

Cellulose Nanocrystals and Jute Fiber-reinforced Natural Rubber Composites: Cure Characteristics and Mechanical Properties

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A comparative study was conducted for three natural rubber (NR) composites: one containing raw jute fibers, one containing bleached jute fibers, and one containing cellulose nanocrystals (CNCs). The composites were prepared by adding each filler at 5 or 10 phr to the latex before chemical coagulation, and the composites were processed in a two-roll mixing mill. The mechanical, thermal, and vulcanization properties of the prepared composites were then determined. The NR stiffness increased with increasing fiber content, particularly when bleached fibers were added, as evidenced by the increased elastic moduli, maximum torques, glass transition temperature, and crosslinking density. Scanning electron microscopy showed agglomerates in the composite with 10 phr CNCs, which led to poorer mechanical performance compared to that of the bleached fiber-reinforced composite (10 phr). On the other hand, the addition of CNCs to the NR increased the cure rate index, thereby significantly advancing the optimum vulcanization time.

Keywords: *Natural rubber, jute fibers, cellulose nanocrystals, mechanical properties, crosslink density.*

1. Introduction

Natural rubber (NR) is the only elastomer that is extracted from a renewable source (*Hevea brasiliensis*)¹. It is an important elastomer, with interesting properties compared to synthetic rubbers, such as high mechanical strength, elasticity, flexibility, resilience, and abrasion resistance^{2,3}. However, these properties are obtained after vulcanization, during which crosslinks are formed between the polymer chains⁴, generally using sulfur as the vulcanizing agent. The mechanical properties of NR can also be improved and adjusted by adding reinforcing fillers.

To develop sustainable composite materials⁵, fillers extracted from renewable sources, such as natural fibers, can be used⁶. These have several advantages over commercial synthetic fillers^{7,8} (low cost, biodegradability, and attractive mechanical properties such as tensile strengths comparable to those of carbon and glass fibers). Jute fiber is a low-cost commercially available natural fiber; it is abundant in nature and has a high cellulose content^{9,10}. Studies^{9,11} have shown that the addition of jute fibers to NR helps improve the modulus of elasticity and hardness of NR. However, to apply lignocellulosic fibers as a reinforcement in polymeric composites, it is desirable to remove lignin and other amorphous materials from these fibers, mainly to allow the separation of their fibrils^{10,12}. The most common procedures for this include mercerization using a sodium hydroxide solution^{8,12} and bleaching using sodium chlorite¹³; these treatments can be applied together. Mercerized and/or bleached fibers can be applied as a reinforcement for different types of polymer matrices¹⁴⁻¹⁶, thus increasing the modulus of elasticity by at least 20%.

Studies are actively being conducted on the extraction of crystalline regions of cellulose, forming nanometric particles, also called cellulose nanocrystals (CNCs)⁵, which have a high surface area associated with good mechanical properties, low density, and biodegradability^{13,17}. The addition of CNCs can help increase the tensile strength and elastic modulus of vulcanized NR; however, this drastically decreases the elongation^{4,18}. Few studies^{17,19} analyzed the vulcanization process of NR with the addition of CNCs and reported increased crosslink density.

In this work, three NR composites, one containing raw jute fibers, one containing bleached jute fibers, and one containing CNCs, were prepared with different filler concentrations (5 and 10 phr). The tensile strength and hardness of the prepared composites were measured, and their morphology, vulcanization characteristics, and thermal behavior were then evaluated.

2. Methods and Materials

2.1. Materials

A centrifuged NR latex with 60% dry rubber and a high ammonia content was purchased from Du Latex (SP, Brazil); the latex is originally from Malaysia. Raw jute fibers were supplied by Castanhal Textile Company (PA, Brazil).

2.2. Bleaching of jute fibers

The bleaching procedure is in accordance with a previous work²⁰. The jute fibers were milled using a mechanical milling machine with a 0.5 mm sieve opening. The fibers (30 g) were dispersed in deionized water at 70 °C and stirred for 3 h.

