Preparation and Characterization of the Structural, Optical, Spectroscopic and Electrical Properties of \( \text{Pr}_2\text{O}_3 \) doped Borate Glass

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We have successfully synthesized \( \text{Pr}_2\text{O}_3 \) doped borate glasses by conventional rapid melt quench method. The XRD pattern indicates the amorphous nature of \( \text{Pr}_2\text{O}_3 \) doped borate glass. An optical property of prepared borate glass was studied using the Photoluminescence spectrum. Determination and differentiation of the various vibrational modes were done using FTIR spectroscopy studies. Raman spectroscopy of \( \text{Pr}_2\text{O}_3 \) doped borate glass was also carried out. Thermal analyses of the glasses were done using the TGA/DTA and DSC analysis. The dielectric properties such as dielectric constant, the dielectric loss and AC conductivity of the \( \text{Pr}_2\text{O}_3 \) doped borate glass were studied in the different frequencies and temperature.

Keywords: \( \text{Pr}_2\text{O}_3 \) XRD, Photoluminescence, FTIR, FT-Raman studies and Dielectric studies

1. Introduction

Rare earth ions have numerous attractive spectroscopic properties in glasses in connection with laser research\(^1\), related applications\(^2\) and basic research\(^3\). The trivalent rare earth ions are easily incorporated in glasses. \( \text{Pr}^{3+} \) doped glasses have transition in the visible region. Some spectral studies have been reported for \( \text{Pr}^{3+} \) doped borate and phosphate glasses earlier\(^4\)\(^5\)\(^6\)\(^7\). Recent studies reported\(^8\)\(^9\) on the applications of \( \text{Pr}^{3+} \) doped borophosphate glasses. In many optical devices like blue up-converters (\( ^{3}\text{P}_0 \rightarrow ^{1}\text{H}_0 \)), solid-state lasers emitting visible (\( ^{1}\text{D}_2 \rightarrow ^{3}\text{H}_0 \)) or near-infrared light (\( ^{3}\text{G}_4 \rightarrow ^{3}\text{H}_0 \)) praseodymium – doped glasses are proved to be effective. In the present work, we report the structural properties of the \( \text{Pr}_2\text{O}_3 \) doped borate glasses that were determined by X-ray diffraction (XRD) analysis and FTIR and FT-Raman analysis. It was confirmed that the prepared glasses were amorphous. The bonding parameters of the glasses were analyzed by using FTIR and FT Raman analysis and were confirmed to be ionic in nature. The optical properties were characterized by using the photoluminescence studies. The electrical properties of \( \text{Pr}_2\text{O}_3 \) doped borate glasses were studied. Since being a member of Lanthanide, Prasuedomium shows efficient luminescence in triplet state. Hence, the prepared borate glass can be used for some applications like optical data reading, colour display etc.

2. Experimental Procedure

2.1. Preparation of Glass Samples

The \( \text{Pr}_2\text{O}_3 \) doped borate glass of composition [\( 60.5\text{B}_2\text{O}_3 + x\text{Li}_2\text{CO}_3 + x\text{ZnO} + x\text{Sr}_2\text{O}_3 + x\text{H}_8\text{NO}_7\text{P} + x\text{Y}_2\text{O}_3 \) + \( \text{Pr}_2\text{O}_3 \) (\( x \) in molecular % ranging from 10 to 50 and \( x \) in molecular % of 0.5] was prepared by melt quenching method. The appropriate quantities of \( \text{B}_2\text{O}_3 \), \( \text{Li}_2\text{CO}_3 \), \( \text{ZnO} \), \( \text{Sr}_2\text{O}_3 \), \( \text{Y}_2\text{O}_3 \) and \( \text{Pr}_2\text{O}_5 \) were weighed and mixed together with a morter, to get fine powders. All powders were taken in 99.99 % of purity. The batches were placed in a porcelain crucible and melted in electric furnace at 1300°C. After finishing the melting process, it transferred to a second furnace which maintained at 400 °C for annealing. The annealing process continued for 3-4 hrs and the sample was gradually cooled under room temperature. The prepared glasses were polished for performing different characterization.

3. Results and Discussion

3.1. XRD characterization

The X-ray diffraction pattern of the prepared borate glasses were recorded in the range of 20°- 80°. The results showed that the XRD pattern of the sample exhibited broad diffusion at lower scattering angles. It indicates the presence of long range structural disorder, which is characteristic of amorphous nature as shown in Figure 1. A broad diffuse scattering at different angles instead of crystalline peaks are exhibited confirming a long range structural disorder which is the characteristic of amorphous network.

