

Nanostructured Powders of AA7075 - SiC Manufactured by High-Energy Ball Milling in a Bath of Isopropyl Alcohol

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In this study, aluminum alloy 7075 (AA7075) nanopowders were prepared by High-Energy Ball Milling (HEBM) in a bath of isopropyl alcohol. The process was investigated in different milling times and silicon carbides (SiC) reinforcement percentual. The effects of these parameters on the samples were characterized by X-ray diffraction (XRD), Laser Diffraction (LD), Scanning Electron Microscopy (SEM), and Energy Dispersive Spectroscopy (EDS). The XRD analyses showed that as the grinding time increases, the micro deformation also increases, while the crystallite and particle size decrease until a constant value at 480 min. If the percentual of SiC reinforcement increases until 5 percent, there is a minimum change in the results compared to AA 7075 milling 480 min with no reinforcement. On the other hand, when the AA7075 was milled for 480 min and reinforced by 10 percent SiC, the best structural refinement result was achieved.

Keywords: Nanostructured Al Powder, High-Energy Ball Milling, Powder Metallurgy.

1. Introduction

The growing demand for aluminum in aerospace sectors has spurred research into enhancing the reinforcement of aluminum metal matrices across technological uses. The aim is to substitute traditional materials and their alloys with improved alternatives. Aluminum metal matrices reinforced with other materials exhibit high mechanical and tribological properties and an attractive strength-to-weight ratio¹⁻⁷. In this context, aluminum alloy 7075 (AA7075) as a matrix is notable for its low density and good workability, offering a favorable combination of strength, ductility, and toughness. On the other hand, introducing Silicon Carbide (SiC) as reinforcement, whether in micron or nanoscale form, enhances wear resistance and hardness due to its low density and high strength⁸⁻¹⁵.

Furthermore, attempts were made to explore severe plastic deformation processes to obtain nanostructured materials¹⁶⁻¹⁹. Among these mechanical methods, High-Energy Ball Milling (HEBM) is a simple and efficient technique to prepare alloys at room temperature with nanocrystalline grains (measuring below 100nm)²⁰⁻²³.

HEBM is characterized by the repeated flattened, cold-welded, fractured, and rewelded welding of powder particles under different conditions until the rate of fracturing is achieved. It introduces shear bands that contain a high-density

network of dislocations and other crystallite defects that reduce crystallite size and particle size and promote changes in morphology up to reach the equilibrium state²⁴⁻²⁶. In this phase, the segregation effects decrease, and a homogeneous distribution of reinforcement into the particles can be obtained²⁷.

The responses obtained by HEBM depend on the type of mill and the process variables, being the most typically studied: milling time, rotation speed, ball-to-powder ratio, and reinforcement percentual, among others²⁸⁻³¹. However, when the HEBM occurs in a liquid medium, the environment can influence the results^{32,33}. It has been reported that wet grinding is a more suitable method than dry grinding to obtain finer-ground products because the solvent molecules are adsorbed on the newly formed surfaces of the particles and lower their surface energy. The less-agglomerated condition of the powder particles in the wet medium is also a helpful factor^{34,35}. Besides that, as a disadvantage, the rate of amorphization and the increase of the contamination of the process is faster during wet grinding than during dry grinding³⁶⁻³⁸. The critical issue arises from the contribution of the influence of the liquid on the resulting grain and particle/morphology size.

The main aim of the research presented here was to investigate the effect of changing milling time and reinforced percentual of SiC applied to AA7075 manufactured by high-energy ball milling in a liquid media of isopropyl alcohol to obtain nanocrystalline powders.

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2. Experimental Procedure

Commercial gas-atomized nanocrystalline powder AA7075 from Aluminum Powder Corporate and nanometric powder of SiC, with a particle size D (50) 50nm, supplied by Iolitec GMBH, were used as matrices and reinforcement, respectively. In Table 1, the chemical composition of AA7075 is shown³⁹.

