Effects of storage temperature and time on false setting behavior of CPI-S Portland cement

(Efeitos da temperatura e do tempo de armazenamento no comportamento de falsa pega do cimento Portland CPI-S)

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Abstract

The false setting is when cement stiffens prematurely in a few minutes after adding water. Some variables could cause false setting in CPI-S-32 Portland cement, for example, alkali concentration in the cement, the formation of alite (C₃S) with low reactivity, and cement storage temperature and time in silos. Temperature increases cause calcium sulfate dihydrate to dehydrate, forming hemihydrate (CaSO₄.0.5H₂O) or anhydrite (CaSO₄), which causes the false setting. In this study, the influence of cement storage temperature (100, 105, 110, 120, and 130 °C) combined with the cement storage time (30, 60, and 120 min) in a silo was studied regarding the CPI-S-32 false setting behavior. It was verified that temperatures above 110 °C and storage time above 60 min are conditions that favor the false setting of CPI-S-32 cement. Physicochemical analysis, TG/DTG, XRF, and XRD were applied as complementary analyzes for the false setting assays of CPI-S-32.

Keywords: false setting, CPI-S-32 Portland cement, cement storage temperature, cement storage time.

Resumo

Falsa pega é a denominação dada ao enrijecimento prematuro anormal do cimento em poucos minutos após a adição de água. Alguns fatores podem ocasionar falsa pega no cimento Portland CPI-S-32, por exemplo, concentração de álcalis no cimento, formação de alita (C_3S) com baixa reatividade e temperatura e tempo de armazenamento do cimento nos silos. O aumento da temperatura causa a desidratação do sulfato de cálcio di-hidratado formando o hemi-hidrato ($CaSO_4$,0,5 H_2O) ou anidrita ($CaSO_4$), que causa a falsa pega. Assim, neste trabalho foi avaliada a influência da temperatura de armazenamento do clínquer (100, 105, 110, 120 e 130 °C) combinada com o tempo de armazenamento do cimento (30, 60 e 120 min) no comportamento de falsa pega do CPI-S-32. Foi verificado que temperatura de armazenamento acima de 110 °C e tempo de armazenamento acima de 60 min são condições que favorecem a falsa pega do cimento CPI-S-32. Análises físico-químicas, TG/DTG, FRX e DRX foram utilizadas como análises complementares aos ensaios de falsa pega do CPI-S-32.

Palavras-chave: falsa pega, cimento Portland CPI-S-32, temperatura de armazenamento, tempo de armazenamento.

INTRODUCTION

The population growth, as a rule, causes an increase in materials used for housing construction [1], and cement can be highlighted among these materials. Cement can be defined as a material with adhesive and cohesive properties, which make it able to join mineral fragments in the form of a compact unit [2]. Portland cement is a hydraulic binder obtained by grinding a Portland clinker with the addition of one or two forms of calcium sulfate during the production process. Portland cement CPI-S-32 is composed of 6% to

10% (m/m) of carbonate material containing at least 75% CaCO₃ [3] processed by milling [4-6]; during the cement milling and storage stages, operational parameters must be controlled according to NBR 16697 standard. Among these parameters, the cement storage temperature may cause an alteration in the physical-chemical behavior of cement, such as the anomalous behavior of a false setting. Gypsum is added to the clinker in this stage to delay the hydration reaction of tricalcium aluminate (C₃A), which is the clinker constituent that features the biggest reactivity with water.

The reactions in which the CPI-S becomes a binder material occur in the paste, constituted by the direct addition of some water molecules and cement. In other words, the presence of water and the silicates and aluminates of cement create hydrated products, which result in a resistant and steady mass over time. Among the several hydration products of CPI-S, the hydration reaction of silicates, calcium hydroxide, and aluminates can be highlighted. Both calcium silicates, named as tricalcium silicate or alite (C₂S) and dicalcium silicate or belite (C₂S), are the main components of cement, with the first experiencing hydration faster than the second [7] due to higher reaction speed compared to the C₂S, considering that C-S-H gel is the final complete hydration product of both C₂S and C₂S [7, 8]. Furthermore, both calcium silicates during their hydration form the hydrated calcium silicate (C-S-H), which performs a dominant role in determining the mechanical and physicalchemical properties of the concrete [9]. The C-S-H normally consists of 50% to 60% of the solid volume in a fully hydrated cement paste, therefore being the most important phase in determining the properties of the paste. Unlike C-S-H, calcium hydroxide has its stoichiometry defined as Ca(OH), and constitutes 20% to 25% of the solid volume in the hydrated cement paste. However, the C₃A reaction with water is immediate and creates crystalline hydrates of C₂AH₄ (tricalcium aluminate hydrate), C₄AH₁₀ (hexagonal calcium aluminate hydrate), and C2AH8 (dicalcium aluminate hydrate) with heat release, and occupy around 15% to 20% of the solid volume in a hydrated cement paste [10]. In the hydration process of CPI-S, both the hydration of C₂A and its hydration in the presence of gypsum are important, being responsible for slowing down the hydration reaction and the immediate setting of the cement. This effect can be explained by the fact that when gypsum and alkalis go into solution quickly, the C₃A solubility is reduced in the presence of hydroxyl, sulfate, and alkalis [8].

