Thin Copolymer-Cased Light-Emitting Display Made with Fluorine-Foped Tin Oxide Substrates

Rudolf Lessmann, Ivo Alexandre Hümmelgen*

Group of Organic Optoeletronic Devices, Departamento de Física, Universidade Federal do Paraná, C.P. 19044, 81531-990 Curitiba - PR, Brazil

Received: December 15, 2003; Revised: May 30, 2004

Keywords: electroluminescence, copolymer, fluorine-doped tin oxide

1. Introduction

Since the development of the first organic light-emitting diode (OLED) with low drive voltage and high electroluminescence, organic electroluminescent devices presented a great evolution in structure. In spite of this evolution ITO (indium-tin oxide) continues to be largely used, even with the known constraint of indium diffusion into the organic layers. It was observed that the use of TO (tin oxide) instead of ITO, as transparent electrode, reduces the photoluminescence suppression of PPV converted from a polyelectrolyte precursor, suggesting higher chemical stability of the TO electrode. OLEDs were demonstrated using TO as transparent electrode, but the lower work function and the higher electrical resistivity of TO, when compared to ITO, inhibited a wider use of TO in OLEDs. Fluorine-doped tin oxide (FTO) has transparency comparable to that of ITO and can now be produced with sheet resistance similar to that of ITO, being much more stable from the chemical point of view. Intermediate layers like PEDOT:PSS, poly(3,4-ethylenedioxythiophene):poly(styrenesulfonate), or sulfonated polianiline can be used to effectively reduce the potential barrier for positive charge carrier injection, so that the low work function of FTO does not more constitute a serious problem limiting FTO use in organic device applications.

In this work we use a complex, but well-established concept of device structure, using FTO as anode, instead of ITO. As the hole injecting layer we used PEDOT:PSS (Baytron P®) (poly(3,4-ethylenedioxythiophene):poly(styrenesulfonate)), and OPPVDBC (poly(1,10-decanedioxy-2-methoxy-1,2-phenylene-1,2-ethylenylene-3,6-dimethoxy-1,4-phenylene-1,2-ethylenylene-3-methoxy-1,4-phenylene)) as hole transport layer (HTL, see Fig. 1). In the organic heterolayer used in this work the recombination zone is expected to be located at the interface between OPPVDBC and Alq3, which was used as electron transporting layer (ETL), due to the energetic discontinuity of the relevant energy levels of the layers (HOMO: highest occupied molecular orbital; and LUMO: lowest unoccupied molecular orbital). Alq3, is still an interesting material due to its high electroluminescent efficiency.

2. Experimental

Commercial glass substrates coated with FTO (sheet re-
sistance of 15 Ω/□) supplied by Flexitec Eletrônica Orgânica Ltda were cut in pieces of 23.0 × 32.5 mm², patterned using conventional lithography and etched14. After patterning, FTO substrates were cleaned in a 1:4 solution of H₂O₂:H₂SO₄ for 1 h and rinsed in ultra-pure water and isopropanol. No change in surface morphology or transmittance spectra of the FTO was noted due to these cleaning, patterning and etching procedures. Absorbance spectra were measured using an Optometrics spectrophotometer. Film thickness was determined using a Dektak3 surface profiler.

PEDOT-PSS dispersed in H₂O was treated in ultrasonic bath for 10 h to dissolve undesired clusters before filtering. PEDOT-PSS was deposited onto glass/FTO by spin-coating at 4.200 rpm. After deposition, the sample was dried in a vacuum oven at 313 K for 4 h. OPPVDBC was dissolved in distilled CHCl₃ and deposited onto glass/FTO/PEDOT-PSS by spin-coating at 200 rpm, resulting in an OPPVDBC film thickness of ~ 45 nm. Alq₃, Ca and Al, were evaporated in high vacuum, at 71 × 10⁻⁶ Pa, with thickness of 45 nm, ~ 200 nm and ~ 200 nm, respectively. The fully evaporation sequence was done without vacuum breaking. Alq₃ and metal film geometry were controlled through shadow masks geometry during evaporation.

Devices without the Alq₃ layer were also made for comparison.

After evaporation, the samples were transported in air to a glove box with dry N₂ atmosphere. After 48 h under N₂ flux the samples were encapsulated15,16. Electrical and optical measurements were made in the dark. Emission spectra of FTO/PEDOT:PSS (50 nm)/OPPVDBC (20 nm)/Alq₃(45 nm)/Ca/Al device and of FTO/PEDOT:PSS (50 nm)/OPPVDBC (20 nm)/Al/Ca device were recorded with an OceanOptics CCD spectrophotometer with a 50 µm aperture optical fiber and with an Hitachi spectrofluorimeter, respectively.

The Ionization Potential (IP) was estimated using cyclic voltammetry, correcting the oxidation onset at Ag/Ag⁺ quasi-reference electrode to vacuum reference level17 and the LUMO was estimated summing the energy of the optical gap to the HOMO-energy18. The voltammetry was performed in a single compartment electrochemical cell. Gold was used as working electrode and a 0.1 mol/l LiClO₄ solution in acetonitrile, as electrolyte.