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The fiber bleaching procedure was based on the TAPPI T19m-54 standard method, as described by Trindade et al.²¹. After washing and filtering, the fibers were dispersed in deionized water at 70 °C, and sodium chlorite (25 g) and glacial acetic acid (10 mL) were added to the aqueous suspension. The system was vigorously stirred for 1 h. This procedure was repeated twice, i.e., sodium chlorite and glacial acetic acid were added again after a 1 h interval. After the final addition of the reagents, the system was stirred for 3 h. The total reaction time was 5 h. After washing and filtering, the fibers were treated with a 2% NaOH (w/w) solution, stirred at 80 °C for 2 h, filtered until a pH of 7 was obtained, and finally, dried in a vacuum oven at 60 °C for at least 24 h. For the subsequent preparation of the NR composites with bleached jute fibers, they were kept wet, and the dry fiber mass was calculated via three aliquots.

2.3. Extraction of cellulose nanocrystals (CNCs)

The CNCs were extracted by acid hydrolysis with sulfuric acid, and the reaction conditions were based on the most discussed parameters in literature²²⁻²⁴. Approximately 10 g of bleached jute fibers were dispersed in 200 ml of 50% (w/w) sulfuric acid (H₂SO₄). The mixture was continuously stirred at 300 rpm for 75 min at 50 °C. The resulting suspension was cooled in an ice bath and diluted four times with deionized water. The neutralization step involved four centrifugation cycles at 15,000 rpm and 10 °C for 5 min, with the addition of deionized water. Thereafter, the resulting suspension was subjected to dialysis using a semipermeable cellulose membrane in deionized water until the pH of the suspension remained in the range of 6-7. Subsequently, the neutralized suspension was subjected to ultrasonication for 30 min. Finally, the suspension was stored in a refrigerated form.

2.4. Preparation of NR composites

The NR composites were prepared by mixing an aqueous dispersion (approximately 2 wt.%) of the different fillers (raw jute fibers, bleached fibers, and CNCs) with the NR latex (60 wt.%) such that the filler concentrations reached the desired values (5 and 10 phr). The mixtures were then dispersed using an Ultra Turrax disperser (IKA) at 10,000 rpm for 5 min. The mixtures were coagulated by adding a sufficient amount of acetic acid (3 mol/L) to small latex portions until the solidification of the rubber was complete. Next, the rubbers were immersed in deionized water for five days for neutralization, and the rubber was then oven dried for 72 h at 60 °C. The rubber mixtures were prepared in a laboratory two-roll mixing mill. First, the NR was masticated for 5 min, and the components, listed in Table 1, were then added in the order presented, with a mixing time of 5 min for each material. The samples were vulcanized at 150 °C in a heated hydraulic press at 20 MPa for their respective vulcanization times (t_{90}). The values were determined using an oscillating disc rheometer ODR 2000 (TEAM, Brazil) based on ASTM D-2084-11.

Table 1. Basic formulation of natural rubber composites.

Materials	Parts
Natural rubber	100
Jute fibers (raw or bleached)	0–10
CNCs	
Plasticizing oil ^a	1.5
Antioxidant oil ^b	1
Zinc Oxide	5
Stearic acid	2
CBS ^c	0.8
ZBEC ^d	0.15
Sulfur	2

^aParaffinic oil; ^b Styrene phenol; ^c N-cyclohexyl-2-benzothiazyl sulfenamide; ^d Zinc dibenzylthiocarbamate.

2.5. Methods

To observe the surface morphology of the raw jute fibers and bleached fibers, the samples were coated with a thin layer of conductive material (gold) and observed using a field-emission scanning electron microscope (FEI, INSPECT F50 SEM-FEG) at an applied accelerating voltage of 15 kV. The fiber diameter measurements were performed using ImageJ software, with at least fifty measures to obtain an average of the values. Subsequently, the NR composites were cryogenically fractured and analyzed using SEM-FEG with the same conditions previously mentioned.

The X-ray diffraction (XRD) analysis of the raw jute fibers, bleached fibers, and CNCs were performed in an XPERT-MPD diffractometer, with Cu-K α radiation ($\lambda = 1.544 \text{ \AA}$). The scattered radiation was detected in a 2θ range of 5 to 40° at a scanning rate of 0.02 °/min. The samples were dried in a vacuum oven and powdered using a high-energy ball mill (attritor mill). The crystallinity index (CI) was estimated by subtracting the amorphous halo²⁵ of the XRD curves by creating a baseline using the Origin 8.0 software. The CI was calculated using Equation 1, where CI expresses the relative degree of crystallinity, A_t is the total area of the XRD curves, and A_c is the sum of the peak areas corresponding to the crystalline structure of cellulose, calculated after the baseline correction.

$$CI(\%) = \frac{A_c}{A_t} \times 100 \quad (1)$$

The CNC samples were observed in a transmission electron microscope (JEOL-1010). An acceleration voltage of 80 kV was used. A drop of CNC aqueous suspension diluted to 0.4 mg/ml was deposited on a copper grid covered with a carbon film and dried in an oven for 2 h at 60 °C. The diameter and length of the CNCs were measured using the ImageJ software with at least fifty measures.