The setting refers to the cement paste going from a fluid state to a rigid state where the paste acquires a certain consistency which makes it improper for work, and in which the physicochemical process occurs by means of exothermal reactions. The initial setting is the start of plasticity loss in the water and cement mixture, followed by the paste temperature rising. The final setting is when the paste ceases to be deformed by a small load and starts to form a rigid block. In addition to the setting, another behavior that is highlighted in CPI-S cement is the false setting, which is when abnormal premature stiffening of the cement occurs a few minutes after adding water. False setting differs from the instantaneous setting since there is no important heat release [10]. When the cement C₃A has low reactivity, as is the case of partially hydrated cement or carbonated cement that may have been stored in inappropriate form, and a large amount of gypsum is present in the cement at the same time, the solution will contain a low concentration of aluminate ions, making it rapidly supersaturated with calcium and sulfate ions, leading to the rapid formation of large gypsum crystal with a corresponding loss in the consistency, which therefore causes the false setting of the cement [8].

Some factors influence the false setting in CPI-S such as cement alkalis, the activation of C₃S due to high moisture, the clinker milling temperature, and storage temperature and time of cement in silos. Partially calcined gypsum

(CaSO₄.2H₂O) loses 75% of its crystallization water, forming a hemihydrate or basanite (CaSO₄.0.5H₂O) or anhydrite (CaSO₄) which possesses five times more solubility than gypsum, and then forms a supersaturated solution when mixed with water which tends to deposit dihydrate crystals. Moreover, the cement hydrates in needle calcium sulfate form when mixed with water. Thus, a false setting can be named 'calcium sulfate setting', resulting in the paste stiffening [7, 8, 10, 11]. In a previous study [12], calcium sulfate hemihydrate was the first chemical species of gypsum dehydration, in which the reaction process began at 106 °C. In another study [13], the occurrence of gypsum dehydration was maximized at temperatures above 125 °C, which led to the cement presenting false setting behavior. An additional factor that must be studied is the storage time of cement in silos. This can cause a false setting as a result of the cement compacting inside the silo, causing a rise in the temperature of the stored cement and, consequently, dehydration reactions in the CPI-S. There are generally aeration systems in the silos which cause the cement to be in constant movement, preventing the increase in cement temperature. This storage period must be sufficient for the cement to lose the necessary heat in the interior of silos before shipping, as this factor can be crucial to the occurrence of problems in the concrete. Thus, the objective of the present study is to evaluate the influences of the storage temperature and time of cement on the false setting behavior of CPI-S-32, which is the Portland cement with the highest production volume in the Brazilian market, used in general construction services with no special properties required.

MATERIALS AND METHODS

Preparation of raw materials: the clinker used in this study came from the clinkerization process of a cement company located in Capanema-PA (Brazil). First, 55 kg of clinker was processed in a jaw crusher (Pragotec, 3020) for the granulometric reduction of the material. After crushing, clinker grinding was performed using a disc mill (Renard, MSA 200-4472) until all the material reached a particle size that was able to pass through a 170-mesh sieve (NBR 14656). The homogenization was performed in a mechanical mixer (Pragotec, Type V) with three volumetric chambers. This stage was necessary to ensure the clinker samples had uniformity. The gypsum was supplied by a cement company located in Capanema-PA (Brazil). The raw material was initially placed in an oven (Fanem, 313/SE) at 55±5 °C for 10 h. After this treatment, 8.0 kg of gypsum was milled in the disc mill until all the material reached between 1.0% and 3.5% of material retained in a 325-mesh sieve. Next, homogenization was performed in a plastic bag and stored in the plastic bag itself. The limestone was provided by a cement company located in Capanema-PA (Brazil). The material was initially dried in the oven at 110±10 °C for 2 h. Next, 8.0 kg of this material was ground in the disc mill until all the material presented a particle size passing through a 170-mesh sieve. Next, the homogenization was performed

in a plastic bag and then stored.

