Synthesis and Characterization of Silica Core/Multilayered Cobalt Ferrite-Silica Shell Particles for Lipase Immobilization

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Multilayered core/shell particles were prepared by four-step deposition of cobalt ferrite layers and then deposition of external silica layer on the surface of silica core particles. The ferrite nanoparticles were obtained by co-precipitation of Co²⁺ and Fe³⁺ ions at pH=11 and pH=15, respectively. The agglomerated silica core/cobalt ferrite shell particles were obtained at extremely basic pH (pH=15), while at pH =11, monodispersed and non-aggregated core-shell particles were obtained. The poly(diallyldimethylammonium chloride)-functionalized silica core/ferrite shell particles were used as templates for deposition of mesoporous silica layers (obtained by neutralization of highly basic sodium silicate solution). The obtained porous multilayered core/shell particles were used as a host for covalent lipase immobilization inside the external silica layer. The initial activity of immobilized enzymes was about 13-fold lower than the native one, however, it showed good reusability and improved thermal stability compared to native ones.

Keywords: Core-shell, multilayered shell, magnetic properties, mesoporous silica, immobilization, lipase.

1. Introduction

Enzymes are versatile biocatalysts and have been used in a wide range of sectors including food¹, textile², ultrasensitive biosensing and development of pharmaceuticals³, fine chemicals and environmental engineering⁴. However, the application of natural enzymes is hampered by difficulties in reuse, product contamination and separation, as well as the poor catalytic efficiency and stored durability. These drawbacks can generally be overcome by immobilization of enzymes using various support materials and immobilization protocols⁵.

Recently, magnetic nanoparticles have attracted significant attention due to their large surface area, functionalizable surfaces, easy separation using external magnetic field, super paramagnetic property and low toxicity⁶. Significant progress has been made in order to develop new biocatalytic systems based on immobilization of enzyme or whole cells onto magnetic nanocarriers. The epoxy-functionalized magnetic nanoparticles were used as the carriers for immobilization of alcohol dehydrogenase, and immobilized enzyme retained 84% initial activity after five cycles⁷. Dopamine functionalized Fe₃O₄ nanoparticles coated with oleic acid were used for immobilization of lipase from Thermomyces lanuginosus via covalent coupling⁸. It has been found that thermostability of the immobilized lipase was 2.9-fold improved at 60 °C, compared to the free enzyme’s thermostability. Furthermore, more than 75% of the initial activity of the immobilized lipase was retained after ten repeated uses⁹. Also, magnetic nanoparticles used as a cofactor regeneration system were applied for immobilization of glycerol dehydrogenase and NADH oxidase, in order to efficiently produce 1,3-dihydroxyacetone from inexpensive substrates⁹.

Furthermore, functionalized magnetic nanocarrier grafted with protein spacers can be used for immobilization of recombinant Escherichia coli harboring glycerol dehydrogenase using specific recognition¹⁰. Studies have shown that the obtained immobilized cell exhibited higher thermal stability compared to the free cell and after ten cycles, the immobilized cell retained 62% initial activity. Magnetic nanoparticles may be organized in core-shell type structures, where the properties emerging from either the core or shell materials can be controlled¹¹. The unique core-shell structures and synergetic effect obtained by combining properties arising from either core or shell materials make them important for application in diverse fields, among them, biocatalysis¹²-¹⁵. Magnetic Fe₃O₄ nanoparticles covered with SiO₂ shell were used for covalent immobilization of lipase¹². Compare to the free enzyme, the immobilized lipase displayed higher relative activity in the range of pH

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from 7.0 to 9.5 and higher thermal stability at temperature range from 30 to 70 °C. The relative activity of immobilized enzyme was 5.8 fold higher than that of the free lipase at 70 °C after 10 h incubation. Also, it has been found that monodisperse core-shell magnetic organosilica nanoflowers with magnetic core and flower-like organosilica shell, which posses radial-wrinkle and tunable perpendicular channels, can be used as a support for Candida antarctica lipase B (CALB) covalent immobilization. This study showed that the flower-like organosilica shell allowed high enzyme loading (93 mg/g\text{_support}), high specific hydrolytic activity (22,700 U/g\text{_support}) and better pH stability than free CALB. The enzyme also retains 65% of initial activity after being incubated in cyclohexane at 60 °C for 96 h, and retains 90% of initial activity after being stored at the room temperature for 42 days.