The sample hardness was determined using a Shore A-type durometer. Compound samples (6-mm thick) were used to carry out the measurements. The tensile tests were

performed based on the ASTM D412-15a standard on a universal testing machine (Kratos, DEK model) fitted with a contact-style extensometer. The crosshead speed was 500 mm/min, and a 500 N load cell was used.

A dynamic mechanical thermal analysis (DMTA) of the NR composites was performed on a DMTA instrument (DMA 8000, PerkinElmer) using rectangular samples with dimensions of 12 mm × 10 mm × 2.5 mm. The experiment was carried out at a frequency of 1 Hz, at temperatures ranging from −100 to −12 °C, at a heating rate of 2 °C/min, and for an oscillation amplitude of 50 μm.

The crosslink density (ν) values were determined by swelling method using Equation 2^{26,27}, where M_c is the molecular weight between the crosslinks, which was calculated using Equation 3^{26,27}, where v_n is the molar volume of the solvent; ρ_p is the density of the polymer; and V_{rf} is the volume fraction of the rubber phase in the swollen gel of the rubber vulcanizate.

$$\nu = \frac{1}{2M_c} \quad (2)$$

where

$$M_c = \frac{-\rho_p \nu_n V_{rf}^{1/3}}{[\ln(1 - V_{rf}) + V_{rf} + \chi V_{rf}^2]} \quad (3)$$

χ is the interaction parameter between the polymer and the solvent and is expressed in Equation 4^{26,27}, where β is the lattice constant; δ_p and δ_s are the solubility parameters of the polymer and solvent, respectively; R is the gas constant; and T is the absolute temperature.

$$\chi = \beta + \frac{\nu_n}{RT} (\delta_s - \delta_p)^2 \quad (4)$$

The volume fraction of the rubber phase in the swollen gel of the filled rubber vulcanizate (V_{rf}) was calculated by considering only the fraction of the rubber compounds, i.e., by subtracting the fiber amount using the empirical density of each filler, as described by Jacob et. al.²⁸, using Equation 5, where d is the de-swollen weight of the sample; f is the volume fraction of the fiber; w is the initial weight of the sample, ρ_s is the density of the solvent, and A_s is the amount of solvent absorbed by the sample.

$$V_{rf} = \frac{(d - fw)\rho_p^{-1}}{(d - fw)\rho_p^{-1} + A_s\rho_s^{-1}} \quad (5)$$

The swelling of the NR composites was carried out in toluene for 72 h at room temperature, using square samples with a width of 10 mm and a thickness of 2 mm.

3. Results and Discussion

3.1. Fiber characterization

Figure 1 shows the SEM images of the jute fibers before and after the bleaching process. The surfaces of the

bleached fibers are softer and more uniform with fewer solid aggregates. In addition, the bleached fibers have a structure with individual microfibrils and are more dispersed than the raw fibers; this may indicate that the treatment was able to remove a significant amount of amorphous mass lying between the surfaces of the microfibrils, such as lignin, impurities, and low molar mass polysaccharides²⁹.

Figure 2 shows the TEM image obtained from the CNCs, which presents individual nanocrystals with a fine needle-like shape. These particles exhibit a certain agglomeration, which may be due to the drying process during the preparation analysis. The average diameter and length of the CNCs were measured. Table 2 lists the results. A comparison of the results shows that, as expected, the acid hydrolysis process helped reduce the size of the fibers, reaching an average diameter of 16 ± 4 nm and a length of 414 ± 72 nm, consistent with those reported in literature for cellulose nanocrystals^{30,31}.

