Improving usual and dielectric properties of ceramic high voltage insulators

(Melhora de propriedades usuais e dielétricas de isolantes cerâmicos de alta voltagem)

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Abstract

The optimization of the formulation (nature and content of chemical components) and the technical parameters of the shaping-up process of ceramic insulators (in replacement of those in glass) are the objectives of this study. The decontamination of SiO_2 (solid-liquid extraction of Fe_xO_y) and the increase of both BaO and ZrO_2 amounts (while Na_2O is partially eliminated) and the firing temperature allowed a significant increase of all the phys-mechanical and insulating properties at the commercial frequency 50-60 Hz. Samples with 15 mm diameter and 2 mm thickness and others for mechanical properties were needed. The characterization concerned $Ba-Zr(Al_xSi_{1.x})O_3$ ceramic samples. By data computing of the test results, a new "computed" formulation (Ba-stabilized and Zr doped mullite) showed excellent phys-mechanical (density, viscosity, flexural and traction strengths) and insulating properties (transversal resistivity, dielectric strength).

Keywords: dielectric ceramic, insulators, mullite, barium oxide, zirconia.

Resumo

A otimização da formulação (natureza e conteúdo dos componentes químicos) e parâmetros técnicos do processo de conformação de isolantes cerâmicos (para substituição dos vítreos) são os objetivos deste estudo. A descontaminação de SiO_2 (extração sólidolíquido de Fe_xO_y) e o aumento das quantidades de BaO e ZrO_2 (enquanto Na_2O é parcialmente eliminado) e a temperatura de queima permitiram uma significativa melhora nas propriedades físico-mecânicas e isolantes na frequência comercial de 50-60 Hz. Foram necessárias amostras com 15 mm de diâmetro e 2 mm de espessura e outras para avaliação de propriedades mecânicas. Foi feita a caracterização de amostras cerâmicas de Ba- $Zr(Al_xSi_{1.x})O_3$. Após computação dos resultados dos testes, uma nova formulação "computada" (mullita estabilizada com Ba e dopada com Zr) mostrou excelentes propriedades físico-mecânicas (densidade, viscosidade, resistências a flexão e tração) e isolantes (resistividade transversal, rigidez dielétrica).

Palavras-chave: cerâmica dielétrica, isolantes, mulita, óxido de bário, zircônia.

INTRODUCTION

Ceramic insulators are widely used in power transmission as well as distribution lines [1-5]. It is also reported that there are many failures of ceramics insulators especially in polluted areas [4, 6-8]. One method to improve the performance of these insulators is optimizing, first, their chemical composition and second, the shaping up and elaboration processes.

In this paper, the mechanical and electrical properties of porcelain ceramic insulator fired and/or sintered at different temperatures have been investigated along with microstructural characterization using scanning electron microscopy (SEM) and X-ray diffraction (XRD) in order to understand the structure–property relationship of the ceramic insulator.

MATERIALS AND METHODOS

The simplicity of the production process and its cost were

the challenge of this study. To obtain Ba-Zr(Al_x-Si_{1,x})O₃ samples of fired and/or sintered samples with different composition (x = 1.0, 0.2, 0.4, 0.6, 0.8, 1.0) and according to well known (dielectric and mechanical) characteristics of alkaline, earth-alkaline, silicone and metal oxides and trying to compare different (barium-zirconium)-doped mullite, the following local raw materials were used: high SiO₂ sea sand: 0.0098 to 0.0051% Fe₂O₂ decontaminated during a solid-liquid diffusion of Fe_QO by [0.25 HF + 0.75 H₂SO₄] in a milling tank; high Al₂O₃ clay with 1.25% Na₂CO₃ + 3.18% CaCO₃; barium carbonate with 0.25 BaSiO_3 : with its large atomic radius ($r_{Ba++} = 13.5 \text{ nm}$), it may contribute to increase the dielectric properties of any silicate formulation while replacing Na+ and K+; pure zirconium oxide: with a higher melting point, chemical stability and mechanical resistance; Na-feldspar (albite: NaAlSi₂O₆): used to decrease the sintering temperature.

The dielectric strength values were determined

according to CEI 93 normalization on breakdown high voltage apparatus ANM-90. The transversal resistivity values were calculated as the ratio $\rho_t = R_t$ (S/e), where R_t (Ω) is the measured transversal resistance, S (m^2) the electrode area, and e (m) the sample thickness. Flexural strength (tree points flexion) values were measured with Netzsch Dynamometer (401/3 model) on parallelepiped samples (150 x 20 x 12 mm³).

Optimizing the formulation

The optimized chemical formulation (amount of each used raw material) must correspond in the same time to the minimal value of the absorption (increase of loss after two hours in boiling water), the maximal value of the flexural strength, the maximal value of the dielectric strength, and the maximal value of the transversal resistivity.

In computing data of simply fired samples, the fourth

formulation is the best according to the following values: $SiO_2 \sim 38.0\%$, $Al_2O_3 \sim 50.5\%$, BaO $\sim 7.5\%$, (K-Na)O $\sim 3.0\%$, other 1.0%. This formulation permitted the following properties: Abs. $\sim 15.22\%$, $\sigma_{Fl} \sim 358$ Kgf.cm⁻², $\rho_{tr} \sim 5.0$ Ω .cm, Rd ~ 6.2 kV/mm.