With these desired characteristics, the design of core-shell particles for lipase immobilization opens up a new way of improving the catalytic activity and stability of the lipases, providing a “shield effect” to protect immobilized enzyme from biological, thermal and chemical degradation and may have potential applications in various lipase-based industrial processes.

In our previous work, we have developed core-shell particles for enzyme immobilization composed of a dense silica core surrounded by a bilayered shell consisted of either internal ferrite layer and external mesoporous layer or of both silica layers. In the first case, the inner ferrite layer was superparamagnetic and the external mesoporous silica layer was prepared for potential enzyme immobilization. In the second case, the internal silica layer was used as a host for lipase immobilization by a physical adsorption, while the external silica layer prevented lipase leaching. The internal ferrite layer obtained via a one-step deposition process was very thin giving the low magnetization in the applied magnetic field. On the other side, the activity of lipase immobilized inside the bilayered shell was reduced due to diffusion resistance within the outer silica layer.

In this research, four-layered and thick cobalt ferrite shell was deposited onto silica core particles by a four step deposition process. The external silica layer was deposited on the surface of four-layered ferrite shell and used as a host for covalent immobilization of lipase.

2. Material and Methods

2.1. Preparation of silica core particles from TEOS

Silica core particles (sample C) were synthesized using hydrolysis and condensation of tetraethylorthosilicate dissolved in anhydrous ethanol and distilled water under basic conditions (25% NH_4OH, Merck). The sample was prepared using a molar ratio TEOS:H_2O:NH_4OH = 1:40:2 and a TEOS concentration of 0.25 mol/L. The water-ethanol-ammonia mixture has been prepared by mixing pure water, pure ethanol and ammonia solution (25%) and after the addition of TEOS, the product suspension was continuously stirred at room temperature for 1 h. The resulting white precipitated powder was centrifuged and washed with distilled water until the effluent was free of NH_4 groups, and finally dried at 120°C for 1 day.

2.2. Preparation of four-layered silica core/cobalt ferrite shell particles

To enable incorporation of divalent metal ions (e.g. Ni^{2+}, Zn^{2+}, Co^{2+}) into ferrite structure, coprecipitation of Fe^{3+} and Co^{2+} ions was done at higher pH values (pH=11 and pH=15, respectively). In order to prevent self-aggregation process of ferrite particles obtained at high pH, surface modification of ferrite particles with citric acid was also performed (using the total cations/citric acid molar ratio 1/1). Equimolar amounts of the precursors (Fe(NO_3)_3\times9H_2O and Co(NO_3)_2\times6H_2O) were dissolved in deionized water, stirred and heated for 10 min at about 80 °C. To allow electrostatic deposition of generated ferrite nanoparticles on the surface of silica core particles, the latter were pre-functionalized with poly(diallyldimethylammonium chloride) (PDDA) (Figure 1a). The warm solution of nitrate ions obtained this way was slowly added into a stirred vessel containing PDDA-functionalized silica core particles dispersed in 0.5 mol/L sodium hydroxide solution containing the sodium citrate (trisodium citrate equimolar amount relative to metal ions) having a constant pH value of 15 and 11, respectively, and temperature of 80°C (Figure 1b). After addition of the last amount of nitrate ions, the reaction continued for 10 min. In the case of sample obtained at pH=11, the procedure was similar, except that the initial pH (pH=11) was set by adding a small amount of sodium hydroxide solution during the addition of the nitrate solution into dispersion of silica core particles.

The resulted precipitated powders were centrifuged and washed with distilled water several times and finally dried at 100 °C. To allow electrostatic deposition of the second cobalt ferrite layer on the surface of silica core/monolayered shell particles, the latter were functionalized with PDDA.
The functionalized silica core/monolayered shell particles were centrifuged and used as templates for deposition of the second cobalt ferrite layered (Figure 2d) prepared as discussed above. The process of deposition of ferrite nanoparticles at pH=11 was repeated 2 more times, while, between each deposition, the functionalization of obtained ferrite shells by PDDA was performed (Figure 2 e-h).