3.2. Optical Properties

The photoluminescence studies given many information like the spontaneous emission of light from a material under excitation and the features of emission spectrum used to identify impurity level, emission
Figure 1: X-ray diffraction pattern of the Pr$_2$O$_5$ (x=10 % and x 1 = 0.5 %) doped borate glasses.

wavelength etc. Figure 2 shows the excited spectrum of praseodymium doped borate glass. Weak features were displayed between 200 and 320 nm and single, strong band centered at 361 nm which indicates the lowest 4f-5d transition of Pr$^{3+}$. In accordance with previous studies of Pr$^{3+}$, it happened due to the charge transfer of Pr$^{3+}$-O$_2^-$ transition at shorter wavelengths$^{18}$.

Figure 2: The excited spectrum of Pr$_2$O$_5$ (x=10 % and x 1 = 0.5 %) doped borate glass

The excited Pr$^{3+}$ ions decay from $^3P$ to ground state by emitting radiations in visible or UV spectrum. Emission peaks were observed at 375 nm, 413 nm, 436 nm and 496 nm for an excitation of 260 nm shown in figure 3. The observed reading agrees well with previous studies$^{19}$. These lines correspond to transitions like $^3P_2 - ^1H_2$, $^3P_0 - ^1H_4$. It is found that, at a peak of wavelength 496 nm, the fluorescence intensity gradually decreases. The transition $^3P_0 - ^2H_4$ can execute Laser action with blue colour. Hence the prepared glass can be used as a blue up converter in Fibre Optic communication field.

Figure 3: Emission spectrum of Pr$_2$O$_5$ (x=10 % and x 1 = 0.5 %) doped borate glass.

3.3. Spectral Analysis

3.3.1 FTIR analysis

Infra-red spectrum of borate glass doped with Pr$_2$O$_5$ in the range of 500 - 4500 cm$^{-1}$ is shown in Figure 4. In this region bands are connected with vibrations of borate network. Borate ring deformation (662 cm$^{-1}$), BO$_3$ bending (791 cm$^{-1}$), stretching vibration of tetrahedral BO$_4$ group (950-1200 cm$^{-1}$) are included along with two absorption bands due to stretching of trigonal and tetrahedral BO$_4$ (1200-1400 cm$^{-1}$) and the bending modes of OH groups (1700 cm$^{-1}$). In this spectrum, bending vibration occurs at lower frequencies than stretching vibration. The other absorption bands which are observed indicate weak transitions. The second broad band shows four lines appearing at 955, 1180, 1376 and 1159 cm$^{-1}$, due to the P-O symmetric ($v_s$) and asymmetric ($v_{as}$) stretching vibrations. The intensity of these lines increases with Pr$_2$O$_5$ content.

Figure 4: IR spectrum of Pr$_2$O$_5$ (x=10 % and x 1 = 0.5 %) doped borate glass
3.3.2 FT-Raman analysis

Raman spectrum of the prepared glass with broad peak is observed in the range of 500–2000 cm\(^{-1}\) as shown in Figure 5. The position of peaks in the spectrum defined the molecular structure of the sample. The symmetric stretching peak at 687 cm\(^{-1}\) looks more intense compared to others. Generally asymmetric stretching vibrations cause less intensity peaks only. The disorder in the glass matrix results in the broadening of the peaks.

![Figure 5: Raman spectrum of Pr\(_2\)O\(_5\) (x=10 % and x = 0.5 %) doped borate glass](image)

3.3.3. Thermal analysis

Thermogravimetric analysis is a technique to assess the stability of various substances. Figure 6 shows the simultaneously recorded thermogravimetric analysis (TGA) and differential thermogravimetric analysis (DTA) curve for Pr\(_2\)O\(_5\) doped borate glass. The amorphous nature of the glasses was confirmed using the TGA and DTA curves. The endothermic peaks corresponding to the glass transition (\(T_g\)) and exothermic peaks due to the crystallization (\(T_p\)) are observed. It allows concluding that melt - quenched samples are glasses. No appreciable weight loss was detected in the TGA measurements in the glass sample. The weight loss of the first step corresponds to water released in the sample and other steps correspond to the decomposition of the samples which is shown in Figure 6. The lack of sharp endothermic and exothermic peaks evidently specifies the formation of homogeneous glass. Figure 7 shows the differential thermal analysis of the glasses under investigation. The glass exhibits change due to the glass transition temperature \(T_g\) at 218 °C. It is also observed that, during higher temperature an exothermic peak \(T_c\) formed at 677 °C due to the crystal growth, which is followed by a single sharp endothermic effect, due to the melting of the glass, as symbolized by \(T_m\) at 1200 °C.

