Investigation of Nitrogen-Doping Influence on the Electrocatalytic Activity of Graphene in Alkaline Oxygen Reduction Reaction

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Fuel cell is considered an energetic solution of low environmental impact which large-scale commercialization is hindered by the expensive Pt-based cathodic electrocatalyst. Therefore, its replacement by an active less-expensive metal-free electrocatalyst such as nitrogen-doped graphene (N-graphene) is desirable. To investigate nitrogen-doping influence on graphene's activity in alkaline oxygen reduction reaction (ORR), we prepared N-graphene through thermal treatment of graphene and urea at 800 °C for 30 min. N-graphene exhibited higher onset potential, half-wave potential, diffusion-limited current density and kinetic current density in ORR (0.78 V vs RHE, 0.63 V vs RHE, -3.36 mA cm⁻², -3.00 mA cm⁻², respectively) than undoped graphene (0.73 V vs RHE, 0.62 V vs RHE, -2.80 mA cm⁻², -2.23 mA cm⁻², respectively), which may be attributed to nitrogen-doping, edges and defect sites in N-graphene, as revealed by elemental analysis, scanning electron microscopy and Raman spectroscopy results. This activity improvement also affected ORR's selectivity and rate-determining step.

Keywords: Oxygen reduction reaction, metal-free electrocatalysis, nitrogen-doped graphene, thermal treatment.

1. Introduction

Efficient, low-cost and environmental-friendly energy solutions are imperative because of the expressive emission of greenhouse gases from fossil fuels exploitation and the projected global population growth¹. Acid or alkaline polymer electrolyte membrane fuel cells (PEMFC) supplied with hydrogen are considered an attractive solution owing to the efficient generation of electricity from oxidation-reduction reactions that produce water and heat as the only byproducts.

However, their large-scale commercialization is currently hindered by the expensive platinum-based electrocatalyst used due to ORR's sluggish kinetics, which accounts for about 50% of the PEMFC production cost². Therefore, a current growing research field in ORR electrocatalysis is that of low-cost nature-abundant metal-free electrocatalysts, especially carbon-based ones³.

Graphene is a material formed by two-dimensional sheets of sp²-hybridized carbon atoms⁴ and has attracted much attention from the scientific community because of its astounding thermal and electronic conductivity, large specific surface area and chemical stability^{5,6}. However, it has very limited electrocatalytic activity in the ORR owing to the homogenous spin and charge density distribution over its atomic structure⁷.

Nitrogen-doping has been reported as a successful approach to improve graphene's electrocatalytic activity³. The electronegativity difference between nitrogen and carbon atoms induces spin and charge density asymmetry in graphene's structure, and carbon atoms with highest positive

spin density or largest positive atomic charge are reported as active sites for the ORR^{8,9}.

Chemical Vapor Deposition (CVD) is the most common method to synthesize N-graphene¹⁰. However, it has the disadvantages of elaborate experimental arrangement¹¹, possible contamination of the electrocatalyst with metal impurities¹⁰, explosion hazard due to H₂ and CH₄ gases used in the process¹², use of toxic NH₃ as nitrogen source¹⁰, time-consuming transfer process and high cost¹². Thermal treatment, on the other hand, is a simple and straightforward doping method that does not require toxic precursors. It can be easily scalable and provides flexible control of experimental parameters^{13,14}, consisting in the high temperature heating of a solid mixture of carbon and nitrogen precursors under inert atmosphere¹⁰.

In order to study only the cathodic reaction of the fuel cell, i.e. ORR, an electrochemical half-cell setup with a rotatingdisk electrode (RDE) modified with the electrocatalyst must be used. The electrocatalytic activity can then be evaluated through the reaction's onset potential (E_{onset}), half-wave potential ($E_{1/2}$), diffusion-limited current density (j_1) and electron transfer number (n). For an operational fuel cell, it is desirable for the ORR to be electrocatalysed as close as possible to the equilibrium potential, with minimum energy loss. Therefore, higher E_{onset} and $E_{1/2}$ values are indicators of an effective electrocatalyst^{15,16}. The value of j_1 depends on n, which indicates the reduction pathway selectivity of the electrocatalysed ORR. In alkaline medium, oxygen can be directly or indirectly completely reduced to hydroxide ion or partially reduced to the reactive hydroperoxide ion

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 $(Table 1)^{17}$. For fuel cell application, oxygen complete reduction is desired.