2.3. Deposition of silica shell onto silica core/ ferrite shell particles

The silica core/four-layered ferrite shell particles (sample CF4-11) were used as templates for assembling the silica layer, which was composed of silica nanoparticles. To allow the electrostatic assembling of the silica layer on the surface of silica core/four-layered ferrite shell particles, the latter were functionalized with PDDA (Figure 2a). The PDDA-functionalized silica core/four-layered ferrite shell particles were prepared by dispersing 0.4 g silica core/four-layered ferrite shell in 40 ml PDDA solution (at a concentration of 6 mg/ml), which also contained 0.02 M NaCl. The suspension has been stirred for 1 hour at room temperature. The modified particles (sample CF4p) were centrifuged and successively washed with distilled water. These particles were dispersed in highly basic sodium silicate solution having a Na₂O/SiO₂ molar ratio of 0.4 and a SiO₂ concentration of 1.25 mol/L (Figure 2b). The weight ratio between the silica core/ferrite shell particles (sample CF4-11) and silica from sodium silicate was 2. Sulfuric acid (1 mol/L) was slowly added to this well-stirred dispersion at 80°C to decrease the pH value to enable generation of pores formed by aggregation of primary silica nanoparticles and their deposition on the surface of the PDDA-functionalized particles (Figure 2c).

2.4. Functionalization of the external silica layer

To allow covalent immobilization of lipase onto the external silica layer, the samples were functionalized in two steps: functionalization of silica layer with APTES (3-aminopropyltriethoxysilane) and grafting of amine groups functionalized silica layer with glutaraldehyde. The APTES-modified silica layer was prepared by dispersing the sample CF4-S (0.5 g) in anhydrous acetone (35 ml), adding 0.15 ml APTES and stirring for 24 h. The obtained particles were centrifuged, extensively washed with anhydrous acetone and finally dried by removing excessive solvent at 80 °C. Amine-modified silica layer was also activated with a 5% glutaraldehyde solution at pH=5.2. The supports have been immersed in the glutaraldehyde solution for 1 hour at room temperature. The powder was centrifuged and washed four times with water to remove all remaining excess of unreacted glutaraldehyde.

2.5. Immobilization of lipase onto external silica layer

The covalent immobilization of lipase from C. rugosa (Sigma–Aldrich) onto the functionalized mesoporous silica layer was performed according to the following procedure: 0.33 g of functionalized core-shell particles and 25 ml of lipase solution (0.06 mg/ml) in phosphate buffer (0.05 M and pH = 7.0) were added to a flask and stirred magnetically for 2 h at ambient temperature (Figure 3a). The pore dimensions allowed immobilization of the enzymes inside inter-particle pores formed by aggregation of primary silica nanoparticles (Figure 3b). The reaction mixture was centrifuged and the supernatant was used for determining the amount of lipase remaining in the solution.

The concentration of lipase solution was measured by the Bradford method²⁰, with BSA (bovine serum albumin) as a standard at 595 nm.

2.6. Characterization of particles

The size and morphology of the particles were examined using a scanning electron microscope (SEM, JSM-6390 LV JEOL, operating at 30 kV) coupled with EDS (Oxford Instruments X-MaxN). Prior to SEM imaging, the samples were sputtered with gold. TEM characterization was performed using a JEM 1400-Plus JEOL device operating at 120 kV. The specific surface area (according to the BET method) and pore size distribution (according to the BJH method) were measured by low-temperature nitrogen adsorption using a Quantachrom Autosorb-3B instrument. The zeta potential of particles was measured by dynamic light scattering (Zetasizer Nano ZS, Malvern Instruments). X-ray analysis of powders was performed using a Rigaku MiniFlex600 diffractometer.
(CuKα radiation, λ=1.5406 Å). X-ray diffraction patterns were recorded in the range of 10–80°, with a scan rate of 0.03°/s.

Thermomagnetic measurements in air were conducted by Faraday method, which presumes the influence of the non-homogenous magnetic field (with an intensity of $H_{app} = 50$ kA/m) on the magnetic sample during heating. Before thermomagnetic measurements, the powder was pressed at pressure of 500 MPa to produce specimen with radius of 8 mm and width 1.5 mm.