Preparation of CPI-S-32 cement: CPI-S-32 was prepared by mixing 88.5% clinker, 5.5% gypsum, and 6.0% limestone (mass percent) using a ball mill (Pragotec, 1A). The milling operation produced a material with 4.0% to 6.5% of granulometry retained in a 325-mesh sieve. Lastly, homogenization was performed in a plastic bag. Influence of storage temperature and time: samples containing 3.0 kg of CPI-S-32 were stored in the oven at different temperatures (100, 105, 110, 120 and 130 °C) and different times (30, 60 and 120 min) inside metal containers with a height of 52 mm and diameter of 304 mm (simulating the storage in a silo), according to Table I. The samples were stored at atmospheric pressure (1 atm).

Table I - Experimental conditions of the CPI-S-32 sample production.

[Tabela I - Condições experimentais de produção das amostras de CPI-S-32.]

CPI-S-32 sample	Temperature (°C)	Time (min)
CPI-S 100.30	100	30
CPI-S 100.60	100	60
CPI-S 100.120	100	120
CPI-S 105.30	105	30
CPI-S 105.60	105	60
CPI-S 105.120	105	120
CPI-S 110.30	110	30
CPI-S 110.60	110	60
CPI-S 110.120	110	120
CPI-S 120.30	120	30
CPI-S 120.60	120	60
CPI-S 120.120	120	120
CPI-S 130.30	130	30
CPI-S 130.60	130	60
CPI-S 130.120	130	120

For all conditions: the mass composition was 88.5% clinker, 5.5% gypsum, and 6.0% limestone; the response variable was false setting.

Determination of the CPI-S-32 false setting: determination of false setting was performed in accordance with the ASTM C451-13 standard [14]. The CPI-S-32 paste preparation was performed in a room at 23 ± 2 °C and relative humidity higher than 50%, with 350.0±0.1 g of CPI-S-32 cement and 122.5±0.1 g of distilled water. The water-cement CPI-S-32 paste was prepared in a stainless vat, and the mixing was performed with the aid of a spreader for 1.5 min. Vicat apparatus (Solotest, 1.111.001) was used to determine the false setting. The paste had no false setting when the Tetmajer probe of the Vicat equipment reached a value ≥33 mm of the base plate after 30 s, starting from the instant it was released.

Characterization of materials: the raw materials and CPI-S-32 were characterized according to NBR

14656 standard [15]. X-ray fluorescence spectroscopy (XRF) was performed with a spectrometer (Oxford Instr., X-Supreme8000) with incident radiation energy of 10 keV. Cast pellets were produced in a Vulcan Fusion machine (FluXana, 2M). Pressed pellets for the gypsum and CPI-S-32 samples were made in a hydraulic press (Carver, 3853-0), at a pressure of 15.0 kgf/cm². The analyzes performed in the physicochemical characterization of CPI-S-32 were loss on ignition (LOI) [16-19], insoluble residue [20], magnesium oxide and sulfur trioxide contents, specific mass [21], retention index with a 75 µm sieve [22], normal paste consistency [23], initial and final setting time [24], Le-Chatelier's heat expansion [25], and specific area [26] after different cement storage temperatures and times in a silo. By the X-ray diffraction (XRD) analysis, the crystalline phases of the CPI-S-32 cement samples were identified employing a diffractometer (Bruker, D8 Advance) with CuKα radiation, 40 kV voltage, 40 mA current, 2θ range from 2° to 65°, a scan speed of 1 °.min⁻¹, step width of 0.02°, and time step of 0.2 s. The mineral crystalline phases were identified using the MATCH! analysis program, in which the ICSD (Inorganic Crystal Structure Database) crystallographic files were inserted [27, 28]. The thermogravimetry/derivative thermogravimetry (TG/DTG) of CPI-S-32 was performed in a thermal analyzer (Netzsch, STA 449 F1 Jupiter) using an aluminum crucible and a flow rate of 100 mL.min⁻¹ of N₂ from 25 to 1200 °C using a heating rate of 10 °C.min⁻¹. The CPI-S-32 samples in powder form were dispersed on a carbon tape and coated with a thin layer of gold in a metallizer (Kurt J Lesker, 108) and then analyzed using a scanning electron microscope (SEM, Jeol, JJSM-6510LV).

