$_2$ also exhibit this signal. The potassium metaphosphate network depolymerisation is also supported by the progressive vanishing of Raman bands at 680 cm$^{-1}$ and 1150 cm$^{-1}$ related with P-O-P and P-O terminal bonds in Q$^2$ tetrahedra. Another important result supporting this structural evolution is the preferential precipitation of potassium metaphosphate KPO$_3$ in sample KT10 at 475ºC which melts at 780ºC whereas sample KT20 exhibits crystallization of potassium pyrophosphate K$_2$P$_2$O$_7$ and cyclic metaphosphate K$_3$P$_3$O$_9$. The other crystallization event observed for KT10 can be related with precipitation of titanium oxide with high melting temperature not observed below 1000ºC. These titanium oxide polyhedra can increase the network connectivity by cross-linked Ti-O-P bonds. In fact, the appearance of the Raman band around 1250 cm$^{-1}$ and attributed to P=O bonds means that all other P-O linkages in these PO$_4$ tetrahedra are bridging bonds like P-O-Ti bonds resulting in this non-resonant P=O species. Based on the well-known glass intermediary behavior of Nb$_2$O$_5$ in phosphate glasses, it is also suggested that niobium oxide incorporation promotes bridging Nb-O-P bonds and higher glass network connectivity, in agreement with higher Tg temperatures for these glasses. In addition, The Raman band centered at 863 cm$^{-1}$ and attributed to TiO$_4$ units enhances from 10% to 20% of TiO$_2$ but further decreases for 30% but a new band at 935 cm$^{-1}$ clearly appears for KT30. Since this Raman signal is known to be due to the axial Ti-O bond in pyramidal TiO$_5$ units, it can be suggested that titanium ions are preferentially four-fold and six-fold coordinated for lower TiO$_2$ contents whereas TiO$_6$ and TiO$_5$ units dominate for higher titanium oxide concentrations. The precipitation of titanium-containing crystalline phases could not be detected by X-ray diffraction of the heat-treated samples KT10 and KT20 but the thermal behavior of sample KT30 also gives important structural clues explaining the limited glass forming ability of this binary system. The Raman spectrum of this sample is dominated by signals centered at 290, 520, 935 and 1250 cm$^{-1}$ related with TiO$_6$, TiO$_5$ and PO$_4$ units whereas the DSC curve exhibits two crystallization peaks and a melting event. Heat-treatment of this sample resulted in precipitation of crystalline phases KPO$_3$ and KTiOPo$_4$. Since the Tg temperature of this glass is higher than compositions KT10 and KT20, it appears that titanium oxide polyhedra effectively modified the phosphate network through P-O-Ti bonds but the thermal treatment induces a spontaneous phase separation of a potassium metaphosphate structure and potassium titanium orthophosphate constituted of TiO$_6$ octahedra and orthophosphate PO$_4^{3-}$ ions. The KPO$_3$ formation has been detected by X-ray diffraction as well as melting event observed at 770ºC. This phase separation tendency for these higher TiO$_2$ contents is in agreement with the spontaneous crystallization of melts with higher TiO$_2$ contents by melt-quenching.
events starting at 720°C, 790°C and 870°C attributed to precipitation of several crystalline phases in the glass. For this reason, heat-treatments were performed at 720°C, 780°C and 880°C and the resulting samples were characterized by X-ray diffraction. As shown in Figure 3 and Table 2, these results allowed attributing the first peak to precipitation of anatase while the second exothermic event is related with crystallization of K₂TiNb₂P₂O₁₅. A stoichiometric analysis of the starting glass composition (35KPO₃-25Nb₂O₅-40TiO₂) corresponding to K₂TiNb₂P₂O₁₅ (average formula) and final crystalline phases TiO₂ and K₂TiNb₂P₂O₁₅ suggest that approximately half of the starting titanium ions is precipitated as anatase resulting in a final composition of the residual glass next to K₂TiNb₂P₂O₁₅ which is close to the composition of the crystalline phase K₂TiNb₂P₂O₁₅. For this composition, it has been shown that anatase can be preferentially precipitated as the first crystalline phase in these glasses resulting in a glass-ceramic containing only anatase crystallites. The Raman data presented in Figure 4 also support this assumption. The Raman spectrum of the precursor glass KN₂ST₄O₄ exhibits intense signals around 230cm⁻¹, 750cm⁻¹ and 850cm⁻¹ also observed for crystalline Nb₂O₅ and attributed to bending and stretching modes of NbO₆ octahedra. The higher polarizability of Nb-O bonds when compared to P-O and Ti-O bonds explains why the former bonds dominate the spectrum when compared to the latter one. On the other hand, The Raman spectrum of the glass-ceramic KN₂ST₄O₄ obtained from heat-treatment of the starting glass at 720°C shows a very distinct signal with dominant narrow bands centered at 150cm⁻¹ and 650cm⁻¹ characteristic of bending and stretching modes of the crystalline phase anatase.

5. Conclusion

Homogeneous and transparent titanophosphate glasses were obtained in the binary KPO₃-TiO₂ and ternary KPO₃-TiO₂-Nb₂O₅ systems by the melt-quenching method. 30 mole% of TiO₂ was successfully incorporated without devitrification in the binary system and the structural changes were monitored by DSC, Raman and X-ray diffraction of the heat-treated glasses. Titanium oxide polyhedra are inserted inside the phosphate chains and increase the network connectivity. However, for higher TiO₂ contents, a phase separation has been identified without precipitation of the desired anatase TiO₂ phase. Addition of Nb₂O₅ as an intermediary compound led to important structural changes, increasing the stability against devitrification and homogeneous glasses containing 40% TiO₂ could be obtained. The glass composition 35KPO₃-25Nb₂O₅-40TiO₂ exhibits a preferential precipitation of anatase in the mother glass resulting in glass-ceramics with a potential photocatalytic activity.

6. Acknowledgments

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7. References

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