3. Results and Discussion

The PEDOT layer is introduced because it is able to effectively eliminate the potential barrier between PEDOT and the substrate10, in our case, FTO. It strongly collaborates to charge injection in this device, so that positive charge carrier injection in the OPPVDBC layer is not inhibited by the energy barrier at the anode interface. Similarly, the introduction of the Alq₃ layer strongly improves the negative charge carrier injection, leading to higher current density through the device, when compared to a device without Alq₃ layer, as can be seen in Fig. 3. It was observed that current-voltage (I(V)) characteristics of both devices, FTO/PEDOT:PSS/OPPVDBC/Alq₃/Ca/Al and FTO/PEDOT:PSS/OPPVDBC/Ca/Al, show negligible rectification.

The device without Alq₃ presents lower electroluminescent intensity, leading to a higher noise-level electroluminescent signal (Fig. 4), but could be seen under amber light conditions. More important, however, is that the introduction of the Alq₃ layer implies in a significant reduction of the onset voltage for light emission, from ~ 11 V to ~ 6 V. Comparing the spectra of both devices, presented in Fig. 5, we observe a red shift of about 30 nm in the device with Alq₃ layer, indicating that the recombination zone in this device is preponderantly located inside the Alq₃ layer22. It is important to observe that the...
optical gaps of OPPVDBC and of Alq₃ are almost the same, permitting, in principle, exciton diffusion across the OPPVDBC/Alq₃ interface. The calculated CIE coordinates are \( x = 0.29 = 0.58 \) for the FTO/PEDOT:PSS (50 nm)/OPPVDBC (45 nm)/Alq₃ (45 nm)/Ca/Al device and \( x = 0.22, y = 0.37 \) for the FTO/PEDOT:PSS (50 nm)/OPPVDBC (20 nm)/Ca/Al device.

In the device with Alq₃, the thickness of the HTL (OPPVDBC) and ETL (Alq₃) is the same. It is interesting to notice that reported mobilities for negative charge carriers in Alq₃ (\( \mu_{\text{Alq₃}} \approx 10^{-6} \text{ cm}^2/\text{V.s} \)) and for positive charge carriers in OPPVDBC (\( \mu_{\text{OPPVDBC}} \approx 10^{-9} \text{ cm}^2/\text{V.s} \)) are quite different. In this case one would expect a displacement of the recombination zone into the OPPVDBC layer, since the electrode interfaces are assumed to present no constraints for charge injection, permitting to assume ohmic injection. Under these conditions it results that radiative recombination is much more favored in Alq₃ than in OPPVDBC or that there is a net exciton flow at the OPPVDBC/Alq₃ to the Alq₃ layer due to a difference in exciton binding energy, which would be higher in the Alq₃.

We have succeeded in turning on all display segments
of a seven-segment display. Segments till 16 mm² were made showing good uniformity (Fig. 6). We made FTO rails wide enough to avoid enhanced resistivity problems. The display was connected to an electronic circuit to count numbers and worked well with the LEDs operating at ~ 8 V, demonstrating that FTO is suitable for use in LEDs with large active areas and consequently, long conducting paths.

Another point that deserves discussion concerns the work function of FTO and its consequences to device operation. In the device, the PEDOT:PSS layer was used to change the effective energy barrier for charge injection, so that it almost effectively equals the difference d of the chemical potential of the PEDOT:PSS and the anode work function. Considering that thermodynamical equilibrium is established equalizing the electrochemical potential of the electrodes, an additional contribution to the built-in potential appears as a consequence of the difference ∆Φ in electrodes work functions. These two contributions result that the built-in potential equals ∆Φ + δ, which is the energy difference between the work function of the cathode and the PEDOT:PSS electrochemical potential, i.e., independent of the anode work function. This means that in principle, in case of devices with intermediate layer like PEDOT:PSS, the anode work function would not be relevant for device operational characteristics. These considerations presume that the PEDOT layer has the capability to compensate the potential difference δ. Since this compensation is done by polarization and charge transfer, it requests a minimum thickness of the PEDOT:PSS layer, otherwise the integration of the electric field throughout the sample thickness can be lower than δ.

Another important aspect that must be stressed is that the substitution of the ITO anode by FTO eliminates the problem of indium diffusion into the electroluminescent polymer layer due to the absence of indium in the anode material composition.

Previous investigations on FTO used in organic LEDs showed that devices prepared onto FTO substrates presented lower turn-on-voltage for current and for light and more light for a given voltage, with one observed drawback, higher leakage currents. In the referred study, however, no intermediate layer was used to improve charge injection from the FTO and device performance was dependent on FTO cleaning procedure, despite the fact that the work function of FTO is not sensible to specific adopted cleaning procedure, like in the case of TO6.

4. Conclusions

We have demonstrated that large seven-segment displays can be constructed using low sheet resistance fluorine-doped tin-oxide as anode material. To demonstrate it we constructed FTO/PEDOT:PSS/OPPVDBC/Alq₃/Ca/Al devices in a geometry useful for low-end electronics.

Acknowledgements

The authors thanks to CAPES and CNPq for financial support, Winston S. L. Fung for the motivation and Flexitec Eletrônica Orgânica Ltda for providing the FTO substrates. We want to thank Dr. D. Ma who provided OPPVDBC material.

References