Figure 3 shows the XRD curves for the jute fibers before and after the bleaching treatment and for the CNCs. The highest intensity peak at approximately $2\theta = 22.3^\circ$ and the lowest intensity diffraction peaks at $2\theta = 15.8^\circ$ and $2\theta = 34.5^\circ$, which was more pronounced for CNCs, indicated the presence of type I cellulose, and were assigned to plane (200), overlapping of (1 $\bar{1}$ 0)/(110), and (004) crystallographic planes, respectively^{22,23}. The increase in the peak intensity after the bleaching and acid hydrolysis of the jute fibers indicates an increase in the crystallinity after applying the treatments. The crystallinity indices for the raw jute fibers, bleached fibers, and CNCs were 56, 59, and 75%, respectively, indicating an increase in the crystallinity of the jute fibers after the bleaching process and a further increase after acid hydrolysis, which indicates the removal of amorphous regions by the treatments performed¹⁹.

3.2. Composite characterization

3.2.1. Cure characteristics

Figure 4 shows the rheometric curves of the NR composites. Table 3 lists the curing characteristics of the NR composites, expressed in terms of the maximum and minimum torques (M_H and M_L , respectively), scorch time (t_{s2}), optimum cure time (t_{90}), delta torque (ΔM), and cure rate index (CRI).

The addition of the fibers to the NR increased the maximum torque values of the composites. A comparison of the composites with the same filler content shows that the effect was more pronounced in composites with bleached jute fibers. This indicates an increase in the stiffness of the vulcanized composites because of the restriction in the mobility of the rubber chains due to the addition of fibers³². For composites with raw and bleached jute fibers, there was no significant difference in the minimum torque compared to pure NR. The addition of these fibers did not hinder the processing of the NR in the two-roll mixing mill. In the case of the addition of the CNCs, there was a slight decrease

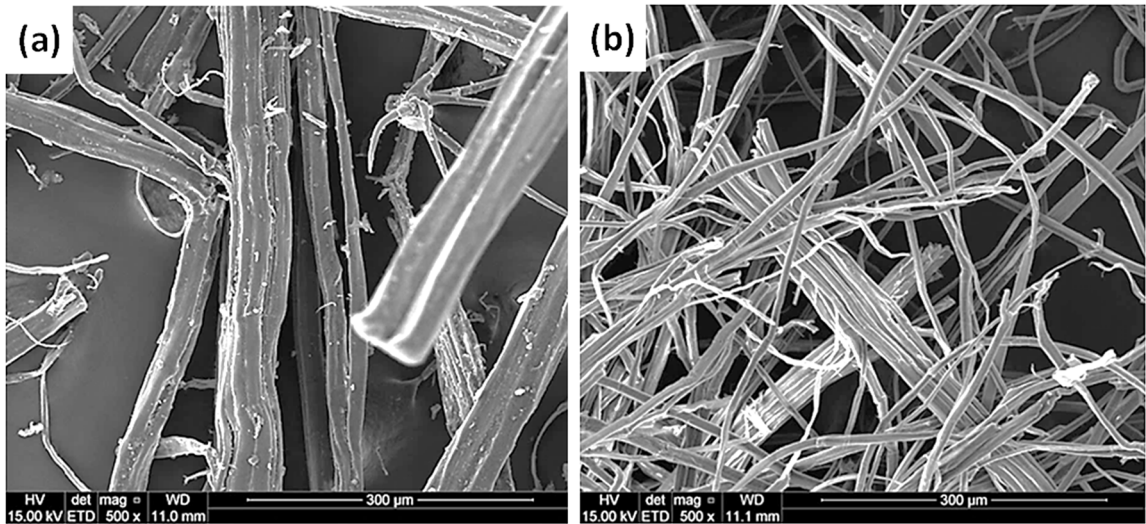


Figure 1. SEM micrographs ($\times 500$) of (a) raw and (b) bleached jute fibers.

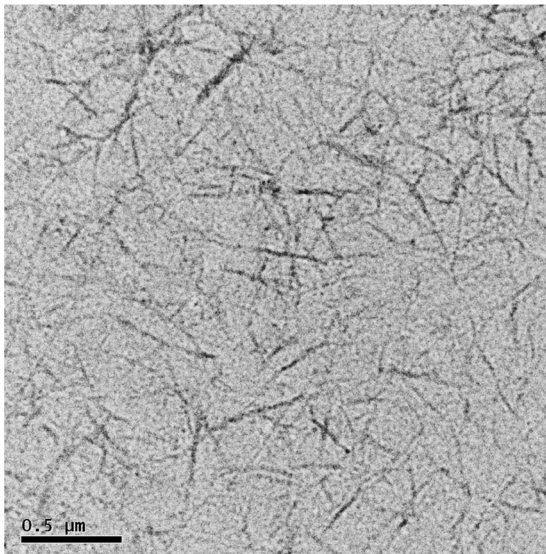


Figure 2. TEM image of dried dispersion of CNCs.