RESULTS AND DISCUSSION

The loss on ignition and compositions of raw materials in terms of oxides used in the preparation of CPI-S-32 samples are shown in Table II. The low loss on ignition value of clinker was due to the decomposition of all organic material and volatiles of the mixture of limestone, clay, and correctives during the clinkerization process. In contrast to the clinker, the limestone showed a higher loss on ignition value combined with the presence of a significant quantity of CO₂ in its composition. As loss on ignition is a chemical requisite established by NBR 16697 standard for CPI-S-32, the added quantity of limestone is limited due to its high values of loss on ignition. As observed in the clinker, the calcium and silicon oxides were the main constituents, corresponding to an amount higher than 80.0% of the chemical composition, proving that clinker is a product that is mostly composed of calcium silicates [3]. For gypsum, the presence of the main components calcium oxide and sulfur trioxide was observed (Table II) with percentages of 32.90% and 40.29%, respectively. The presence of these compounds related to the chemical composition of calcium sulfate was much higher than the other oxides present in the gypsum composition. According to the literature [29], the mean stoichiometric composition of gypsum is around 32.0%

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Raw material	LOI	SiO ₂	Al_2O_3	Fe ₂ O ₃	CaO	MgO	SO ₃	Other oxides ¹
Clinker	0.41	19.84	4.95	3.89	65.06	4.31	0.44	< 0.50
Gypsum	19.10	4.91	0.89	0.52	32.90	0.52	40.29	<1.00
Limestone	38 59	7 95	1 44	0.85	45.22	4 63	_	<1.50

Table II - Loss on ignition (LOI) and chemical composition of raw materials by XRF (wt%). [Tabela II - Perda ao fogo (LOI) e composição química das matérias-primas por FRX (% em massa).]

CaO and 45.0% SO₃. A predominance of calcium oxide is observed in the limestone composition with a content higher than 45.0%. The limestone added to the mixture of clinker and gypsum for the production of CPI-S-32 contained 85.7% CaCO₃, which constituted a value higher than 75% established by NBR 16697 as a condition to add carbonate materials.

Table III shows the result of the percentage of CPI-S-32 retained in the 75 µm sieve, a procedure performed before determining the CPI-S false setting. It was noted that the evaluated granulometry was below the limit established by NBR 16697 and equal to 0.50% retained in the 75 µm sieve. Table IV presents the false setting values for the CPI-S-32 cement samples. It was verified that the storage temperature and time of the cement in the silo influenced the false setting behavior of the cement. It was verified that CPI-S-32 samples were falsely adjusted for cement storage temperatures above 110 °C and silo storage time over 60 min due to gypsum dehydration reactions and hemihydrate formation above 106 °C [12]. It was also observed that the CPI-S-32 presented a false setting of 29 mm when the storage temperature was 110 °C and the cement storage time was 60 min, which represented a much higher value when compared to the CPI-S-32 samples that presented no false setting. There was a transition stage in these specific experimental conditions where there was a need for a deeper investigation in terms of dehydration kinetics to more accurately verify the onset on the loss of plasticity in the CPI-S until its premature stiffness.

After suffering a dehydration process, the gypsum (calcium sulfate dihydrate) contained in the clinker mixture forms calcium sulfate hemihydrate between its products, which deposits dihydrate crystals due to having a higher solubility than gypsum, thereby causing the paste to stiffen [7, 8, 11]. This result was confirmed by the XRD analyses, which showed that the cement with false setting behavior lost the characteristic gypsum peak when compared to the cement that did not show a false setting. The CPI-S-32 storage time in the silo only influences the false setting of the CPI-S-32 if combined with any factor that may cause an anomaly in the cement. Table IV shows that the false adjustment occurred in all CPI-S-32 samples for storage time above 60 min and temperatures above 110 °C. This occurred due to the compaction effect of the cement inside the silo since the temperature of cement grains rises, causing advanced stiffening of the cement paste. However, it was also observed that the cement samples in the studied

Table III - Fineness of CPI-S-32 retained in the 75 μm sieve. [Tabela III - Finura do CPI-S-32 retido na peneira de 75 μm.]

Sample	75 μm sieve (%)	NBR 16697 limit (%)
CPI-S-32	0.50	≤12.0

Table IV - False setting values obtained for CPI-S-32 at different experimental conditions.

[Tabela IV - Valores obtidos de falsa pega do CPI-S-32 para diferentes condições experimentais.]

