

Synthesis and Crystal Structure of Three New Quaternary Compounds in the system Cu-Mn-III-Se₃ (III = Al, Ga, In)

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Chalcogenide alloys CuMnAlSe $_3$, CuMnGaSe $_3$ and CuMnInSe $_3$, new members of the semiconductor system I-II-III-VI $_3$, were synthetized and structurally characterized by the Rietveld method using X-ray powder diffraction data. All compounds crystallize in the tetragonal space group $P\overline{4}$ 2c (N° 112) with a CuFeInSe $_3$ -type structure.

Keywords: Chalcogenides, Semiconductors, Chemical synthesis, X-ray powder diffraction, Structural characterization.

1. Introduction

The Cu-III-Se, (III= Al, Ga, In) ternary chalcopyrite family have been object of a great quantity of work in the last years, because they form an wide group of semiconductor materials with diverse optical and electrical properties1. These materials crystallize with tetragonal symmetry in the space group $\overline{14}$ 2d (N°122), and the addition of a Fe-Se binary compound produces alloys of the type (Cu-III- Se_2 _{1-x} $(FeSe)_x^2$. Some previous results on the formation, structural characterization, thermal and magnetic properties on these alloys with compositions Cu-Fe-III-Se, $(x = \frac{1}{2})$, $Cu-Fe_2-III-Se_4$ (x = $\frac{2}{3}$) and $Cu_2-Fe-III-Se_5$ (x = $\frac{1}{3}$) have been reported³⁻¹⁵. All these phases fulfill the rules of formation of adamantane compounds² and belong to the normal semiconductor compound families of the third, fourth and fifth-order derivatives of the II-VI binary semiconductors, respectively¹⁶. Adamantane compounds are binary, ternary or quaternary normal tetrahedral structure compounds which are closely related to either cubic or hexagonal diamond.

The first crystal structure characterization of one I-II-III-VI₃ semiconductor member, CuFeInSe₃, indicated a degradation of symmetry from the chalcopyrite structure I\(\bar{4}\)2d to a related structure P\(\bar{4}\)2c⁶. In this work, we report a detailed synthesis and structural analysis of three new members of this family; CuMnAlSe₃, CuMnGaSe₃ and

CuMnInSe₃, which was performed using X-ray powder diffraction by means of the Rietveld method.

2. Experimental Procedures

2.1 Synthesis

Ingots of Cu-Mn-III-Se₃ (III=Al, Ga, In) were prepared by the melt and annealing technique. Starting materials (Cu, Mn, Al, Ga, In and Se), with a nominal purity of at least 99.99 % (GoodFellow) in the stoichiometric ratio, were mixed together in an evacuated and sealed quartz tube with inner walls previously carbonized. Polycrystalline ingots of about 1 g were prepared by the usual melting and annealing technique, lowering the temperature from 1500 to 850 K at a rate of 20 K/h, keeping this temperature for 30 days, and finally, cooling to room temperature at a rate of 10 K/h. Previous experience indicates that this procedure usually gives samples showing conditions corresponding to equilibrium near room temperature.

2.2 Chemical analysis

Compositional analysis of the ingots was determined at several points by energy dispersive X-ray (EDX) analysis using a Kevex Model Delta-3 system connected to a Hitachi Model S-2500 scanning electron microscope (SEM). In each

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case, the average chemical composition of the central part of the ingot from which the crystals were cut, gave the atomic percentage in good agreement with the ideal composition 1:1:1:3 The error in standardless analysis was around 5 %.

2.3 X-ray powder diffraction

For the X-ray analysis, small quantities of the samples were ground mechanically in an agate mortar and pestle. The resulting fine powders, sieved to 106μ , were mounted on a flat zero-background holder covered with a thin layer of petroleum jelly. The X-ray powder diffraction data were collected at 293(1) K, in θ/θ reflection mode using a Siemens D5005 diffractometer equipped with an X-ray tube (CuK α radiation: λ = 1.54056 Å; 40kV, 30mA) using a secondary beam graphite monochromator. A fixed aperture and divergence slit of 1 mm, a 1 mm monochromator slit, and a 0.1 mm detector slit were used. The specimens were scanned from 10° - 100° 2 θ , with a step size of 0.02° and counting time of 40s. Quartz was used as an external standard.