Table 2. Diameter, length, and aspect ratio for raw jute fibers (JF), bleached jute fibers (BJF), and CNCs.

Sample	Diameter	Length (nm)	Aspect Ratio
FJ*	$22 \pm 10 \mu\text{m}$	-	-
BJF*	$7 \pm 2 \mu\text{m}$	-	-
CNC	$16 \pm 4 \text{ nm}$	414 ± 72	26 ± 6

*measured from SEM images

in M_L . This may indicate that the CNCs decreased the viscosity of the NR, favoring its processing in the two-roll mixing mill.

The addition of raw and bleached fibers to the NR reduced the optimum cure time of the composites when compared to the NR, and a higher reduction in t_{90} and therefore a higher cure rate index was observed for the NR composites containing CNCs.

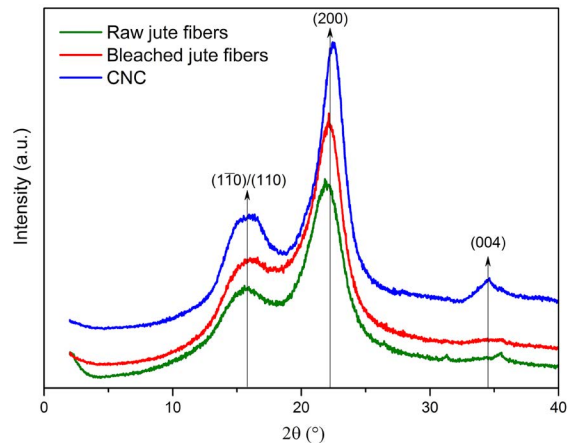


Figure 3. X-ray diffraction curves for raw, bleached jute fibers, and CNCs.

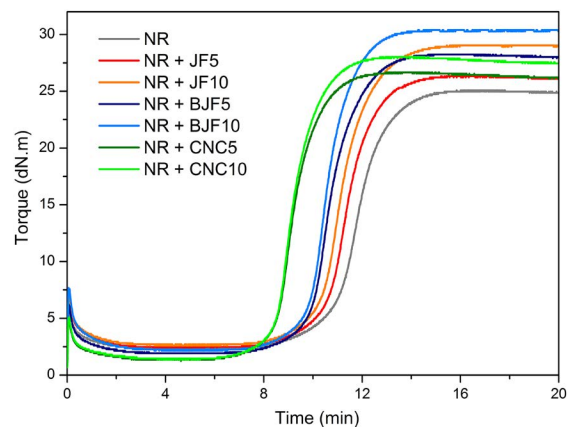


Figure 4. Rheometric curves for the three NR composites containing raw jute fibers (JF), bleached jute fibers (BJF), and CNCs.

Table 3. Vulcanizing behavior of NR and NR composites at 150 °C

Sample	M_L (dN.m)	M_H (dN.m)	ts_2 (min)	t_{90} (min)	$\Delta M'$ (dN.m)	CRI**(min^{-1})
NR	2.3	25.1	10.16	13.33	22.8	31.6
NR + JF 5	2.4	26.3	9.92	12.83	23.9	34.4
NR + JF 10	2.6	29.0	9.73	12.80	26.4	32.6
NR + BJF 5	1.9	28.2	9.43	12.27	26.3	35.2
NR + BJF 10	2.1	30.4	9.34	11.98	28.3	37.9
NR + CNC 5	1.2	26.7	8.19	10.60	25.5	41.5
NR + CNC 10	1.4	28.0	8.22	10.62	26.6	41.7

$$*\Delta M = M_H - M_L$$

$$**\text{CRI} = \text{Cure rate index} = 100/(t_{90} - ts_2)$$

Other studies^{32,33} have reported similar results for composites based on NR and natural fibers. However, for NR composites containing CNCs, a study¹⁷ showed that the addition of CNCs can cause a delay in the vulcanization time, as CNCs have acidic groups on their surface³⁴ (when obtained by acid hydrolysis with H_2SO_4), which do not favor the vulcanization reaction¹⁷. In contrast, in the case of this work, the addition of the fibers, particularly those that were chemically treated, favored the vulcanization reaction. In fact, Thomas et al¹⁹ showed that cellulose can form a chemical structure with zinc from ZnO, contributing to the formation of the crosslinked network. Thus, the smaller size of the CNCs and consequently the larger surface area, compared with the other fillers, may have contributed to the further decrease in t_{90} of the NR composites.