The materials, matrix, and reinforcement were deposited into a stainless-steel attrition ball mill equipped with a K-type thermocouple and a temperature controller. Milling was carried out at 900 rpm with balls 100C6 (1%C, 1,5%Cr) of 6,4mm and balls to powder mass ratio of 20:1. For each sample, 50g of material was manufactured at a bath of 100ml isopropyl alcohol (C₃H₇OH - 99,82%) and 1wt% of zinc stearate (C₃₆H₇₀O₄Zn) was used as process control agent (PCA). The process temperature of 25°C was maintained via a jacket refrigerated with water around the attritor mill⁴⁰. The following milling conditions were studied:

- Milling time: 60, 120, 240, and 480 minutes.
- Reinforce weight percentual (SiC): 1wt%, 2wt%, 5wt%, and 10wt%.

After gridding, the samples were dried at 100°C to evaporate the residues. The effects of crystallite size, micro deformation, particle size, morphology, and chemical composition were analyzed.

The crystallite size and micro deformation were investigated by X-ray diffraction, XRD (Rigaku Ultima III) in the range of 5-120° with a step rate of 0.02°/s at 40kV and 30mA. The Match Phase Analyses software was used to identify the phases and to index the diffraction peaks.

Utilizing the linear regression analysis of the Williamson – Hall plot equation (Equation 1)^{41,42}, the determination of crystallite size and the assessment of micro deformation to the Full Width at Half Maximum (FWHM) of the peak were conducted for the four principal aluminum peaks,

with a confidence level exceeding 92 percent. Comparatively, the instrument effect on crystallite size was not considered.

$$FWHM = k\lambda / L\cos\theta + 4\varepsilon \tan\theta \quad (1)$$

where “FWHM” is the full width at half maximum in radians; “k” is a constant (0.94); “λ” is the wavelength of the x-rays (15.4nm); “L” is the average crystallite size; “θ” is the Bragg angle, and “ε” is the micro deformation measured.

The particle size was determined by Laser Diffraction (Malvern Mastersizer 2000), where the sample was suspended in water and agitated by ultrasound to size range 0.02μm to 2000μm⁴³.

The laser beam incident by an ensemble of particles dispersed in either a liquid or an air stream promotes light scattering, and the particle size is calculated as spheres of equal volume. The scattering or diffraction angles exhibit distinct traits related to particle size, as they progressively decrease with an increase in particle size. The equations for the average and uncertain of particle sizes assume the size distribution is available as a histogram. The measure D (0.5) represents the median particle diameter corresponding to the 50th percentile of the cumulative undersize distribution⁴⁴⁻⁴⁶.

Lastly, the morphology and composition of the particle were analyzed by Scanning Electron Microscopy (Hitachi TM 3000) operating at 20kV, equipped with an EDX probe.

3. Results and Discussion

The results of eight samples compare the AA7075 as-received, AA7075 as a function of milling time, and AA7075 as a function of SiC reinforce percentual.

Figure 1 presents the morphology and alloy elements detected by EDS microanalysis of AA7075 starting powder. Table 2 shows all values found to the crystallite size, micro deformation, and particle size.

Table 1. Chemical Composition AA 7075.

Material	Al	Cr	Cu	Fe (max)	Mg	Mn (max)	Si (max)	Ti (max)	Zn
Weight %	87.1-91.4	0.18- 0.28	1.2-2.0	0.5	2.1-2.9	0.3	0.4	0.2	5.1-6.1

