Degradation Studies on Plasticized PVC Films Submited to Gamma Radiation

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Poly (vinyl chloride), PVC, is a rigid polymer and for several of its applications must be compounded with plasticizing agents. The plasticizers minimize the dipolar interactions, which exist between the polymer's chains, promoting their mobility. In this work we studied the properties of PVC/plasticizer systems submitted to different doses of gamma radiation. We have used four commercial plasticizers amongt them di(2-ethylhexyl) phthalate, DEHP, which is present in a great number of commercial applications. The PVC/plasticizer systems have been studied as films made by the solvent evaporation technique. Irradiated and non-irradiated films have been characterized by viscosimetric analysis, mechanical essays and infrared spectroscopy. The results have shown that the rigid, non plasticized, PVC film presented the greatest degradation index, while among the plasticized films the one which presented the larger degradation index due to chain scission was the DEHP plasticized PVC.

Keywords: PVC, gamma radiation, degradation, plasticizer, DEHP

1. Introduction

Poly (vinyl chloride), PVC, has a polyhalogenated chain with chlorine atoms covalently linked to atoms of carbon, providing thus many points of dipolar interaction along its chain which give rise to strong interchain interactions and consequent rigidity of the polymeric material. Plasticizing additives break interchain dipole interaction providing a material with mobility and flexibility characteristics of a polymer with less interchain interaction¹. PVC compounded with plasticizers has many applications among them medical devices, packaging material and children's products². Many of the applications require sterilization by gamma radiation.

When polymeric materials are submitted to gamma radiation changes are observed on their molecular structure, mainly chain scission which leads to reduction on molar mass, and reticulation which increases molar mass and reduces solubility³. These molecular alterations lead to changes on mechanical properties of the material.

2.2. Methods

PVC films were prepared by solvent evaporation of solutions 75%:25% by weight of PVC and plasticizer. The conditions for the preparation of uniform films were determined by a quimiometric technique of planning⁴.

In this paper we present a study of the properties of PVC/plasticizer systems submitted to different doses of gamma radiation. PVC was compounded with four commercial plasticizers: DEHP di(2-ethylhexyl)phthalate), DIBP (diisobutyl phthalate), TOTM (tris(2-ethylhexyl)trimellitate), and viernol, a polymeric plasticizer.

2. Experimental

2.1. Materials

PVC, in pellets, was obtained from Tiletron®, and DEHP, DIBP, TOTM and viernol were obtained from CIBA - GEYGY®. Tetrahydronfuran, THF, of analytical grade was purchased from VETEC and used with no further treatment.

Differently compounded PVC films were exposed to gamma radiation from a 60 Co source with a dose rate of 5.87 kGy/h and doses of 10, 25 and 60 kGy. Both exposed and non-exposed films, were analyzed by IR spectroscopy and viscosimetry, and submitted to mechanical testing.

Viscosimetric analyses were carried on an Ostwald-Fenske viscosimeter, where the draining time of THF solutions was measured at the constant temperature of 25 °C. Molecular weights were calculated through the following equations:

$$\eta_{red} = (t - t_o)/t_o$$
 Reducted viscosity (1)

$$\eta_{esp} = \eta_{red} C$$
 Specific viscosity (2)

$$\frac{\eta_{esp}}{C} = [\eta] + K_{SB}[\eta]\eta_{esp}$$
 Schulz – Blaschke equation (3)

$$[\eta] = KM_y^a$$
 Mark-Howink-Sakurada equation (4)

where t = solvent's draining time (THF), t_0 = polymeric solution's draining time, C = polymeric solution's concentration (g/dl), $K_{\rm SB}$ = Schulz-Blaschke's constant and a, K = constants obtained from the literature⁵.

In order to obtain the Schulz-Blaschke constant ($K_{\rm SB}$), THF solutions (0.8; 0.4; 0.2; 0.1 g/dl) of non-irradiated PVC/plasticizer films were used. Draining time of irradiated samples was measured in solutions of 0.2 g/dl.