3.2.2. Hardness

Table 4 lists the hardness values of the NR composites. As expected, the hardness of the composites increased with the increase in the fiber content. Studies^{11,35} reported that the addition of fibers helps in significantly increasing the hardness of NR composites. This effect was more pronounced in composites with bleached fibers. For the composites containing CNCs, the addition of this filler caused a slight increase in the hardness, consistent with other composites based on rubber and nanometric fillers³⁶.

3.2.3. Tensile strength

Table 4 lists the mechanical properties of the NR composites. A slight reduction in the tensile strength values and elongation at break with the addition of fillers was observed. Other works^{11,32} reported a similar behavior when natural fibers were added to NR, whereas the addition of CNCs^{4,37} led to an increase in the tensile strength with a significant reduction in the elongation.

In this work, only the elastic modulus at 100 and 300% elongation (M_{100} and M_{300}) increased with the addition of fibers and CNCs, compared to pure NR. This behavior was more pronounced in the case of bleached jute fibers. This effect is similar to that reported by other authors^{11,19,33}, who concluded that the use of natural fibers as a reinforcement in NR matrices leads to an increase in the modulus of elasticity because of the strong interactions between the surface of the fibers and the matrix.

3.2.4. Dynamic mechanical thermal analysis (DMTA)

The glass transition temperature (T_g) of the NR composites was determined using the tan delta peak position (Table 5). The addition of fibers to the NR slightly increased the T_g values, which can be related to the restriction imposed by the fibers or CNCs on the mobility of the polymer chains.

Table 4. Mechanical properties of NR and NR composites.

Sample	Tensile Strength (MPa)	Elongation at break (%)	M_{100} (MPa)	M_{300} (MPa)	Hardness (Shore-A)
NR	22.3 ± 0.8	687 ± 12	0.78 ± 0.02	1.82 ± 0.04	34.4 ± 0.3
NR + JF5	21.0 ± 0.7	640 ± 22	1.06 ± 0.04	2.18 ± 0.09	38.2 ± 0.1
NR + JF10	20.1 ± 0.5	618 ± 25	1.25 ± 0.03	2.47 ± 0.07	41.6 ± 0.3
NR + BJF5	20.2 ± 1.1	649 ± 28	1.41 ± 0.07	2.55 ± 0.09	39.3 ± 0.4
NR + BJF10	20.5 ± 0.8	653 ± 19	2.12 ± 0.06	3.42 ± 0.05	47.8 ± 0.2
NR + CNC5	21.8 ± 0.7	649 ± 14	0.92 ± 0.02	2.63 ± 0.08	36.4 ± 0.4
NR + CNC10	21.3 ± 0.7	652 ± 10	0.98 ± 0.01	2.51 ± 0.05	37.1 ± 0.3

Table 5. Glass transition temperature of the NR and NR composites.

Sample	tan delta (°C)
NR	-50
NR + JF5	-48
NR + JF10	-49
NR + BJF5	-49
NR + BJF10	-48
NR + CNC5	-48
NR + CNC10	-48

Table 6. Volume fraction of the rubber phase in the swollen gel of the rubber vulcanizate (V_{rf}) and crosslinking density of the NR and NR composites.