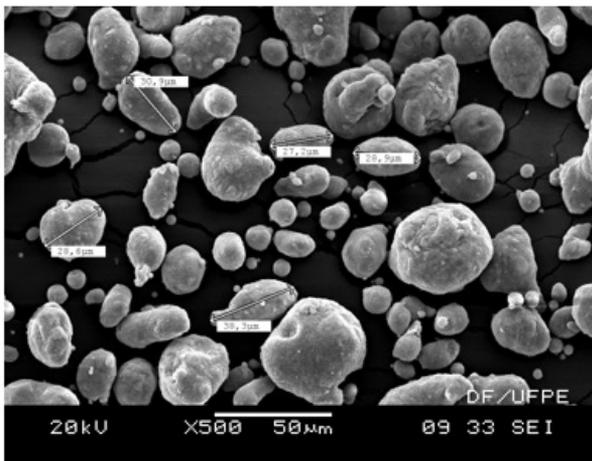
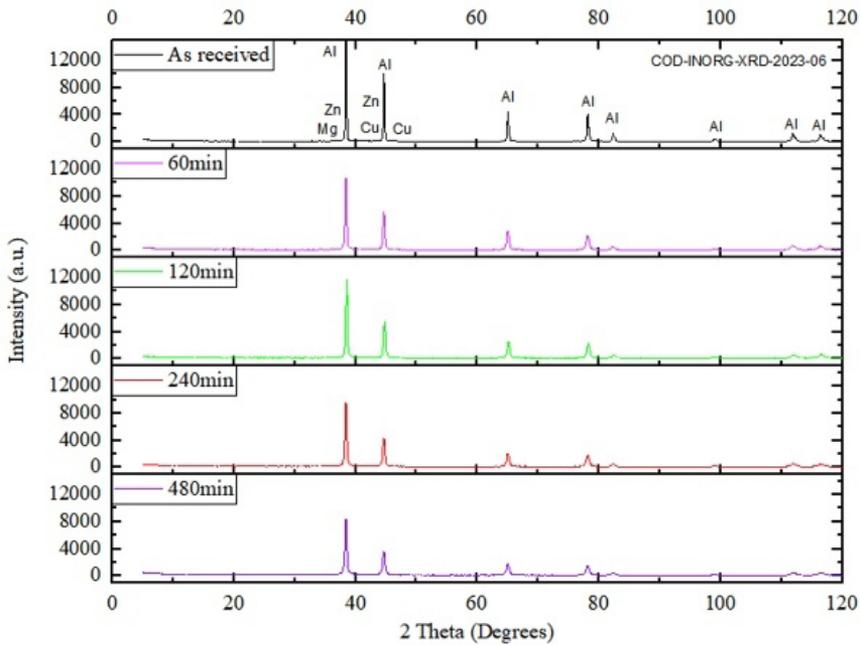


Figure 1. SEM AA7075 as received (x500) and EDS microanalysis.

EDS AA7075 As Received		
Element	Weight (%)	Error
Aluminum	88.70	±3.5
Oxygen	3.73	±0.7
Magnesium	2.20	±0.1
Zinc	3.85	±0.1
Copper	1.52	±0.1

Table 2. Values of crystallite size, micro deformation, and particle size for all samples AA7075.

Sample Number	Crystallite Size [nm]	Micro Deformation [%]	Particle Size D (0.5) [μm]	Milling Time [min]	Reinforce (SiC) [wt%]
AA075	49	0.02	31.71	As Received	
1	33	0.06	37.62	60	0
2	32	0.06	25.44	120	0
3	30	0.10	16.28	240	0
4	29	0.12	10.64	480	0
5	29	0.12	11.18	480	1
6	32	0.10	13.66	480	2
7	29	0.12	10.80	480	5
8	25	0.16	9.04	480	10

**Figure 2.** Standard of X-rays diffraction AA7075 as a function of the milling time.

The AA7075 as-received powder was round because of its production process⁴⁴⁻⁴⁶. Some particle sizes presented values around 30 μm , according to particle size measured by Laser Diffraction D (0.5) 31.71 μm . Besides that, the crystallite sizes and micro deformation measured 49nm and 0,02 percent, respectively.

3.1. AA7075 as a function of milling time

Figures 2-5 present, at the sequence, the graphs of X-rays diffraction, particle size distribution, crystallite size, micro deformation, and particle size, particle morphologies, and AA7075 milled 480 min EDS as a function of the milling time.

The values obtained to crystallite size standard agree with the particle size distribution and micro deformation.

Table 2 and Figure 4 present the internal structures refinement as a function logarithmic of milling time. The crystallite size decreased from 49nm to about 30nm while the micro deformation increased from 0.02 percent to around 0.12 percent. It is evidenced by diffraction peaks that became wider, smaller, and not shifted. The X-rays are reflected in a diffraction peak when a crystalline material

is struck. However, the effects that lead to an increase in its width and a subsequent decrease in intensity are attributed to the reduction of crystallite size and increased micro deformation⁴⁷⁻⁴⁹.