2.7. Activity assay of free and immobilized enzymes

The immobilization efficiency (%) of the enzyme was calculated using the equation:

$$\text{Immobilization efficiency} = \left( \frac{C_i - C_f}{C_i} \right) \times 100$$

Where $C_i$ and $C_f$ are the initial and final concentration of the enzyme, respectively.

The activities of the free and immobilized lipase were measured by monitoring the catalytic hydrolysis of 4-nitrophenyl acetate (pNPA) to 4-nitrophenol. Due to poor solubility in water, pNPA was first dissolved in acetonitrile and then added to the phosphate buffer (0.05 M, pH=7). Activity assays were carried out by adding 100μL of free enzyme solution (at a concentration of 0.016mg/ml) or 30 mg of the enzyme-loaded material (the amount of bound enzyme on support CF4-S was 2.69 mg per gram of support) to 10 ml of lipase substrate (the concentration of 4-nitrophenyl acetate was 0.55, 1.1, 2.2, 4.4 and 6.6 mM) at 28 °C. The reaction time of the native and immobilized lipase was 5 minutes. Afterwards, the amount of 4-nitrophenol was determined spectrophotometrically at 410 nm. Enzymatic activity was performed in three replicates. The negative control for the enzyme assay was prepared using the same procedure, but without enzyme.

The thermal stability of the immobilized and native enzymes was examined by incubating two types of enzymes (free and immobilized enzyme) in 0.05 M phosphate buffer (pH = 7) at 45, 60 and 75°C for 10 minutes, respectively. To prevent degradation of p-nitrophenyl acetate at elevated temperatures, the effect of thermal treatment on enzymatic activity was determined at 28°C and substrate concentration of 1.1 mM.

Desorption and leakage of immobilized enzymes was examined after the fifth cycle of reusability testing by mixing 30 mg of reused CF4-S-E composite in 10 ml 0.05 M phosphate buffer.

The material was centrifuged and 0.5 ml of supernatant was put into 10 ml of lipase substrate (the concentration of 4-nitrophenyl acetate was 1.1 mM) at 28 °C to examine the potential presence of the enzyme via substrate hydrolysis.

3. Results and Discussion

3.1. Preparation of silica core/cobalt ferrite shell particles

SEM and TEM micrographs of the particles obtained by coprecipitation of Co<sup>2+</sup> and Fe<sup>3+</sup> ions at pH=15 onto PDDA-functionalized silica core particles via two-step coating method (sample CF2-15) followed by functionalization of previously deposited ferrite layer with poly(diallyldimethylammonium chloride) (PDDA) are shown on Figure 4 a-b, respectively.

Figure 5 shows the TEM micrographs of the silica core/cobalt ferrite shell particles (samples CF1-11 and CF4-11 obtained by one-step and four-step layer deposition, respectively, at pH=11). The obtained non-agglomerated particles were monodispersed with uniform shells. This indicates a slower agglomeration process at lower pH (pH=11). A very thin shell was obtained after one step layer deposition (Figure 5a). On the other side, a homogenous shell with average thickness of ~50 nm was obtained after four-step layer deposition (Figure 5b).

It has been observed that the thickness of 50 nm, obtained by a four-step deposition process at pH=11, allowed the separation of silica core/cobalt ferrite shell particles from...
aqueous solution using a N45 NdFeB magnet (Figure 6). Taking this into account, and since the further increase in the number of deposits would result in a more intense response to the magnetic field, a four-stage deposition was considered sufficient.

Aggregation of the silica core/cobalt ferrite shell particles at pH = 15 and absence of aggregation were also confirmed by the particle size distribution shown on Figure 7. There is unimodal distribution for CF4-11 sample obtained at pH = 11, representing the separate silica core/cobalt ferrite shell particles. On the other hand, there is a bimodal particle size distribution for CF2-15 sample obtained at pH = 15, showing a high intensity peak at 5 μm, which represents aggregated silica core/cobalt ferrite shell particles.