CPI-S-32 sample	False setting* (mm)
CPI-S 100.30	40
CPI-S 100.60	40
CPI-S 100.120	40
CPI-S 105.30	40
CPI-S 105.60	40
CPI-S 105.120	40
CPI-S 110.30	40
CPI-S 110.60	29
CPI-S 110.120	3
CPI-S 120.30	40
CPI-S 120.60	3
CPI-S 120.120	0
CPI-S 130.30	40
CPI-S 130.60	0
CPI-S 130.120	0

^{* -} no false setting for values ≥33 mm.

temperatures did not show false setting behavior for storage periods close to 30 min, from which it was inferred that there was not enough time for the stored cement to suffer a compaction process and consequently an increase in the grain temperature, and thus the false setting was not possible.

The results obtained for the chemical requirements of loss on ignition and insoluble residue of the CPI-S-32 samples for all storage temperature range and storage time in a silo are shown in Table V. All the produced CPI-S-32 samples presented loss on ignition values below 6.5%, which is the limit established by NBR 16697. It was also noticed that the loss on ignition values did not have a major change when comparing the six samples presenting a false

 $^{^{1}}$ - $K_{2}O+Na_{2}O+TiO_{2}+P_{2}O_{5}+SrO$.

setting to those which did not. Thus, the mass loss of the cement after heating in a furnace around 1000 °C was maintained stable for all the storage temperature and time

Table V - Loss on ignition and insoluble residue values of CPI-S.

[Tabela V - Valores de perda ao fogo e resíduo insolúvel do CPI-S-32.]

CPI-S-32 sample	Loss on ignition (%)	Insoluble residue (%)		
CPI-S 100.30	4.03	0.76		
CPI-S 100.60	4.06	0.62		
CPI-S 100.120	3.84	0.80		
CPI-S 105.30	4.03	0.95		
CPI-S 105.60	3.64	0.77		
CPI-S 105.120	3.45	0.97		
CPI-S 110.30	4.44	1.07		
CPI-S 110.60	4.23	0.75		
CPI-S 110.120	4.01	0.92		
CPI-S 120.30	3.99	0.94		
CPI-S 120.60	3.81	1.01		
CPI-S 120.120	3.83	0.76		
CPI-S 130.30	3.77	0.86		
CPI-S 130.60	3.53	0.91		
CPI-S 130.120	3.64	0.82		

ranges in a silo. Also, the insoluble residue calculated for all conditions showed values less than 3.5%, which is the limit established by NBR 16697. This showed that the false setting phenomena of the studied CPI-S-32 samples did not influence the chemical properties of the insoluble residue.

Table VI shows the obtained results for the physical requisites of the specific mass, normal paste consistency, initial and final setting time, heat expansion, and specific area for both the CPI-S-32 samples, which presented false setting behavior as compared to the ones which did not. It was possible to verify that the specific mass of the cement samples which presented false setting behavior was similar to the samples which did not show alteration. Although it is an optional requisite, its determination is fundamental to calculate the specific area of CPI-S samples. A normal paste consistency was obtained with enough water to provide a standardized pattern. The test value is presented in terms of water/cement mass ratio, which indicated how much water the CPI-S demanded to produce a workable concrete. According to the results, a slight percentage increase of water was needed to produce a workable paste with the increase in both the CPI-S-32 storage temperature and time in the silo. This is probably justified on the dehydration process suffered by the calcium sulfate dehydrated [8, 10], as proven by the XRD analyses of the CPI-S cement 130.120 when compared to CPI-S 100.30. The determination of normal paste consistency identified the necessary quantity of water to be added to the cement for the determination tests of initial and final setting time, constituting a normative physical requisite for CPI-S production.

Table VI - Physical requirements of specific mass, normal consistency, initial and final setting times, heat expansion, and specific area of CPI-S-32.

[Tabela VI - Requisitos físicos de massa específica, consistência normal, tempo de início e fim de pega, expansibilidade a quente e área específica das amostras de CPI-S-32.]