Furthermore, Table 2 and Figure 3 show particle size decreased from about D(50) 30 μm to the value constant D(50) 10 μm in 480 min. Until 240 minutes of milling, there is an indication of competition between cold welding and fracturing because the particles are still flattened, and at 480 min, the fracture domain is observed⁵⁰⁻⁵² (Figure 5). Various mechanical factors and parameters, including wet milling and energetic conditions, have influenced this response. Wet milling contributed to the formation of good-proportion particles. The high-energy collision among the milling balls played a crucial role in uniformly dispersing the stress in the matrix⁵³⁻⁵⁵.

Finally, AA 7075 milled 480 min EDS (Figure 5) reveals no contamination traces from the steel balls or the stainless steel attritor mill were detected. The presence of liquid alcohol in the milling process increased the oxygen percentage in the sample.

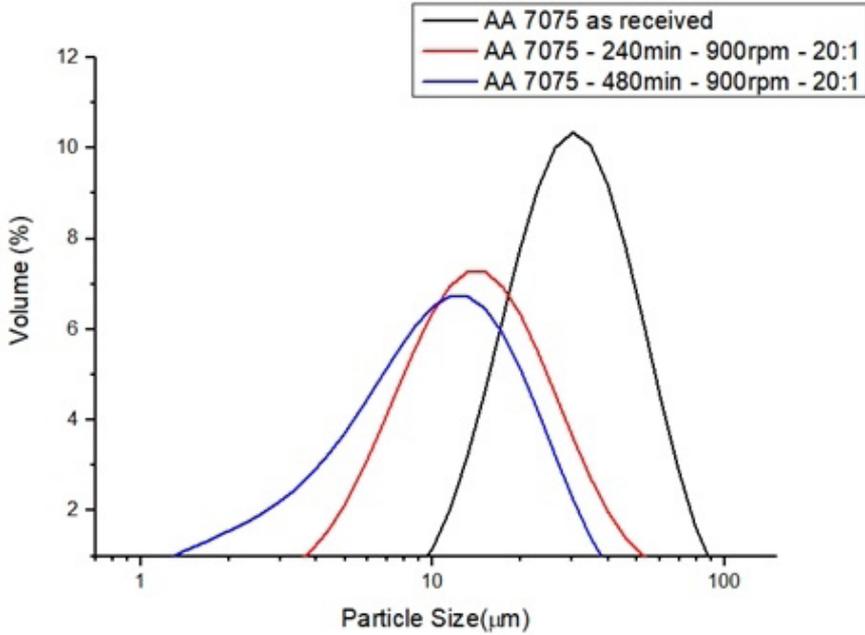


Figure 3. Particle size distribution AA 7075 as a function of the milling time.