3. Results and Discussion

The X-ray diffractograms of three alloys Cu-Mn-III-Se₃ (III=Al, Ga, In) showed single phases. The powder patterns were indexed using the program Dicvol04¹⁷, and tetragonal

cells with similar magnitudes to the parent chalcopyrite structures, CuAlSe₂¹⁸, CuGaSe₂¹⁹, CuInSe₂²⁰ were founds. Systematic absences are consistent with a P-type Bravais lattice. A detailed pattern examination taking in account the sample composition, cell parameters and lattice-type, suggested that all compounds are isostructural with previously reported CuFeInSe₃⁶ which crystallizes in the space group P42c.

The Rietveld refinements²¹ of the structures were carried out using the Fullprof program²². The atomic coordinates of CuFeInSe₃⁶ were used as starting model for each refinement. The angular dependence of the peak full width at half maximum (FWHM) was described by the Caglioti's formula²³. Peak shapes were described by the parameterized Thompson-Cox-Hastings pseudo-Voigt profile function²⁴. The background variation was described by a polynomial with six coefficients. The thermal motion of the atoms was described by one overall isotropic temperature factor. The results of the Rietveld refinement for the three alloys are summarizes in Table 1. Fig. 1 shows the observed, calculated and difference profile for the final cycle of Rietveld refinements. Atomic coordinates, isotropic temperature factor, bond distances and angles for each compound are shown in Tables 2, 3 and 4.

CuMnAlSe₃, CuMnGaSe₃ and CuMnInSe₃ are normal adamantane-structure compounds², where occurs a degradation of symmetry from the chalcopyrite structure I42d to a related

Table 1. Rietveld refinement results for the alloys CuMnAlSe, CuMnGaSe, and CuMnInSe,

	CuMnGaSe ₃	CuMnInSe ₃
1019.49	1133.43	1253.67
5.6034(6)	5.6230(4)	5.7907(5)
10.977(1)	11.028(1)	11.648(1)
1.96	1.96	2.01
0.98	0.98	1.00
344.66(6)	348.68(5)	390.58(8)
1	1	1
tetragonal	tetragonal	tetragonal
P-42c (N° 112)	P-42c (Nº 112)	P-42c (Nº 112)
4.91	5.40	5.33
298(1)	298(1)	298(1)
1.54056 Å	1.54056 Å	1.54056 Å
10-100	10-100	10-100
0.02	0.02	0.02
40	40	40
4501	4501	4501
134	134	134
9.2	9.6	9.6
6.7	6.9	6.9
1.4	1.4	1.4
8.2	8.9	8.9
9.5	9.6	9.6
	5.6034(6) 10.977(1) 1.96 0.98 344.66(6) 1 tetragonal P-42c (N° 112) 4.91 298(1) 1.54056 Å 10-100 0.02 40 4501 134 9.2 6.7 1.4 8.2 9.5	5.6034(6) 5.6230(4) 10.977(1) 11.028(1) 1.96 1.96 0.98 0.98 344.66(6) 348.68(5) 1 1 tetragonal P-42c (N° 112) P-42c (N° 112) 4.91 5.40 298(1) 298(1) 1.54056 Å 1.54056 Å 10-100 10-100 0.02 0.02 40 40 4501 4501 134 134 9.2 9.6 6.7 6.9 1.4 1.4 8.2 8.9 9.5 9.6