CPI-S-32 sample	Specific mass (g/cm³)	Normal consistency (%)	Initial setting (h:min)	Final setting (h:min)	Heat expansion (mm)	Specific area (m²/kg)
CPI-S 100.30	3.09	25.0	3:10	4:05	0.5	386
CPI-S 100.60	3.10	25.0	3:05	4:00	0.5	385
CPI-S 100.120	3.11	25.2	3:20	4:10	0.5	385
CPI-S 105.30	3.09	25.6	3:05	4:15	0.8	386
CPI-S 105.60	3.12	25.0	2:50	4:15	0.5	384
CPI-S 105.120	3.11	26.4	3:15	4:40	0.0	385
CPI-S 110.30	3.11	25.0	3:00	4:05	0.5	385
CPI-S 110.60	3.11	25.6	3:35	4:25	1.0	384
CPI-S 110.120	3.10	26.6	3:50	5:00	0.0	385
CPI-S 120.30	3.11	25.4	3:05	4:10	0.0	385
CPI-S 120.60	3.12	26.6	3:35	4:45	0.0	385
CPI-S 120.120	3.11	28.0	3:55	5:00	0.0	385
CPI-S 130.30	3.12	26.0	3:55	4:30	0.5	386
CPI-S 130.60	3.12	28.0	3:50	4:55	0.0	386
CPI-S 130.120	3.13	30.4	4:00	5:05	0.0	385

Table VII - Chemical composition of CPI-S-32 samples by XRF (wt%).
[Tabela VII - Composição química das amostras de CPI-S-32 por FRX (% em massa).]

CPI-S-32 sample	SiO,	Al ₂ O ₃	Fe ₂ O ₃	CaO	MgO	SO ₃	P,O ₅	TiO,	Na,O	K,O	SrO
CPI-S 100.30	18.65	4.56	3.96	65.12	3.79	2.98	0.33	0.24	0.05	0.20	0.08
CPI-S 100.60	18.55	4.62	3.93	65.32	3.71	2.99	0.33	0.24	0.05	0.19	80.0
CPI-S 100.120	18.64	4.50	3.95	65.37	3.72	2.98	0.33	0.23	0.04	0.18	80.0
CPI-S 105.30	18.61	4.62	3.96	65.29	3.76	2.88	0.32	0.24	0.05	0.18	0.07
CPI-S 105.60	18.47	4.51	3.98	65.62	3.70	2.87	0.33	0.24	0.03	0.19	0.07
CPI-S 105.120	18.56	4.57	3.97	65.23	3.84	2.94	0.34	0.24	0.04	0.19	80.0
CPI-S 110.30	18.46	4.57	3.97	65.33	3.72	2.99	0.34	0.25	0.05	0.19	0.10
CPI-S 110.60	18.52	4.58	3.95	65.19	3.83	3.03	0.33	0.25	0.04	0.20	0.08
CPI-S 110.120	18.68	4.56	3.94	65.03	3.87	3.02	0.33	0.24	0.04	0.20	80.0
CPI-S 120.30	18.72	4.63	3.94	65.04	3.79	2.98	0.34	0.24	0.06	0.20	0.07
CPI-S 120.60	18.50	4.53	3.96	65.32	3.75	3.03	0.35	0.23	0.05	0.18	0.07
CPI-S 120.120	18.54	4.52	3.95	65.30	3.80	3.01	0.33	0.23	0.04	0.18	0.08
CPI-S 130.30	18.56	4.56	3.96	65.35	3.76	2.91	0.33	0.24	0.04	0.19	0.10
CPI-S 130.60	18.46	4.53	3.98	65.43	3.79	2.91	0.34	0.24	0.04	0.20	80.0
CPI-S 130.120	18.44	4.52	3.97	65.46	3.78	2.92	0.34	0.25	0.05	0.20	0.08

In evaluating the setting times, all the CPI-S-32 samples presented an initial setting time longer than 1 h, which is the limit established by NBR 16697 as the physical demand in CPI-S-32 production and a final setting time less than 10 h. The results showed that all samples which presented false setting behavior had more difficulties for plasticity loss of the water and cement mixture, which meant longer times of setting, showing that the increase in temperature caused an increase in the initial and final setting time of this cement [7, 10]. All CPI-S-32 samples exhibited Le-Chatelier heat expansion values less than 5.0 mm, which is the limit imposed by NBR 16697. This meant that all CPI-S-32 samples did not present volumetric variation due to the hydration process of cement [3, 7]. It was noted that the specific area of the samples of CPI-S cement presented values with low variation, both for samples that presented false setting and those that not. The specific area had a direct relationship with the cement granulometry due to the grains presenting in its composition; moreover, the CPI-S samples were prepared with the same granulometry (retained material in 75 µm sieve), justifying the obtained results.