EDS analysis with chemical mapping of the silica core/cobalt ferrite shell particles obtained at pH = 15 (a,c) and pH = 11 (b, d) is shown on Figure 8. This revealed that these layers are composed of uniformly distributed iron, cobalt, oxygen and silicon atoms. The presence of sodium in the sample obtained at extremely high pH value (pH = 15) is due to high concentration of sodium hydroxide used during precipitation process. The presence of carbon is due to functionalization with PDDA. The chloride atoms come from sodium chloride used in functionalization with PDDA. In addition, EDS mapping of the sample surface showed that Fe and Co atoms are uniformly distributed over the sample surface.

Figure 9a shows X-ray pattern of the silica core particles (red lines) and ferrite particles (black lines) obtained by coprecipitation of Co$^{2+}$ and Fe$^{3+}$ ions at pH = 15 in the presence of equimolar amount of citric ions. X-ray pattern of the silica core particles shows a broad reflection peak centered at about $2\theta = 24^\circ$, which is a characteristic of amorphous silica. On the other hand, the cobalt ferrite particles revealed an ill-crystallized material approaching the amorphous state. Three weak and very broad peaks, at...
2θ = 35.43° (311), 43.05° (400) and 62.58° (440) represent the most intense diffraction peaks of cobalt ferrite particles with a cubic spinel structure.

X-ray diffraction patterns of the monolayered (sample CF1-11; black lines) and four-layered silica core/cobalt ferrite shell particles (sample CF4-11; red lines) are shown in Figure 9b. The shell thickness affects the amplitude of XRD peaks (Figure 9b). Diffraction peak which is a characteristic of amorphous silica is clearly visible in X-ray pattern of the monolayered silica core/cobalt ferrite shell particles (sample CF1-11). On the other hand, this diffraction peak is unexpressed in X-ray pattern of the four-layered silica core/cobalt ferrite shell particles (sample CF4-11), while the diffraction peaks located at (311) and (440) are expressed.

After the heat treatment at 600°C, the crystallinity of the monolayered and four-layered cobalt ferrite shell was more pronounced with diffraction peaks located at (311), (400), (511) and (440) for the sample CF1 (Figure 10a), as well (220), (311), (400), (511), (440) and (533) for the sample CF4 (Figure 10b).

Magnetization vs temperature curves of the pressed powder (sample CF4-11) are shown in Figure 11. The results of the magnetization measurements indicate that, after the first heating of the sample up to a temperature of 320°C, the magnetization of the cooled sample was $M_1=0.16 \text{ Am}^2/\text{kg}$ and was 14\% higher than the initial magnetization obtained before heating ($M_0=0.14 \text{ Am}^2/\text{kg}$). After the heating of the same sample up to a temperature of 420°C, the magnetization of the cooled sample was $M_2=0.17 \text{ Am}^2/\text{kg}$ and was 21.4\% higher, relative to the initial magnetization obtained before heating ($M_1$). After heating of the same sample up to a temperature of 600 °C, the magnetization of the cooled sample was $M_3=0.13 \text{ Am}^2/\text{kg}$ and was 7\% lower, relative to the initial magnetization obtained before heating ($M_0$).

The increase in magnetization of the cooled sample obtained after heating up to a temperature of 320°C and 420°C, respectively, is caused by a process of structural relaxation of the powder.

It should be noticed that, the as-synthesized powder contains a certain density of defects and mechanical microstresses,
which interfere the orientation of the magnetic domains in the applied magnetic field. The heating process influenced the structural relaxation of the ferrite shells and resulted in annihilation of defects and mechanical microstresses. After heating the sample to 420 °C, a relaxed amorphous structure is obtained, whereby the system changes from a less stable to a more stable structural state. The relaxed amorphous structure of the powder provides easier movement of the magnetic domain walls and their better orientation in the applied magnetic field, which consequently resulted in increased magnetization of the sample.

Heating the sample to 600 °C caused the amorphous phase crystallization process (XRD patterns shown on Figure 10), which resulted in a decrease in magnetization of about 7% compared to the initial value of magnetization before the first heating. The measurement results show that

Figure 9. X-ray diffraction patterns: a) silica core particles (C) and cobalt ferrite particles (CoFe₂O₄); b) monolayered (CF₁₁) and four-layered (CF₄₁₁) silica core/cobalt-ferrite shell particles obtained at pH=11.

Figure 10. X-ray diffraction patterns of samples calcined at 600°C: a) silica core/monolayered cobalt ferrite (CF₁₁); b) and silica core/ four-layered cobalt ferrite particles (CF₄₁₁).