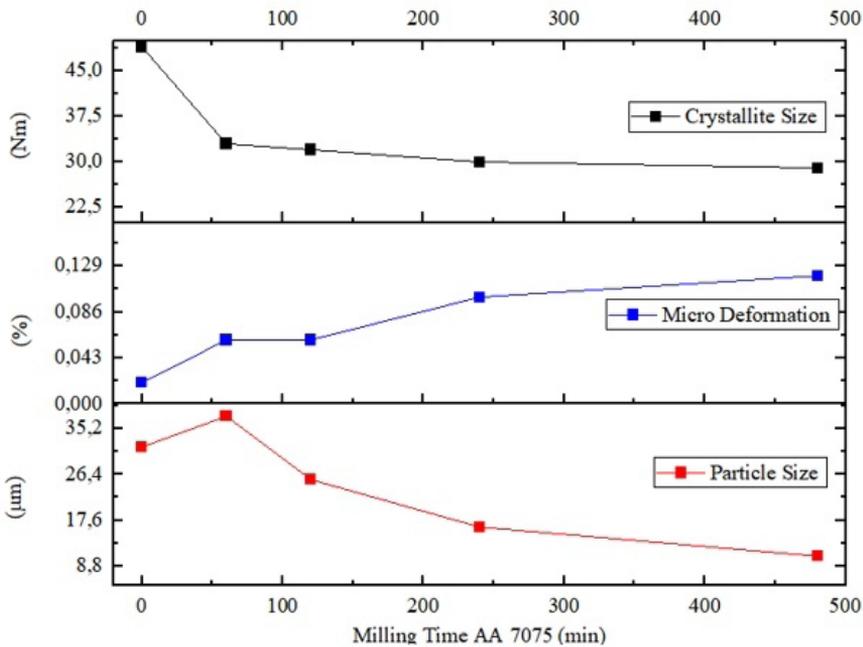


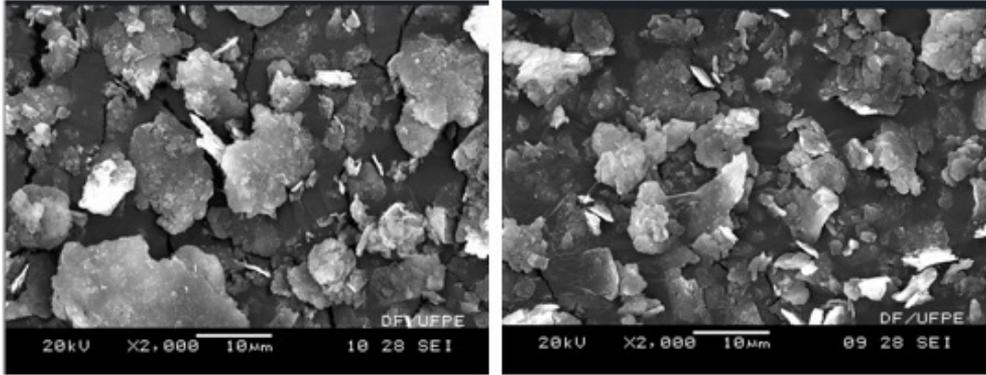
Figure 4. Crystallite size, micro deformation, and particle size of AA 7075 as a function of the milling time.

3.2. AA7075 as a function of SiC reinforce percentual.

In its turn, Figures 6-9 show the standard of X-rays diffraction, particle size distribution, crystallite size, micro deformation, particle size, particle morphologies, and AA7075 milled 480 min + 10% SiC EDS as a function of SiC reinforcement percentual.

The crystallite size, micro deformation, particle size, morphologies, and chemical composition present similar to

values found in AA7075 milled 480 min, independently if no reinforcement was used or if the SiC reinforces percentual was used until 5 percent due to the AA7075 can support high-stress structurals⁵⁶. Significant changes were observed when 10 percent of SiC reinforcement was applied; in this condition, the best refinement structural results were achieved: 25nm of crystallite size, 0.16 percent of micro deformation, and 9 μ m of particle size.



a)

b)

EDS AA 7075 Milled 480 min		
Element	Weight (%)	Error (%)
Aluminum	70.78	±3.8
Oxygen	21.76	±0.9
Magnesium	2.11	±0.1
Zinc	3.86	±0.1
Copper	1.49	±0.1

Figure 5. SEM AA7075 as a function of the milling time (2000x): a) 240min e b) 480min and EDS AA7075 milled 480 min.