The viscosimetric molecular weight was related to the degradative process by the Degradation Index, DI, calculated through Eq. 5:

$$DI = \left[\frac{M_{v}}{M_{vi}} - 1 \right] \tag{5}$$

where M_v and M_{vi} are the viscosimetric molecular weight of the non-irradiated and irradiated samples, respectively⁶.

Mechanical tests were done in a Universal Trial Machine Testometric Micro 350 in accordance to ASTM D-882 in the following conditions: claw velocity: 100 mm/min; initial claw distance: 30 mm; sample width: 15.2 mm.

Fourier transform infrared analyses were done in a Brucker IFS66 spectrofotometer on KBr pellets. The spectrum of each film sample irradiated at 10 kGy was measured and the film was further irradiated at 25 kGy before another spectrum of the same film was taken, a third spectrum was then taken after irradiation of the film at 60 kGy. Evaluation of changes in band intensity from spectrum to spectrum was then done.

3. Results and Discussion

Figure 1 presents the results of the degradation index

(Eq. 5) for PVC and PVC/plasticizer systems as a function of the radiation doses to which they have been submitted. These results show that non plasticized PVC as well as the plasticized systems PVC/DEHP, PVC/TOTM and PVC/viernol have suffered degradation by chain scission upon irradiation. The system PVC/DEHP presented the highest index of degradation, while PVC/TOTM and PVC/viernol systems presented less degradation than PVC/DEHP suffering both the highest degradation upon irradiation at 10 kGy. The system PVC/DIBP on the other hand has not suffered scission of the polymer's chains at any irradiation dosage. PVC's molecular weight decrease associated to exposition to degradative sources has been observed and reported in the literature⁷.

Results on measurements of tensile strength of plasticized PVC films before and after irradiation, presented in Fig. 2, revealed a decrease on tensile strength upon irradiation which is proportional to radiation dose. Films of PVC/DEHP presented a sharp decrease on tensile strength when irradiated at 25 kGy, which is the dosage used for sterelization of comercial products based on the PVC/DEHP composition. For the PVC/viernol films a sharp decrease on tensile strength was observed after irradiation even at 10 kGy. Non irradiated films of PVC/DIBP and PVC/TOTM presented smaller tensile strength than the other two systems studied, and both sufferred relatively large losses on tensile strength upon irradiation.

Results on measurements of elongation of plasticized PVC films before and after irradiation presented in Fig. 3 revealed a decrease on elongation at break upon irradiation. The PVC/DEHP films had a sharp reduction (74%) on elongation at break already upon irradiation at the lowest dosage, 10 kGy, used in our study. Films of PVC/TOTM, PVC/DIBP and PVC/viernol presented gradual reduction on elongation at break as the radiation dose was increased.

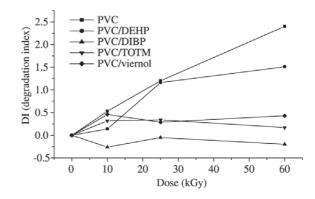


Figure 1. Degradation index of pure PVC and plasticized PVC irradiated films as a function of radiation dose.

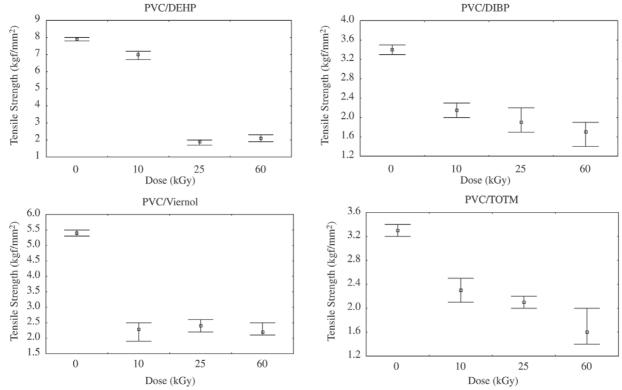


Figure 2. Tensile strength of PVC/Plasticizer irradiated and non irradiated films as a function of radiation dose.