Maouche et al. (2019) used thermal treatment at 900 °C to prepare three-dimensional N-graphene. The E_{onset} , $E_{1/2}$ and j_1 values observed for graphene (0.665 V vs RHE, 0.544 V vs RHE and 0.88 mA cm⁻², respectively) increased after nitrogen-doping (0.879 V vs RHE, 0.808 V vs RHE and 5.21 mA cm⁻², respectively)¹⁸. Kumar et al. (2016) observed a similar behavior for N-graphene prepared by thermal treatment at 800 °C. It showed higher j_1 and *n* values for the ORR (5.08 mA cm⁻² and 3.2, respectively) than undoped graphene (2.70 mA cm⁻² and 2.9, respectively)¹⁹.

In this report, our aim was to study how nitrogen-doping affects graphene's electrocatalytic activity in alkaline ORR. We prepared N-graphene by thermal treatment of graphene and urea at 800 °C. Scanning electron microscopy (SEM) and Raman spectroscopy were used to study nitrogen-doping and heating effects on graphene's structure and morphology, and CHN elemental analysis was used to confirm the presence of nitrogen in N-graphene.

N-graphene's electrocatalytic performance in alkaline ORR was evaluated using a RDE system. Its activity was extensively compared to those of undoped graphene and thermally-treated graphene using the aforementioned electrocatalytic parameters and also the kinetic current density (j_k) and Tafel slope (*b*) values. Therefore, we aim to provide a careful, thorough report of the activity of N-graphene prepared through thermal treatment in alkaline ORR, and comprehensive data that can be readily used for comparison by other works.

2. Experimental

2.1. Materials preparation

Graphene was prepared using the ball-milling technique, which is an efficient high-yielding method. In a Spex high-energy ball mill, 3 g of graphite flakes (99% purity) were mixed with 3 stainless steel balls (diameter of 3 cm) for 12 h at 1024 rpm. N-graphene was prepared through thermal treatment of a solid mixture of graphene and urea. Graphene (100 mg) was ultrasonicated in ultrapure water (200 mL) for 30 min. After urea addition (500 mg), the dispersion was ultrasonicated for 90 min and then dried at 55 °C. The obtained solid mixture was put in an alumina boat inside a tubular furnace and thermally-treated at 800 °C for 30 min (2 °C min⁻¹) under N₂ flow. Pure graphene was submitted to the same heating conditions in the absence of urea to obtain thermally-treated graphene (T-graphene) as a blank sample of the process.

2.2. Characterization

SEM analysis was carried out on a VEGA3 scanning electron microscope (Tescan). Raman spectra of graphene,

T-graphene and N-graphene were recorded on a Confocal T64000 Raman spectrometer (Jobin Yvon) equipped with a 532 nm laser. Graphite's Raman spectrum reproduced here is a part of the RRUFF Project database (deposition N° R050503, http://rruff.info), obtained by Marcus Origlieri on an Almega XR (Thermo) Raman spectrometer equipped with a 532 nm laser at 100% of 150 mW²⁰. CHN elemental analysis was carried out on a vario MAX CHN elemental analyzer (Elementar).

2.3. Electrochemical evaluation

Voltammetry tests were carried out on a WaveDriver 10 potentiostat/galvanostat system (Pine Research) at room temperature using a three-electrode electrochemical cell. The reference electrode used was a reversible hydrogen electrode prepared in the aqueous electrolyte solution (0.1 M KOH) calibrated with respect to the conventional reversible hydrogen electrode (RHE); their potential difference is 0.055 V. A platinum foil was used as counter electrode and an electrocatalyst-modified glassy carbon RDE (0.1964 cm²), as working electrode.