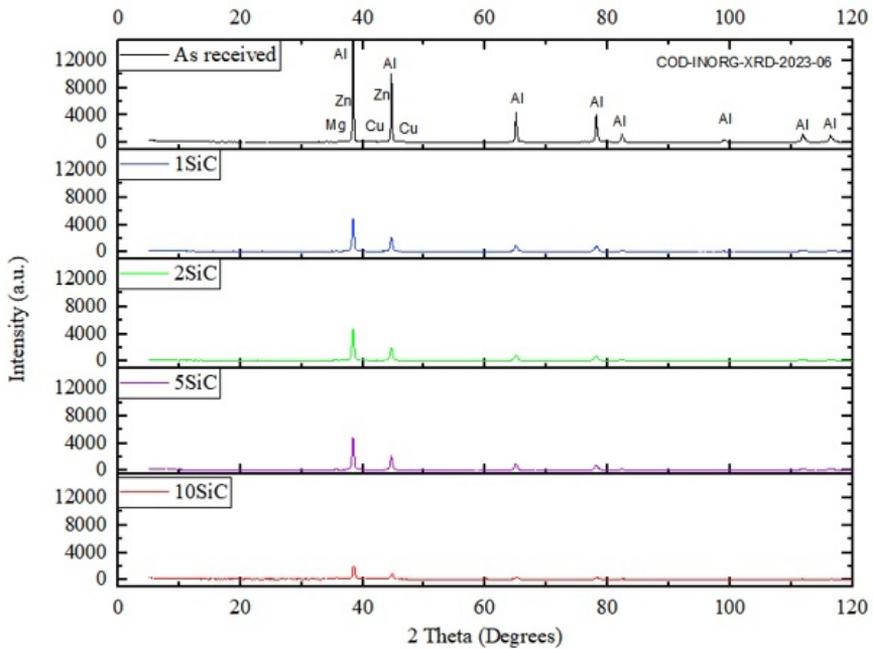


Figure 6. Standard of X-rays diffraction as a function of SiC reinforce percentual.

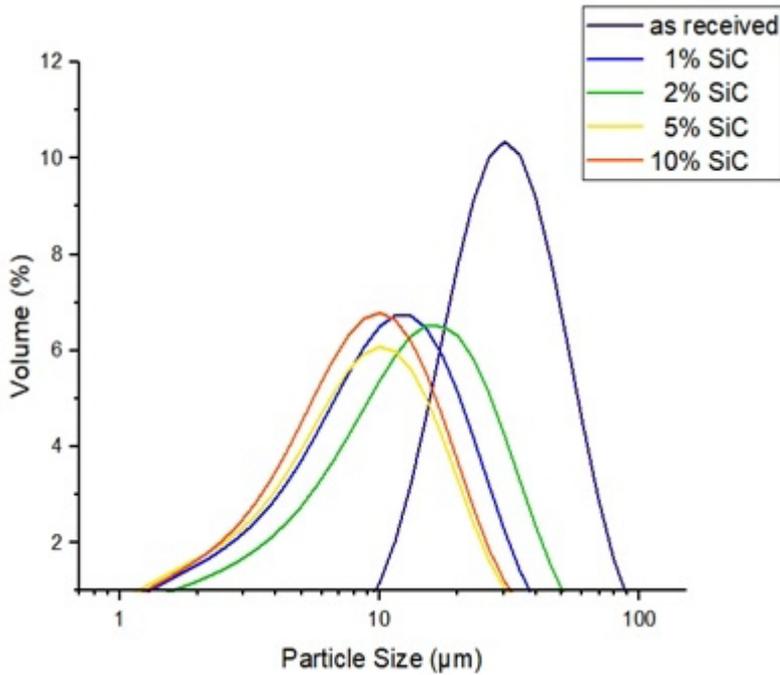


Figure 7. Particle size distribution AA 7075 as a function of SiC reinforce percentual.