Sample	V_{rf}	Crosslink density $\times 10^{-5}$ (gmol/cm ³)
NR	0.2003	4.16
NR + JF5	0.1978	4.04
NR + JF10	0.1980	4.05
NR + BJF5	0.2015	4.22
NR + BJF10	0.2070	4.51
NR + CNC5	0.2051	4.41
NR + CNC10	0.2108	4.72

3.2.5. Crosslink Density

Table 6 lists the values of the volume fraction of the rubber phase in the swollen gel of the filled rubber vulcanizate (V_{rf}) and the crosslink density of the NR composites. The crosslink density slightly decreased with the addition of raw jute fibers to the NR. On the other hand, the addition of bleached fibers and CNCs to the NR increased the values of V_{rf} and consequently, the crosslink density. According to a study²⁸, as the amount of filler added to the NR increases, the amount of solvent absorbed by the sample decreases, thus increasing the V_{rf} values and in turn increasing the crosslink density. Other authors^{17,26,38} reported similar results for rubber matrix composites and attributed this behavior to the interfacial interaction between the matrix and the filler^{17,19}. Strong matrix-filler interactions can hinder solvent permeation in the rubber matrix, reducing the volume of the solvent absorbed after the swelling test¹⁷. The increase in the stiffness of the NR with the addition of fibers or CNCs (confirmed by the increase in the elastic modulus), the increase in the glass transition temperature, and the increase

in the maximum torque can be attributed to the reduction in the mobility of the rubber molecules. This behavior can be attributed to the strong matrix-filler interaction and increased crosslink density.

However, the increase in the stiffness of CNC-added samples is not as significant as that of the bleached fiber-added samples. In the former case, the reduction in the amount of solvent absorbed by the sample decreased not only because of the strong polymer-solvent interaction, but also because of the significant increase in the surface area of the fillers (variable not considered in the equations used), which further hindered the permeation of the solvent into the polymer, interfering with the obtained crosslink density value. Thus, the sample with CNCs may exhibit a high value of V_{rf} but not necessarily a significantly high crosslink density. In contrast, for the bleached fiber composites, whose addition in NR does not cause such a significant increase in the surface area, the obtained crosslink density values may be more realistic.

3.2.6. Morphology of NR composites

Figure 5 shows the SEM images of the NR composites. The images obtained of pure NR and of the composites indicate the presence of microscale particles (white arrows), which are probably related to the vulcanization additives. The micrographs of the composites with raw and bleached fibers [Figures 5(b), (c), (d), and (e)] show individual fibers distributed in the NR matrix (black arrows). The NR composites with raw jute fibers exhibit larger fibers than the composites with bleached fibers, as observed in the fiber micrographs (see Figure 1). Some cavities are observed in the composites, probably because of the pullout of the fibers from the matrix during cryogenic fracture. In the composites added with 10 phr of bleached jute fibers, the fibers appear to be more interconnected than those in the other composites. This morphology may explain the increase in the tensile strength and hardness values. In addition, the composites based on NR and jute fibers have a more homogeneous fiber dispersion in the matrix phase than that observed in other works based on rubber and natural fibers^{11,29}.

In the images obtained using SEM of the NR composites containing 5 phr of CNCs [Figures 5(f) and (h)], no agglomerates are present. Conversely, in the composite with 10 phr of CNCs, some clusters can be observed (black circles), consistent with other studies^{4,17} on composites containing a high concentration of CNCs (above 10 phr).