Before each measurement, the electrochemical cell, the reference electrode and the counter electrode were first washed with a solution of 10 wt.% KOH in ethanol, and then, thoroughly washed with ultrapure water. The RDE was polished to a mirror finish using alumina polishing suspensions, and then ultrasonicated for 3 min in isopropyl alcohol and for another 3 min in ultrapure water to remove all residue of alumina. Then it was electrochemically conditioned with cyclic voltammetry (CV) at 100 mV s⁻¹ in the potential range of 0 - 0.6 V vs RHE, for typically 40 cycles or until a stable voltammogram was recorded, rinsed with ultrapure water, dried and mounted on the electrode rotator inverted shaft.

Meanwhile, 1 mg of the electrocatalyst was ultrasonicated in 1 mL of ultrapure water, isopropyl alcohol and 5 wt.% Nafion solution (volume ratio of 79.5:20:0.5, respectively). Then, 30 μ L of the electrocatalyst ink was transferred using a micropipette and spread over the surface of RDE's glassy carbon disk, yielding a electrocatalyst loading of 152.75 μ g cm⁻², and the RDE was rotated at 500 rpm until complete drying of the electrocatalytic film. The rotational drying method was reported elsewhere and yields reproducible, visually homogenous electrocatalytic films²¹.

A potential range of 0 - 1 V vs RHE was used for both CV and linear sweep voltammetry (LSV) subsequent measurements. Before them, the electrolyte solution was purged with N₂ or saturated with O₂ for at least 20 min, and during them, the N₂ or O₂ flux was maintained in the solution, away from the RDE's surface. CV measurements were recorded in N₂ purged solution in the anodic direction with a scan rate of 50 mV s⁻¹. LSV measurements were recorded in the cathodic direction with a scan rate of 10 mV s⁻¹ in

Table 1. Oxygen reduction pathways in alkaline medium.

Reduction pathway	Chemical equation	Standard reduction potential
Direct complete reduction	$O_{2 (g)} + 2 H_2 O_{(l)} + 4 e^- \rightarrow 4 OH^{(aq)}$	0.401 V vs RHE
Indirect complete reduction	$\begin{array}{c} O_{2 (g)} + H_2 O_{(l)} + 2 e \rightarrow HO_2^{-} _{(aq)} + OH^{-} _{(aq)} \\ HO_2^{-} _{(aq)} + H_2 O_{(l)} + 2 e \rightarrow 3 OH^{-} _{(aq)} \end{array}$	-0.065 V vs RHE
Partial reduction	$O_{2 (g)} + H_2O_{(l)} + 2 e^- \rightarrow HO_{2 (aq)}^- + OH^{(aq)}$	0.867 V vs RHE

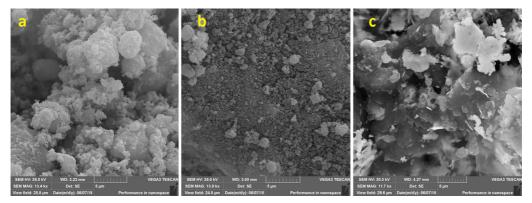


Figure 1. SEM images at 5 µm scale of (a) graphene, (b) T-graphene and (c) N-graphene.

 O_2 and in N_2 saturated electrolyte solution, with different electrode rotation rates (400, 900, 1600 and 2500 rpm). The obtained curves in each of these rates were subtracted in order to correct the background current in the ORR curves.

ORR curves were also iR-drop corrected. The electrolyte resistance value (35.6 Ω on average) was obtained from the high-frequency region of the electrochemical impedance spectra of the electrocatalysts in 0.1 M KOH. An Autolab PGSTAT302N potentiostat/galvanostat system (Metrohm) was used to record these spectra in OCP condition with a potential perturbation of 25 mV within the frequency range of 10 mHz to 100 kHz. Current values were normalized with respect to the RDE geometric surface area (0.1964 cm²).