Table VII presents the chemical constituents of the CPI-S-32 samples. The calcium and silicon oxides were the major constituents in the studied samples and together were responsible for the hydration reactions of silicates, corresponding to more than 60% of volume in the cement reactions [6]. The variation of the calcium and silicon elements in the cement directly interferes in the C₃S and C₂S anhydrous phases, in which a higher concentration of calcium leads to a preponderant production of alite, and is the main element responsible for the hydrated C-S-H and Ca(OH)₂ phases [30, 31]. The aluminum and iron oxides are no less important, being responsible for aluminate reactions

and corresponding to 20% of the volume of the reactions which occur in the cement. The oxide percentages were not influenced by the false setting phenomena of the CPI-S-32 samples. The magnesium oxide and sulfur trioxide were below the values required by NBR 16697, being lower than 6.5% and 4.5%, respectively. The low MgO content makes the CPI-S-32 normally self-hydratable and loses the capacity to cause cracks by expanding inside the already hardened concrete. Likewise, low values of SO₃ originated from the gypsum decreases the possibility of interference in the initial strength of the cement, as shown by theoretical studies [10].

X-ray diffraction (Fig. 1) and TG/DTG analyses (Fig. 2) were performed on two CPI-S-32 samples, sample CPI-S-32 130.120, which presented false setting, and CPI-S-32 100.30 sample, which did not show false setting. It was possible to note the presence of the main crystalline compounds in both samples as alite (C₃S), belite (C₂S), tricalcium aluminate (C₂A), tetracalcium ferroaluminate (C4AF), and calcite (CaCO₃) [32-39]. In addition to these compounds, the presence of gypsum (CaSO₄.2H₂O) was observed in the CPI-S-32 100.30 sample at 11.6° and 20.8° (2θ) . Also, the disappearance of dihydrated calcium sulfate peaks was observed in the CPI-S-32 130.120 sample, since the gypsum was consumed by dehydration process forming basanite (CaSO₄.0.5H₂O). As this compound is about 5 times more soluble than gypsum, it allowed the formation of a supersaturated solution that tended to deposit dihydrate crystals, which caused early stiffness in the mass of CPI-S-32 130.120, characterizing the false setting in this studied sample [7, 8, 10-12].

The results of thermogravimetric analysis (TG/DTG) for the CPI-S-32 100.30 and CPI-S-32 130.120 samples are presented in Fig. 2. The CPI-S-32 samples presented

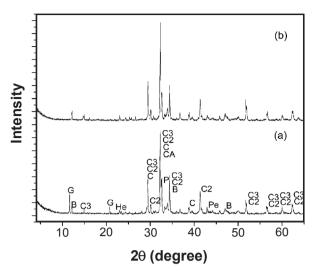


Figure 1: X-ray diffraction patterns of CPI-S-32 100.30 (a) and CPI-S-32 130.120 (b) samples. C3: alite; C2: belite; CA: tricalcium aluminate; C: calcite; B: brownmilerite; G: gypsum; Pe: periclase; He: hematite.

[Figura 1: Difratogramas de raios X das amostras CPI-S-32 100.30 (a) e CPI-S-32 130.120 (b). C3: alita; C2: belita; CA: aluminato tricálcico; C: calcita; B: brownmilerita; G: gipsita; Pe: periclásio; He: hematite.]

continuous mass loss with lower or larger loss rates in the analyzed temperature range. The thermograms of the CPI-S-32 100.30 and CPI-S-32 130.120 samples showed the decomposition in three stages. The first occurred between 30 and 180 °C, with the highest band maximum identified by the DTG at 130 °C being attributed to water loss from the dehydration of calcium sulfate dihydrate, equivalent to 1.3% for CPI-S-32 100.30 and 0.2% for CPI-S-32 130.120. Between 180 and 500 °C, the highest band value at 440 °C was observed with a mass loss of 0.7% for CPI-S-32 100.30 and 0.6% for CPI-S-32 130.120, corresponding to dehydration of calcium hydroxide [40]. Then between 500 and 800 °C, at the highest band maximum at 730 °C, mass losses of 3.2% for CPI-S-32 100.30 and 3.4% for CPI-S-32 130.120 were observed due to the loss of CO₂ by decomposition of calcium carbonate. Thus, the total mass loss was 5.2% for CPI-S-32 100.30 and 4.2% for CPI-S-32 130.120. Accordingly, it was verified that the mass loss of the CPI-S 100.30 sample was higher than CPI-S 130.120 because of the higher dehydration mass loss of calcium sulfate dehydrate. It was observed that the decomposition band of dehydrated calcium sulfate of the CPI-S 130.120 cement was smaller than that of the CPI-S 100.30 cement,