 $\begin{array}{l} R_{p} = 100 \; \Sigma|_{yobs-ycalc}| \; / \; \Sigma|_{yobs}| \; R_{wp} = 100 \; [\Sigma_{w}|_{yobs-ycalc}|^{2} \; / \; \Sigma_{w}|_{yobs}|^{2}]^{1/2} \; \chi^{2} = [R_{wp} \; / \; R_{exp}] \; R_{B} = 100 \; \Sigma_{k}|I_{k} - I_{c \; k}| \; / \; \Sigma_{k} \; |I_{k}| \; R_{exp} = 100 \; [(N-P+C) \; / \; \Sigma_{w}(_{yobs}^{2})]^{1/2} \; \chi^{2} = [R_{wp} \; / \; R_{exp}] \; R_{B} = 100 \; \Sigma_{k}|I_{k} - I_{c \; k}| \; / \; \Sigma_{k} \; |I_{k}| \; R_{exp} = 100 \; [(N-P+C) \; / \; \Sigma_{w}(_{yobs}^{2})]^{1/2} \; \chi^{2} = [R_{wp} \; / \; R_{exp}] \; R_{B} = 100 \; \Sigma_{k}|I_{k} - I_{c \; k}| \; / \; \Sigma_{k} \; |I_{k}| \; R_{exp} = 100 \; [(N-P+C) \; / \; \Sigma_{w}(_{yobs}^{2})]^{1/2} \; \chi^{2} = [R_{wp} \; / \; R_{exp}] \; R_{exp} = 100 \; [N-P+C] \; / \; \Sigma_{w}(_{yobs}^{2}) \; R_{exp} = 100 \; [N-P+C] \; / \; \Sigma_{w}(_{y$

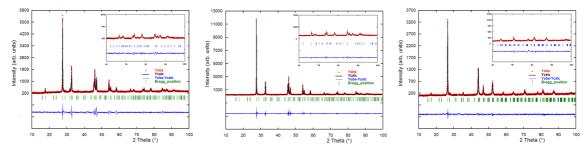


Figure 1. Rietveld final plots of a) CuMnAlSe₃, b) CuMnGaSe₃ and c) CuMnInSe₃. The lower curve represents the difference between observed and calculated patterns. The Bragg reflections are indicated by vertical bars.

Table 2. Atomic coordinates, isotropic temperature factors, bond distances (Å) and angles (°) for CuMnAlS_{e3}, derived from the Rietveld refinement.

Atom	Ox.	Site	x	Y	z	foc	$B(Å^2)$
Cu	+1	2c	0	1/2	1/4	1	0.6(5)
Mn	+2	2e	0	0	0	1	0.6(5)
Al	+3	2b	1/2	0	1/4	1	0.6(5)
Cul	+1	2f	1/2	1/2	1/2	1/3	0.6(5)
Mn1	+2	2f	1/2	1/2	1/2	1/3	0.6(5)
A11	+3	2f	1/2	1/2	1/2	1/3	0.6(5)
Se	-2	8n	0.257(1)	0.253(1)	0.125(1)	1	0.6(5)
Cu-Se	2.423(8)	Mn-Se	2.443(8)	Al-Se ⁱ	2.397(8)	2.397(8) 2f-Se ⁱⁱ	
Seiii-Cu-Seiv	106.7(2) x2	Se ⁱⁱⁱ -Cu-Se ^v		111.5(2) x2	Se ⁱⁱⁱ -Cu-Se		110.3(2) x2
$Se^{vi}\text{-}Mn\text{-}Se^{vii}$	108.0(2) x2	Sevii-Mn-Se		112.6(2) x2	Se-Mn-Seviii		108.0(2) x2
Se-Al-Seviii	111.7(2) x2	Se-Al-Se ⁱ		106.8(2) x2	Se-Al	l-Se ^{ix}	110.1(2) x2
Sevi-M-Se	109.5(2) x4	Se-M-Se ⁱⁱ		109.4(2) x2			

Symmetry codes: (i) 1-x, y, 0.5-z; (ii) 1-x, 1-y, z; (iii) -x, 1-y, z; (iv) -x, y, 0.5-z; (v) x, 1-y, 0.5-z; (vi) -y, x, -z; (vii) -x, -y, z; (viii) y, -x, -z; (ix) 1-x, -y, z.

Table 3. Atomic coordinates, isotropic temperature factors, bond distances (Å) and angles (°) for CuMnGaS_{e3}, derived from the Rietveld refinement.