3. Results and Discussion

The morphology observed for graphene consisted of sphere-like particles with different sizes (Figure 1a), which are most likely resultant from the sliding and curving of graphite layers upon the impact of the stainless steel balls²², and has also been reported by other works for samples obtained through the ball-milling of graphite^{22,23}. As revealed by SEM images, thermal treatment and nitrogen-doping modified graphene's morphology in different ways.

Thermal treatment caused a significant particle size reduction, as observed for T-graphene (Figure 1b). On the other hand, particles with microplatelet structure and defective edges could be observed for N-graphene (Figure 1c). The effectiveness of thermal treatment as a doping process was confirmed by elemental analysis data, which revealed a nitrogen bulk composition of 0.714% in N-graphene.

The main features of graphene's Raman spectrum (D, G and 2D bands) appeared in the Raman spectra of all the prepared electrocataysts (Figure 2). G band is a first order Raman scattering process originated from the E_{2g} in-plane vibration of sp² graphitic carbon atoms^{16,24,25}. D band is assigned to the A_{1g} in-plane breathing vibrational modes of sp² rings and is originated from a second-order scattering process in graphene involving one iTO phonon and one defect^{16,24,26}. Because it is defect-activated, D band indicates the presence of symmetry-breaking perturbations such as edges, non-sixatoms rings, functionalities, doped heteroatoms, sp³ defects, vacancy sites and grain boundaries^{27,29}. 2D band, which is characteristic of graphene structures^{26,30,31}, is a second-order

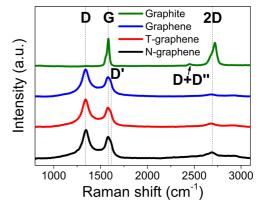


Figure 2. Raman spectra of the electrocatalysts and of graphite.

Table 2. Raman bands positions and $I_{\rm D}/I_{\rm G}$ ratios for the electrocatalysts.

Electropotelyzat	Band position (cm ⁻¹)			1./1
Electrocatalyst	D	G	2D	$I_{\rm D}/I_{\rm G}$
Graphene	1341.52	1579.80	2706.60	1.70
T-graphene	1340.78	1580.57	2694.13	1.82
N-graphene	1343.34	1582.51	2692.61	1.84

Raman scattering process as well, but involves two iTO phonons near the K point and is not defect-activated^{24,32,33}.

For graphite, the almost imperceptible D band indicates low density of defect in the material, and its feeble D+D" band at approximately 2452 cm⁻¹ results from a combination of a D phonon and an acoustic longitudinal phonon, D"³⁴. The 2D band for graphite is asymmetric, whether for graphene it is symmetric, and consists of two components, because the increase in graphene layer number causes a split of the electronic band structure³⁴. The D' band that appears as a shoulder of the G band for graphene, T-graphene and N-graphene indicates the presence of few-layered graphene^{22,35}.

These different features between the Raman spectra of graphene and graphite (Figure 2), e.g. the increased intensity of the D band, the more symmetric shape of the 2D band, the broadening of the G band and the appearance of the D' band, indicate that graphene flakes were successfully obtained by the ball-milling of graphite in this work. This same evolution of Raman spectral features was also reported in other works for the mechanochemical exfoliation of graphite to obtain graphene^{22,35}.

The G and D bands of N-graphene were shifted to higher frequencies compared to the bands of graphene (Table 2). Since chemical doping changes the electronic state of graphene and breaks the symmetry of its sp² carbon network³³, D and G bands upshift is expected for doped-graphene regardless of the type of doping^{13,33}. Therefore, the observed Raman spectrum of N-graphene corroborates the elemental analysis to confirm the effectiveness of the thermal treatment to prepare this electrocatalyst.

The ratio of D and G bands intensity (I_D/I_G) is commonly used to evaluate the degree of structural disorder in graphene materials^{13,25}. Thermal treatment and nitrogen-doping introduced defects in the structure of graphene and decreased its graphitization degree, as indicated by the higher I_D/I_G values of T-graphene and N-graphene compared to that of graphene (Table 2). This is most likely due to edges exposure when oxygen moieties are removed during heating^{13,28} and defects originated from the substitutional doping of nitrogen atoms in the framework of graphene^{13,14,25}.