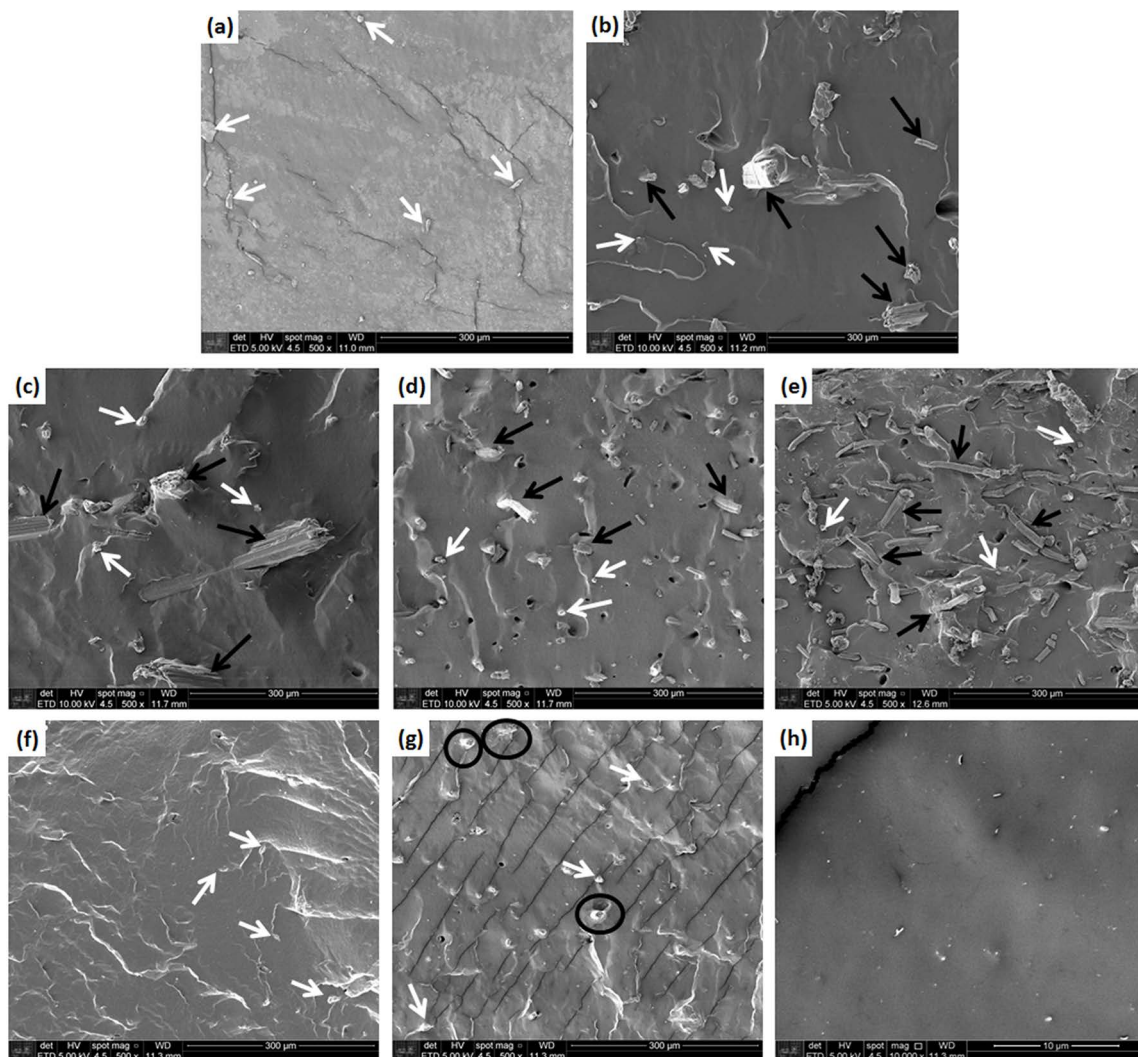


Figure 5. SEM images of (a) neat NR composite ($\times 500$); NR composites with (b) 5 phr of raw jute fibers ($\times 500$), (c) 10 phr of raw jute fibers ($\times 500$), (d) 5 phr of bleached jute fibers ($\times 500$), (e) 10 phr of bleached jute fibers ($\times 500$), (f) 5 phr of CNCs ($\times 500$), (g) 10 phr of CNCs ($\times 500$); and (h) composite with 5 phr of CNCs at higher magnification ($\times 10000$).

4. Conclusions

The jute fiber bleaching process proved to be efficient, as the morphological analysis indicated that the treatment helped remove the solids from the surface of the fibers. The treatment also allowed the separation of microfibrils and the removal of amorphous mass from the fibers. Cellulose nanocrystals (CNCs) were obtained from the acid hydrolysis of bleached jute fibers. The XRD analysis of CNCs indicated an increase in the crystallinity index when compared to the raw and bleached jute fiber samples. TEM images showed the presence of thin needles, with dimensions consistent with those of cellulose nanocrystals. In relation to the composites, an increase in the stiffness of the NR matrix was observed with the addition of all three fillers (raw and bleached jute fibers and CNCs).

A comparison of the properties of the different composites with those of pure NR compound showed that the maximum torque values, the modulus at 100 and 300% elongation, hardness, and T_g values increased. In particular, the increases were more significant in the samples containing bleached fibers. These results are related to the reduction in the mobility of the rubber molecules due to the presence of the fillers. The samples containing bleached fibers and CNCs showed an increase in the V_{rf} values, which can be attributed to the strong matrix-filler interaction and increased crosslink density. In addition, the rheometric results indicated that fibers, mainly bleached fibers and CNCs, favored the vulcanization reactions of the NR, significantly advancing the optimum vulcanization time, thereby increasing the cure rate index of the NR.

5. Acknowledgments

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