Atom	Ox.	Site	x	y	z	foc	$B(Å^2)$	
Cu	+1	2c	0	1/2	1/4	1	0.6(4)	
Mn	+2	2e	0	0	0	1	0.6(4)	
Ga	+3	2b	1/2	0	1/4	1	0.6(4)	
Cul	+1	2f	1/2	1/2	1/2	1/3	0.6(4)	
Mn1	+2	2f ½		1/2	1/2	1/3	0.6(4)	
Gal	+3	2f	1/2	1/2	1/2	1/3	0.6(4)	
Se	-2	8n	0.255(1)	0.257(1)	0.124(1)	1	0.6(4)	
Cu-Se	2.419(8)	Mn-Se	2.452(8)	Ga-Se ⁱ	2.432(8)	2f-Se ⁱⁱ	2.374(8)	
Seiii-Cu-Seiv	107.1(2) x2	Se ⁱⁱⁱ -0	Cu-Se ^v	109.5(2) x2	Se ⁱⁱⁱ -Cu-Se		111.1(2) x2	
Sevi-Mn-Sevii	108.4(2) x2	Sevii-Mn-Se		111.7(2) x2	Se-Mn-Seviii		108.4(2) x2	
Se-Ga-Seviii	110.9(2) x2	Se-Ga-Se ⁱ		107.6(2) x2	Se-Ga	a-Se ^{ix}	110.0(2) x2	
Sevi-M-Se	109.5(2) x4	Se-M-Se ⁱⁱ		109.4(2) x2				

Symmetry codes: (i) 1-x, y, 0.5-z; (ii) 1-x, 1-y, z; (iii) -x, 1-y, z; (iv) -x, y, 0.5-z; (v) x, 1-y, 0.5-z; (vi) -y, x, -z; (vii) -x, -y, z; (viii) y, -x, -z; (ix) 1-x, -y, z.

structure $P\overline{4}2c$. This situation can be clearly seen in Fig. 2 where a comparison is made between the chalcopyrite Cu-III-Se₂ $I\overline{4}2d$ structure and the $P\overline{4}2c$ structure of Cu-Mn-III-Se₃ (III= Al, Ga, In). Therefore, in the quaternary structures,

the introduction of an additional cation (Mn) produces an effect of "dilution" of this cation in the chalcopyrite structure leaving the cell volume almost unchanged⁶. Table 5 show a comparison between the unit cell parameters and the bond distances for the three phases of both families of compounds.

Table 4. Atomic coordinates, isotropic temperature factors, bo	ond distances (Å) and angles (°) for C	uMnInS ₆₃ , derived from the Rietveld
refinement		

Atom	Ox.	Site	x	У	z	foc	B (Å ²)
Cu	+1	2c	0	1/2	1/4	1	0.6(4)
Mn	+2	2e	0	0	0	1	0.6(4)
In	+3	2b	1/2	0	1/4	1	0.6(4)
Cu1	+1	2f	1/2	1/2	1/2	1/3	0.6(4)
Mn1	+2	2f ½		1/2	1/2	1/3	0.6(4)
In1	+3	2f	1/2	1/2	1/2	1/3	0.6(4)
Se	-2	8n	0.227(1)	0.255(1)	0.124(1)	1	0.6(4)
Cu-Se	2.428(8)	Mn-Se	2.448(8)	In-Se ⁱ	2.614(8)	2.614(8) 2f-Se ⁱⁱ	
Se ⁱⁱⁱ -Cu-Se ^{iv}	114.5(2) x2	Se ⁱⁱⁱ -0	Cu-Se ^v	108.6(2) x2	Se ⁱⁱⁱ -Cu-Se		105.6(2) x2
$Se^{vi}\text{-}Mn\text{-}Se^{vii}$	110.4(2) x2	Sevii-Mn-Se		107.7(2) x2	Se-Mn-Seviii		110.4(2) x2
Se-In-Seviii	105.6(2) x2	Se-In-Se ⁱ		111.2(2) x2	Se-In	-Se ^{ix}	111.7(2) x2
Sevi-M-Se	108.5(2) x4	Se-M-Se ⁱⁱ		111.5(2) x2			

Symmetry codes: (i) 1-x, y, 0.5-z; (ii) 1-x, 1-y, z; (iii) -x, 1-y, z; (iv) -x, y, 0.5-z; (v) x, 1-y, 0.5-z; (vi) -y, x, -z; (vii) -x, -y, z; (viii) y, -x, -z; (ix) 1-x, -y, z

Table 5. Comparative table of unit cell parameters and bond distances for the Cu-III-Se2 chalcopyrite compounds and the related Cu-Mn-III-Se3 alloys (III= Al Ga, In). ([*] = this work).