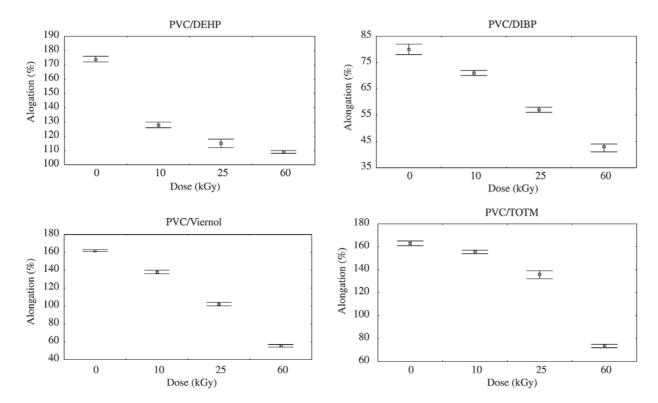


Figure 3. Elongation at break of of PVC/Plasticizer irradiated and non irradiated films as a function of radiation dose.

Figure 4. Formation of insaturations at PVC chain by the "zipper" effect.

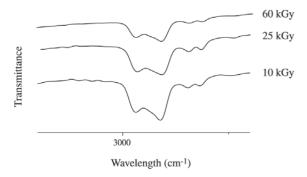


Figure 5. Infrared spectra of non plasticized PVC film that was sequencially irradiated at 10, 25 and 60 kGy, in the region between 3200 and 2700 cm⁻¹.

The irradiated PVC films darken and lose transparency. Darkening is proportional to the radiation dose⁸ and is explained by the formation of double bonds along the chain in a process known as the zipper effect⁹, illustrated in Fig. 4.

Our results of infrared spectroscopic analysis of the films have shown a decrease in the intensity of the bands attributed to the C-H methylene stretching, appearing at 2912 and 2970 cm⁻¹, an evidence of the loss of methylene protons which agrees with the formation of double bonds. In Fig. 5 we present the infrared spectra of non plasticized PVC irradiated consecutively at different doses.

4. Conclusion

Plasticized poly (vinyl chloride) films as well as the non plasticized film degrade when exposed to gamma radiation. Both chain scission and double bond formation have been observed. As a consequence of degradation we have observed a decrease on tensile strength and elongation at break of the PVC/Plasticizer films.

Among the irradiated plasticized poly (vinyl chloride), the PVC/DEHP film have shown the highest degradation due to chain scission, particularly when exposed to the dose of 25 kGy which is the sterilization dosage, used for medical and pharmaceutical devices and packaging materials.

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References

- 1. Verceze, A.V. *Master's Dissertation*, University of Campinas, SP, Brazil, 1996.
- 2. Freitas, V. Aplicações do PVC, PVC on line- Instituto do PVC, São Paulo, 2002.
- 3. Yagoubi, N.; Baillet, A.; Pellerin, F.; Ferrier, D. *Beam Interactions with Materials and Atoms*, v. 105, p. 340-344, 1995.
- 4. Neto, B.; Searminio, B.; Bruns, R.E. *Planejamento e Otimização de Experimentos*, 2nd ed., UNICAMP, SP, Brasil, 1995.
- 5. Brandrup, J.; Immerggut, E.H. *Polymer Handbook*, John Wiley & Sons, USA, 1989.
- Aquino, K.A.S.; Master's Dissertation, Federal University of Pernambuco, PE, Brazil, 2000.
- 7. Genova-Dimitrova, P. *Polymer Degradation and Stability*, v. 33, p. 355-365, 1991.
- 8. Vinhas, G.M.; Lapa, C.M.; Almeida, Y.M.B.; Souto Maior, R.M. *Anais do 6º CBPol / IX INTERNATIONAL MACROMOLECULAR COLLOQUIUM*, Gramado, RS, p. 707-709, 2001.
- 9. Gökçel, H.I.; Balköse, D.; Köktürk, European Polymer Journal, v. 35, p. 1501-1508, 1999.