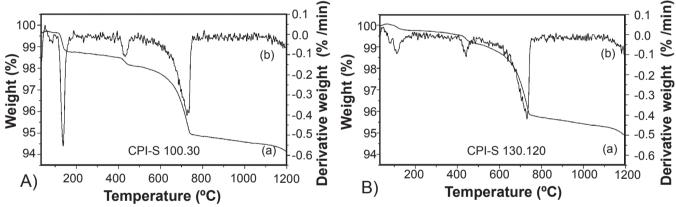


Figure 2: TG (a) and DTG (b) curves of the CPI-S-32 100.30 (A) and CPI-S-32 130.120 (B) cement samples. [Figure 2: Curvas de TG (a) e DTG (b) das amostras CPI-S-32 100.30 (A) e CPI-S-32 130.120 (B).]

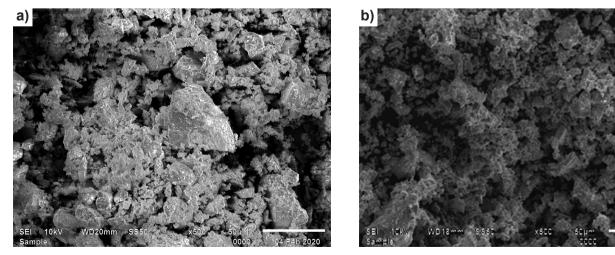


Figure 3: SEM micrographs of the samples CPI-S-32 100.30 (a) and CPI-S-32 130.120 (b). [Figure 3: Micrografias de MEV das amostras CPI-S-32 100.30 (a) e CPI-S-32 130.120 (b).]

probably due to the partial or total decomposition of dehydrated calcium sulfate added during the storage, which can transform into hemihydrate calcium sulfate.

Analyzing Fig. 3, it was noted that the sample CPI-S-32 130.120 had a uniform morphology and little porosity, compared to the sample CPI-S-32 100.30, which can be attributed to the interaction temperature and time used. Probably this little porosity made it difficult to hydrate the constituents of the cement, which may have led to a false setting.

CONCLUSIONS

Based on the performed tests and analyses, it was concluded that the storage temperature above 110 °C and the storage time of CPI-S-32 above 60 min influenced the false setting behavior of cement. Six of 15 samples of analyzed cement showed this anomaly, with the cement paste stiffening instantaneously without the release of characteristic heat. The false setting did not influence the parameters of loss on ignition, insoluble residue, fineness retained in the 75 μm sieve, MgO and SO, content, initial and final setting time, heat expansion, specific mass, specific area, and normal paste consistency as established by NBR 16697 standard. X-ray diffraction analysis showed in a cement sample with false setting behavior (CPI-S-32 130.120 - stored at 130 °C and 120 min) the absence of CaSO₄.2H₂O phase, since the gypsum was consumed by the dehydration process during the storage stage and transformed into CaSO₄.0.5H₂O, differently from that observed in a sample that did not show false setting (CPI-S-32 100.30 - stored at 100 °C and 30 min) in which the gypsum was present. This was confirmed by thermogravimetric analysis that showed a higher mass loss of the CPI-S-32 100.30 cement sample (5.2%) than the CPI-S-32 130.120 cement (4.2%), which the difference mainly occurred between 30 and 180 °C when water loss from the dehydration of the calcium sulfate dihydrate occurred. In the other temperature ranges, the mass losses by calcium hydroxide dehydration and due to the loss of CO₂ by decomposition of calcium carbonate were similar. The results showed the optimal operating conditions of the storage temperature and time of the cement in order to avoid false setting and a loss in the CPI-S-32 product quality. It was identified that the CPI-S-32 could present a false setting in the storage temperature range of 110 to 130 °C combined with storage time from 60 to 120 min, which causes financial loss to the industry. As a false setting is a specific quality parameter, mainly for CPI-S-32 cement, its control is necessary to attend the consumers. Higher consumption of energy and raw materials can be avoided by knowing that the temperature and the storage time of the cement in silos are variables that need to be controlled by the cement industry.

ACKNOWLEDGMENTS

The authors are grateful to CAPES, FAPITEC-SE, the

Federal University of Sergipe, the Laboratory of Industrial Chemistry, and LMDCEM/UFS for their support in the accomplishment and participation of the obtained results. To the cement industry located in Capanema-PA, for the supply of clinker, gypsum, and limestone and support in carrying out the experimental activities.

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- (Rec. 13/12/2019, Rev. 08/02/2020, 06/04/2020, Ac. 09/04/2020)