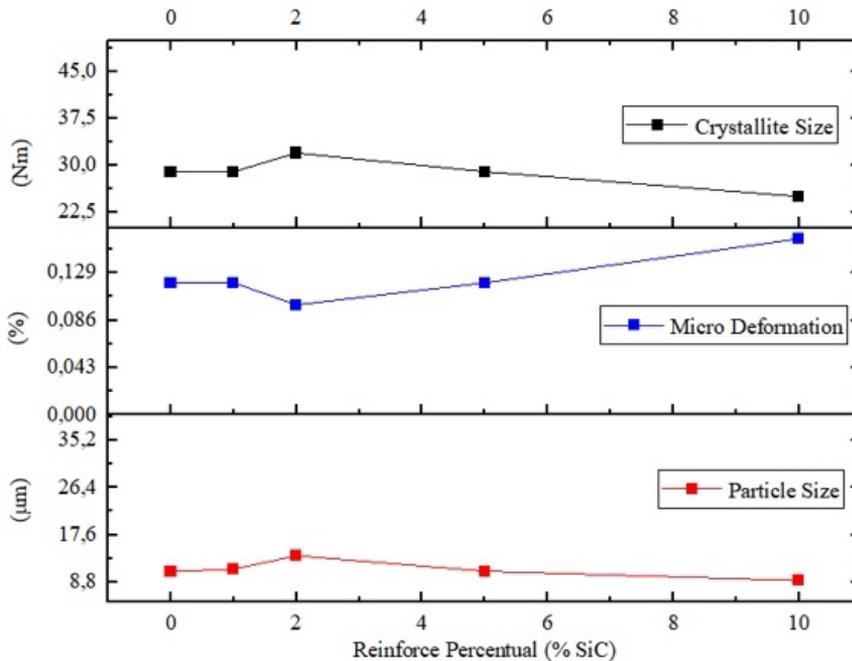
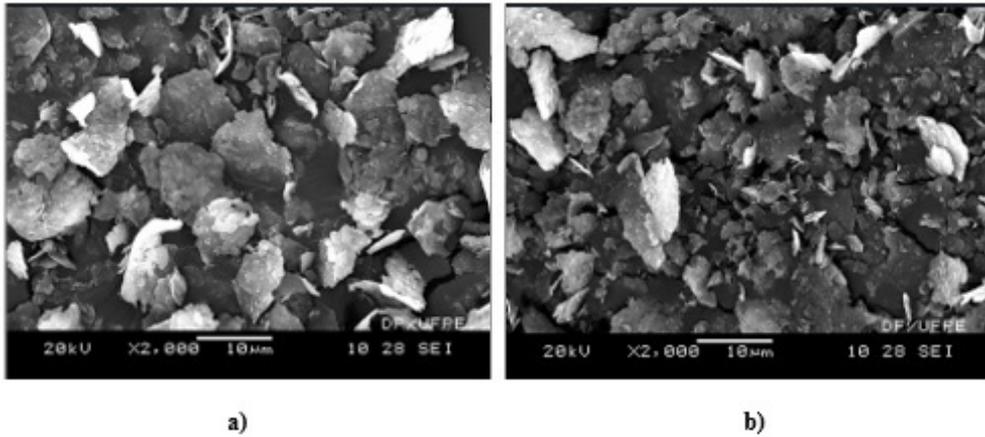


Figure 8. Crystallite size, micro deformation, and particle size of AA 7075 as a function of SiC reinforce percentual.

4. Conclusions

Nanosize powders of AA7075 were manufactured by high-energy ball milling, and the milling time and SiC reinforce percentual were studied. The following results can be resumed according to below:

- 1- For AA7075 as a function of milling time, the crystallite size and particle size decrease from 49nm up to about 30nm and from D (50) 30 μ m to D (50) 10 μ m, respectively, while the micro deformation increases from 0.02 percent to around 0.12 percent to a constant value of 480 minutes.



EDS AA7075 Milled 480 min + 10%SiC		
Element	Weight (%)	Error
Aluminum	71.59	±2.6
Oxygen	20.57	±0.8
Magnesium	2.04	±0.1
Zinc	2.51	±0.1
Copper	1.20	±0.1
Silicon	2.09	±0.1

Figure 9. SEM AA7075 as a function of reinforce percentual (2000x): a)5% SiC e b)10% SiC and EDS AA7075 Milled 480 min + 10% SiC.

- 2- The presence of liquid alcohol in the milling process increased the oxygen percentage in the sample.
- 3- For the AA7075 as a function of SiC reinforcement percentual, the crystallite size, micro deformation, particle size, morphologies, and chemical composition presented like AA7075 milled 480 min, independently if no reinforcement was used or if the SiC reinforces percentual was used until 5 percent.
- 4- The best structural refinement result was achieved when 10 percent of SiC reinforcement was applied AA7075 milling 480 min (25nm to crystallite size, 0.16 percent to micro deformation, and 9µm to particle size).

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