The cyclic voltammetry profile of the electrocatalysts was typical of graphene-based materials – featureless semirectangular waves (Figure 3). Regarding ORR electrocatalysis, none of the electrocatalysts investigated in this work exhibited comparable activity to Pt/C 20 wt.% (Figure 4). Despite that, nitrogen-doping enhanced graphene's activity (Figure 4), as evidenced by the higher E_{onset} , $E_{1/2}$ and j_1 values registered for N-graphene, compared to the undoped electrocatalysts (Figure 4 and Table 3).

An ORR diffusion-limited current density plateau, commonly observed for platinum-based electrocatalysts³⁶, was not obtained for our electrocatalysts (Figure 4). This behavior has also been reported for other metal-free electrocatalysts^{30,32}. Therefore, j_1 values were sampled in the middle of the diffusion control potential range (0.05 – 0.55 V vs RHE), at 0.3 V vs RHE (Table 3). Graphene and T-graphene ORR activities were very similar (Figure 4 and Table 3), indicating that, in the case of graphene prepared through the ball-milling method, thermal treatment does not enhance its activity in the ORR.

ORR curves were recorded at different electrode rotation rates for graphene (Figure 5), T-graphene (Figure 6) and N-graphene (Figure 7) in order to calculate *n* values from their Koutecký-Levich plots (Figure 8). For each electrocatalyst, current density values were sampled in each rotation rate at 0.63 V vs RHE and at 0.3 V vs RHE, these potential values are in the mixed kinetic-diffusion control (0.55 - 0.70 V vs RHE) and in the diffusion control potential ranges (0.05 - 0.55 V vs RHE), respectively.

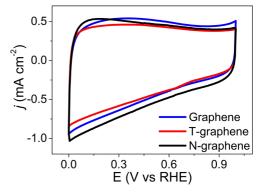


Figure 3. Cyclic voltammograms of the electrocatalysts in N_2 saturated 0.1 M KOH at 50 mV s⁻¹.

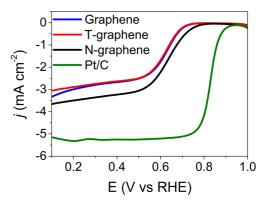


Figure 4. ORR curves of the electrocatalysts and of Pt/C 20 wt.% (Pt loading of 25.46 μg_{Pt} cm⁻²) in O₂ saturated 0.1 M KOH at 10 mV s⁻¹ and 1600 rpm.

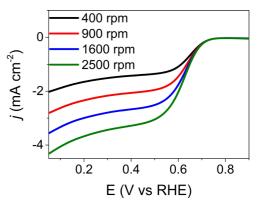


Figure 5. ORR curves in O_2 saturated 0.1 M KOH at 10 mV s⁻¹ and different electrode rotation rates for graphene.

Table 3. ORR activity parameters extracted from Figures 4, 8 and 9 for the electrocatalysts.

		Graphene	T-graphene	N-graphene
E	onset (V vs RHE)	0.73	0.72	0.78
F	E _{1/2} (V vs RHE)	0.62	0.62	0.63
	$j_1 ({\rm mA \ cm^{-2}})$	-2.80	-2.75	-3.36
	$j_{\rm k}$ (mA cm ⁻²)	-2.23	-2.17	-3.00
	0.63 V vs RHE	2.1	1.9	2.9
n at ().3 V vs RHE	2.3	2.2	2.9
	$b \text{ (mV dec}^{-1})$	72.27	70.68	98.11

The slope of the line for each potential value in the Koutecký-Levich plots of the electrocatalysts was used to estimate the *n* value through the Koutecký-Levich equation (Equation 1), where *j* is the measured current density, j_1 is the diffusion-limited current density, j_k is the kinetic current density and ω is the electrode rotation rate. *B* factor is defined by Equation 2, where *F* is the Faraday constant (96,485 C mol⁻¹), D_0 is the diffusion coefficient of oxygen in the electrolyte (1.9 10⁻⁵ cm² s⁻¹), *v* is the kinematic viscosity of the electrolyte (0.01 cm² s⁻¹), C_0 is the

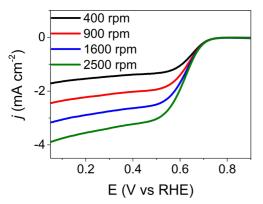


Figure 6. ORR curves in O_2 saturated 0.1 M KOH at 10 mV s⁻¹ and different electrode rotation rates for T-graphene.