As expected, sintering samples containing some (2-4)% ZrO₂ allowed a substantial increase of the considered properties (see reported values on Table I). This is the result of the increase of the vitreous phase quantity that means some diminution of the absorption values. One precise that before being sintered, the samples were compressed at 20 t/cm²; the Zr-doping of the precipitated mullite (that means the increase of the strength values).

Fresnolite and hyalophane appeared because of a little Ti and K amounts in the Na-feldspar. Bazerite is the compound with the chemical formula indicating a double silicate: BaSiO₃-ZrSiO₃. In this material, mullite seems to be (Ba-Na) or (Ba-K) or Ba-doped. The Zr doping of

Table I - Computed amounts of oxides and properties of studied samples. [Table I - Teores de óxidos e propriedades das amostras estudadas.]

N°	Fired 1350 °C	Sintered 1500 °C	Al ₂ O ₃ %	ZrO ₂	BaO %	(Na-K)O ₂	SiO ₂ %	Rest.	$\sigma_{\rm R}$ kgf/cm ²	Abs _c (%)	ρ. 10 ⁷ Ω.cm	R _d kV/mm
1			50.0		7.79	2.65	39.25	0.78	479.31	12.98	5.384	5.85
2-+-			49.94		9.43	2.64	37.12	0.79	404.12	16.83	2.808	6.12
3-++			50.22		9.41	3.16	36.37	0.76	316.65	17.38	0.958	6.62
4+			50.46		7.32	3.15	38.24	0.77	358.08	15.22	4.940	6.17
5+++			64.95	0.0	9,27	2.93	22.50	0.38	256.84	20.25	2.946	6.12
6++-	+	-	64.95		9.30	2.64	24.29	0.37	202.66	16.65	2.303	5.23
7+-+			64.97		7.22	2.95	24.47	0.32	205.04	17.69	3.708	6.08
8+			65.29		7.24	2.66	24.35	0.54	175.45	18.37	6.595	6.58
9 moy			57.63		8.09	2.77	30.89	0.56	281.08	18.10	6.587	5.65
10			40.0	2.0	8.0	4.05	45.75	0.2	/	5.39	187.6	7.87
11	-	+	49.25	4.25	5.35	3.15	38.5	0.25	/	5.25	220.5	8.17
Pure sintered Al ₂ O ₃ [2]									300	0.0	1013	17
Pure sintered ZrO ₂ [2]									180	0.0	10 11	51

XRD qualitative characterization of sample 11 (see Fig. 1) showed seven peaks corresponding (according to their $I = f(2\theta)$ values) to the following combinations:

Compound	Empirical formula	Chemical formula; %
Banalsite	BaNa ₂ Al ₄ SiO ₁₆	23.25 BaO, 9.40 Na ₂ O, 30.92 Al ₂ O ₃ , 36.44 SiO ₂
Bazerite	BaZrSi ₃ O ₉	33.57 BaO, 26.98 ZrO ₂ , 39.46 SiO ₂
Walstromite	BaCa ₂ Si ₃ O ₉	34.4 BaO, 25.16 CaO, 40.44 SiO ₂
Fresnolite	$\mathrm{Ba_2TiSi_2O_8}$	$60.52 \text{ BaO}, 15.76 \text{ TiO}_2, 23.72 \text{ SiO}_2$
Hyalophane	$K_{0.75}Ba_{0.25}Al_{1.75}Si_{2.25}O_{8}$	11.6 K ₂ O, 12.7 BaO, 29.54 Al ₂ O ₃ , 44.76 SiO ₂
Celsian	$\mathrm{BaAl_2SiO_8}$	$40.84 \text{ BaO}, 27.16 \text{ Al}_2\text{O}_3, 32.01 \text{ SiO}_2$
Paracelsian	$\mathrm{BaAl}_{2}\mathrm{SiO}_{8}$	$40.84~{\rm BaO}, 27.16~{\rm Al_2O_3}, 32.01~{\rm SiO_2}$

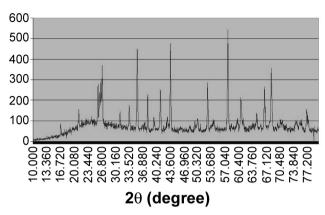


Figure 1: X-ray diffraction pattern of the sample 11. [Figure 1: Difratograma de raios X da amostra 11.]

mullite, that could lead to $Zr_x(SiO_3-Al_2O_4)_y$ did not occur because of the complexity of the studied formulation. One must take in account that oxides are the product of the thermal decomposition of raw materials. Better results could be achieved if using pure oxides (e.g. alumina in place of high Al₂O₂ clay).

SEM analysis showed (see Fig. 2) a structure with a high heterogeneity level. However, this heterogeneous microstructure is representative of all ceramics: black polyedrical grains are amorphous quartz, while black fibbers are grains of mullite dispersed into this vitreous matrix. Porosity of the material is represented in its black areas.



Figure 2: SEM micrograph of the sample. [Figura 2: Micrografia MEV da amostra.]

CONCLUSIONS

One can confirm that the conjugated action of sintering and the addition of zirconia is responsible for the dielectric and mechanical strength improvement. The influence of barium oxide (that partially replaces alkaline oxides) permits a substantial diminution of the material open porosity. This leaded to obtain more amorphous phase in consequence of what we noticed the amelioration of its behavior. An upcoming study of the Zr²⁺ content influence on the studied properties of sintered ceramics containing Ba²⁺, Na⁺ and/or K⁺ oxides must be carried out to envisage a real improvement of their dielectric behavior. Nevertheless, the obtained material is suitable for industrial exploitation because of its high properties (mechanical and dielectric strength).

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