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Compound	SG	a (Å)	c (Å)	η	V (Å ³)	Cu-Se (Å)	Mn-Se (Å)	III-Se (Å)	Ref.
CuAlSe ₂	I 4 2d	5.606(5)	10.90(1)	0.97	342.6(5)	2.438(1)	-	2.373(1)	[18]
$CuGaSe_2$	$1\overline{4}2d$	5.614(1)	11.022(1)	0.98	347.4(1)	2.446(9)	-	2.387(9)	[19]
CuInSe ₂	$1\overline{4}2d$	5.781(1)	11.642(3)	1.01	389.1(2)	2.432(1)	-	2.591(1)	[20]
CuMnAlSe ₃	P 4 2c	5.6034(6)	10.977(1)	0.98	344.66(6)	2.423(8)	2.443(8)	2.397(8)	[*]
$CuMnGaSe_3$	P42c	5.6230(4)	11.028(1)	0.98	348.68(4)	2.419(8)	2.453(8)	2.432(8)	[*]
CuMnInSe ₃	P42c	5.7907(5)	11.648(2)	1.00	390.58(8)	2.428(8)	2.448(8)	2.614(8)	[*]

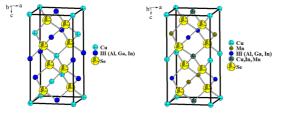


Figure 2. Unit cell diagram for the chalcopyrite Cu-III-Se_2 compared to the Cu-Mn-III-Se_3 compounds.

The structure of the tetragonal phases Cu-Mn-III-Se₃ can be described as derivative of the sphalerite structure². In this structure the Se atoms form a close-packed arrangement where each anion is coordinated by four cations located at the corners of a slightly distorted tetrahedron. All cations are similarly coordinated by four anions. Fig. 3 shows a polyhedral view of the crystal structure with four types of atoms-centered tetrahedra MS₄ [CuS₄, MnS₄, IIIS₄ and (CuInMn)S₄] where all polyhedra are oriented in the same direction and are connected by the corners.

An important structural characteristic of the compounds under study is the parameter of tetragonal lattice distortion, which is determined as a deviation of the ratio $\eta = c/2a$ (a and

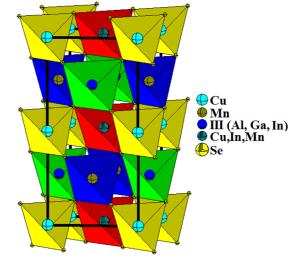


Figure 3. Polyhedral view of the Cu-Mn-III-Se₃ structures showing tetrahedral units.

c are unit-cell parameters) from unity²⁵. The Table 5 contains the a, c, and η values for the Cu-Mn-III-Se₃ compounds. One can see that η is close to unity for all compositions, which is indicative of small lattice distortions in the samples synthesized.

The Cu-Se, Mn-Se and III (Al,Ga,In)-Se bond distances in all compounds (Tables 2, 3 and 4) are in good agreement with those observed in the parent chalcopyrite structures (Table 5) and other quaternary adamantane structure compounds such as CuFe(Al,Ga,In)Se₃^{6,12}, CuNi(Ga,In)Se₃¹³, CuFe₂(Al,Ga,In)Se₄^{9,14}, Cu₂FeSnSe₄²⁶, Cu₂ZnGeSe₄²⁷, Cu₂ZnSnSe₄²⁸, Cu₂(Cd,Mn)GeSe₄²⁹ and CuGaMnSe₂³⁰.

4. Conclusions

The crystal structure of the semiconductor compounds CuMnAlSe₃, CuMnGaSe₃ and CuMnInSe₃ were determined using X-ray powder diffraction data. All compounds crystallize in the tetragonal space group P42c, with a sphalerite derivative structure, and are isomorphic with CuFeInSe₃.

5. Acknowledgements

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