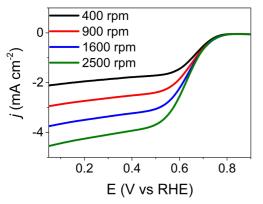


Figure 7. ORR curves in O_2 saturated 0.1 M KOH at 10 mV s⁻¹ and different electrode rotation rates for N-graphene.

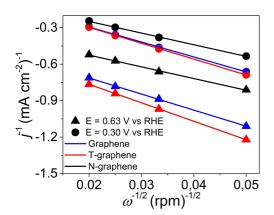


Figure 8. Koutecký-Levich plots of the electrocatalysts.

solubility of oxygen in the electrolyte (1.2 10 ⁻⁶ mol cm⁻³) and 0.2 is a constant used when the electrode rotation rate is expressed in rpm¹³.

$$\frac{1}{j} = \frac{1}{j_1} + \frac{1}{j_k} = \frac{1}{B\omega^{1/2}} + \frac{1}{j_k}$$
(1)

$$B = 0.2nFD_0^{2/3} v^{-1/6} C_0$$
(2)

Nitrogen-doping increased graphene's n value for the ORR (Table 3). Graphene and T-graphene lead ORR through the 2 electron reduction pathway, i.e. the reduction of O₂ molecule to HOO⁻, in the entire monitored potential range (Table 3). After nitrogen-doping, n value increased to 3 (Table 3), indicating that N-graphene promotes oxygen partial and complete reduction pathways simultaneously, with possible formation of HOO⁻ and OH⁻ species. This combined 2 and 4 electrons reduction process has also been reported in other works for nitrogen-doped carbon-based electrocatalysts^{28,37-39}.

Our result is similar to that reported by Kim et al. (2020) for a nitrogen-doped reduced graphene oxide electrocatalyst at 1.0 mA cm⁻², that was prepared by the thermal treatment of a solid mixture of urea and graphene oxide (5:1 mass ratio) at 800 °C for 30 min under N₂ atmosphere (almost the same conditions used in our work, except for the carbon precursor and preparation procedure of the solid mixture prior to thermal treatment)⁴⁰. Kim et al. (2020) found out that the partial and complete reduction pathways in their electrocatalyst were completely independent from each other and suggested that more than one type of site or structure in the electrocatalyst were active in the ORR.

Nitrogen-doping also increased graphene's j_k value in the ORR (Table 3), another parameter used to evaluate the electrocatalytic activity¹⁵. Values of j_k were extracted through the intercept of each electrocatalyst's Koutecký-Levich plot at 0.63 V vs RHE, which according to the Koutecký-Levich equation, corresponds to an infinite value of ω where current density is no longer affected by mass transport limitations⁴¹. For this reason, Shin et al. (2015) consider j_k a better parameter to judge ORR activity than j_i , which is dependent on the oxygen concentration in the electrolyte and, therefore, can be affected by different pressure and temperature conditions⁴².

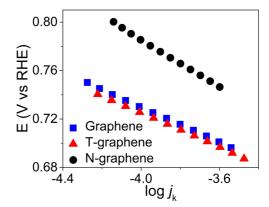


Figure 9. Tafel plots of the electrocatalysts at 1600 rpm.

Tafel plots were used to gain further understanding regarding the electrocatalysts activity in the ORR (Figure 9). Tafel relation is the linear relation between $\log j$ and potential, since the dependence of j on potential is exponential⁴³. This relation can only be observed for irreversible reactions of sluggish kinetics that require significant activation overpotentials⁴⁴, such as ORR. The values of b extracted from these plots in the kinetic control potential range (0.68 – 0.80 V vs RHE, for our electrocatalysts), where there is no mass-transport limitation for j ⁴⁵, can be used to indicate the ORR ratedetermining step (rds) for each electrocatalyst^{46,47}.