![Figure 6: TGA of Pr\(^{3+}\) (x=10 % and x = 0.5 %) doped borate glass.](image)

![Figure 7: DSC of Pr\(^{3+}\)(x=10 % and x = 0.5 %) doped borate glass.](image)

3.4. Dielectric Studies

The dielectric properties of the Pr\(_2\)O\(_5\) doped borate glass was measured using HIOKI 3532 - 50 LCR meter in the frequency ranging from 50 Hz to 5 MHz. The frequency dependence of dielectric constant at different temperatures is shown in Figure 8 (a). The dielectric constant decreases very rapidly at low frequencies and then slowly, as the frequency increases and finally, it becomes almost a constant at higher frequencies. The high value of dielectric constant at low frequencies may be associated with the establishment of polarizations namely: space charge, orientational, electronic and ionic polarization. Its low value at higher frequencies is attributed due to the loss of significance of these polarizations gradually. At high frequencies, normally orientation and space charge polarization exists. This behavior can be attributed to the applied electric field, which assists electron hopping between two different sites in glasses. The jump frequency of the charge carrier becomes large and comparable with the frequency of the applied field at high temperatures. At low frequency the charge carriers hop easily out of the sites with low free energy and tend to accumulate at sites with high free energy barriers. Hence a net polarization is developed and gives an increase in the dielectric constant and dielectric loss. However at high frequency, the charge carriers will not be able to rotate sufficiently, so their oscillation will begin to lay behind field resulting in a decrease of dielectric constant and dielectric loss. Jump frequency of the charge carries
becomes smaller than the frequency of the applied field at low temperatures. The periodic reversal of the applied field occurs so rapidly that excess charge carriers jumping in the field direction is nil. The polarization due to charged piling up at higher free energy barrier sites disappears, which causes a decrease in the values of the dielectric constant and the dielectric loss\textsuperscript{20-22}. At low frequencies all the mechanism of polarizations are active and with increasing frequency the contribution from different polarizations start decreasing. As the frequency increases the dipoles do not comply with the varying external field therefore decreasing the value of dielectric constant at low frequency region. From Figure 8 (b), it is observed that the dielectric loss decreases with increase in frequency at different temperatures. The low value of dielectric loss suggests that the glasses are of moderately good quality. Thus the low value of dielectric loss at higher frequencies is important for extending the applications towards photonic, electro-optic devices.

The frequency dependence of the ac conductivity for various temperatures is shown in Figure 9. It is observed that the conductivity increases with increasing frequency. At a given temperature the conductivity is directionally proportional to frequency confirming polaron hopping. From the figure we can observe that conductivity increases as the temperature increases\textsuperscript{23}. This increase in conductivity is attributed to the reduction in the space charge polarization at higher frequencies. The poly-dispersive behaviour exhibited is proved by the frequency dependence of the ac conductivity. Further as the frequency decreases more and more charge accumulation occurs at the interface between electrode and electrolyte, and hence there is a drop in conductivity at low frequencies.

4. Conclusion

Pr\textsubscript{2}O\textsubscript{5} doped borate glasses were prepared by the conventional melt quenching method. The XRD spectrum revealed that the Pr\textsubscript{2}O\textsubscript{5} doped borate glasses were amorphous in nature. Photoluminescence properties of Pr\textsubscript{2}O\textsubscript{5} doped borate glass were analyzed which proves that the prepared glass can be used as a blue up converter in Fibre Optic communication field. FTIR and FT Raman studies were carried out of Pr\textsubscript{2}O\textsubscript{5} doped borate glass. The disorder in the glass matrix results in the broadening of the obtained peaks. Thermal analyses of the glasses have been done to see the structure of the glasses. The variations of the dielectric constant, the dielectric loss and AC conductivity with frequency and temperature for Pr\textsubscript{2}Odoped borate glass were analyzed. The dielectric studies revealed that both the dielectric constant and the dielectric loss decreased with an increase in the frequency at different temperatures. The low value of dielectric loss suggests that the glasses are of moderately good quality. Also the low value of dielectric loss at higher frequencies is important for extending their applications towards photonic, electro-optic devices.

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6. References