For ORR, two *b* values are commonly observed: 60 mV dec⁻¹ or 120 mV dec⁻¹, indicating that the rds is either a chemical or an electrochemical step, respectively⁴⁶. Considering the associative mechanism of the electrocatalyzed ORR, this chemical step would be the reaction of adsorbed O_2^- species with H₂O to form OH⁻ and adsorbed OOH, and the electrochemical step would be the first electron transfer step to adsorbed O₂ to form O₂⁻ species, if the Tafel slope was obtained at low overpotentials⁴⁷. These values may vary depending on the electrode material and the potential range⁴⁸. In this work, *b* values obtained for graphene and T-graphene (Table 3) indicated that a chemical step is most likely the rds. N-graphene's *b* value (Table 3), on the contrary, indicated that an electrochemical step is most likely the rds. Therefore, nitrogen-doping affected the ORR mechanism.

Similar *b* values to the ones we found have been reported for other electrocatalysts. For example, Chai et al. (2019) reported a *b* value of 75.4 mV dec⁻¹ for a solid carbon spheres electrocatalyst⁴⁹. Ramirez-Barria et al. (2019) prepared a nitrogen-doped reduced graphene oxide electrocatalyst through the thermal annealing of reduced graphene oxide in ammonia-containing atmosphere at 700 °C, which was synthesized from a modified Brodie's method using 100 mesh graphite powder³⁹. This electrocatalyst's *b* value was 99 mV dec⁻¹. Miao et al. (2017) prepared nitrogendoped reduced graphene oxide electrocatalysts through a hydrothermal method and posterior thermal treatment at 600 and at 700 °C, which exhibited *b* values of 95.3 and 94.3 mV dec⁻¹, respectively³⁸. All these experiments were conducted in O₂ saturated 0.1 M KOH electrolyte solution.

The elemental analysis indicated the presence of nitrogen in N-graphene electrocatalyst after thermal treatment of graphene with urea at 800 °C. The enhanced activity of N-graphene in the ORR, compared to the undoped electrocatalysts, might be ascribed to the incorporation of nitrogen atoms in the framework of graphene. As explained by Zhang and Xia (2011), this incorporation changes the atomic charge and spin density distribution of N-graphene, and leads to the formation of active sites, carbon atoms with highest spin density or largest positive atomic charge⁸.

N-graphene SEM image revealed a morphology enriched with exposed edges, which are reported as highly active ORR sites⁵⁰⁻⁵². The edges exposure and the exfoliated structure may facilitate the accessibility of O_2 molecules to the active sites. The I_D/I_G ratio obtained from the Raman spectrum of N-graphene indicated a highly defective structure. This defective structure might be associated with N-graphene's higher activity in the ORR, compared to graphene and T-graphene because, as explained by Ortiz-Medina et al. (2019), structural defects modify atomic orbitals and create localized electronic states that enhance electrocatalytic activity⁵³.

In our work, most likely a synergistic effect of nitrogen incorporation and edge and defect sites may have contributed to N-graphene's activity in the ORR, as observed by its higher E_{onset} , $E_{1/2}$, j_1 and j_k values compared to undoped graphene and the blank sample, T-graphene. This improved activity also altered the oxygen reduction pathway selectivity and the rds of the ORR electrocatalysed by N-graphene.

4. Conclusions

Nitrogen-doping improved graphene's activity in alkaline ORR and altered its oxygen reduction pathway selectivity and rds. This improvement did not originate from heat treatment only, since T-graphene's activity in ORR was similar to that of graphene, but it is likely that a combined influence of nitrogen-doping, edges and defect sites simultaneously contributed to the reported activity of N-graphene. Future investigations should focus on the optimization of thermal treatment conditions in order to achieve a N-graphene electrocatalyst with higher activity in the ORR.

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