Figure 11. The dependence of magnetization vs. temperature of the pressed powder (sample CF₄₁₁).
the Curie temperature did not change after heating thus remaining 240°C.

3.1.1. Deposition of mesoporous silica layer onto silica core/ferrite shell particles

The silica core/ferrite shell particles (sample CF4-11) were used as templates for deposition of silica nanoparticles generated by slow addition of sulphuric acid into highly basic sodium silicate solution. Silica nanoparticles were generated at basic conditions and, in order to allow electrostatic deposition of generated silica nanoparticles on the surface of silica core/ferrite shell particles, the latter were functionalized with PDDA. As shown in Table 1, the zeta potential of PDDA functionalized silica core/ferrite shell particles (sample CF4-p) was 27.6 mV, while the zeta potential of the sample obtained after deposition of silica layer onto PDDA-functionalized silica core/ferrite shell particles was -20 mV. This indicates that the silica layer is deposited around positively charged PDDA-functionalized ferrite shell.

The formation of the external silica layer (generated from highly basic sodium solution) around silica core/ferrite shell particles, obtained from highly basic sodium silicate solution, was also confirmed by SEM and TEM micrographs (Figure 12). The SEM and TEM micrographs show small aggregates composed from several core-shell-particles and large aggregates with average size of about 6 μm. This indicates that the deposited silica layer shows tendency to form networks near the gel point resulting in aggregate formation. This is in a correlation with a bimodal particle size distribution (Figure 13) with a peak at 1 μm representing a small size aggregates, and a peak at 6 μm, representing bigger aggregates. TEM micrograph (Figure 12b) shows small aggregate composed of several aggregated core-shell particles with continuous silica shell. On the other hand, the silica shell was mesoporous with non-uniform thickness ranging from 40 to 100 nm (Figure 12c).

As it was shown in our previous work, the nitrogen adsorption–desorption isotherms of the silica particles obtained from sodium silicate solution at 80 °C belong to Type IV, indicating that these particles have a mesoporous structure with the average pore size of ~24 nm.

3.1.2. Immobilization of enzymes onto mesoporous silica layer

The effect of the substrate concentration on the initial velocity of the free and immobilized enzyme is shown on Figure 14. The activity of enzyme was significantly affected by immobilization onto mesoporous silica layer of core/shell particles (sample CF4-S). The initial velocity of the free and immobilized lipase was investigated at different substrate concentrations ranging from 0.55-6.6 mM. The activity of immobilized lipase was lowered by 13-fold compared with a native form. The activity of the lipase immobilized inside of silica layer (sample CF4-S) was about 4.6 μmol/mg min at initial substrate concentration of 6.6 mmol/dm³, while the activity of native one was about 62 μmol/mg min at the same concentration. The activity of the lipase immobilized onto the silica shell was decreased due to the limited diffusion of the substrate, which was caused by the internal mass transfer resistance. This resistance occurs in interstitial pores of mesoporous silica layer and this problem could be solved by increasing the silica shell pore size.

The effect of pH on the activity of native and immobilized enzyme is shown in Figure 15. The immobilized lipase shows a broader pH activity profile compared to the native lipase. The immobilized lipase possessed more than 90%

<table>
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<tr>
<th>Sample</th>
<th>CF4-p</th>
<th>CF4-S</th>
</tr>
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<tbody>
<tr>
<td>Zeta potential</td>
<td>27.6</td>
<td>-20.1</td>
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Figure 12. SEM (a) and TEM (b, c) micrographs of silica core/four-layered ferrite/silica shell (sample CF4-S).
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Relative activity in the pH range 7.0–10.0. The higher relative activities might be attributed to the multipoint attachment of lipase to mesoporous silica shell, which buffered variability of enzyme molecules when the pH was changed.

Reusability of enzymes is an important factor for their industrial application. Reusability of CF4-S-enzyme composite was studied for ten cycles (Figure 16). The relative immobilized enzyme activity in reaction medium slowly decreased from 100% (initial cycles) to 80% (after the tenth cycle).

To examine the potential effect of enzyme leaching from the material, the activity of the supernatant, obtained after mixing the immobilized enzyme (previously used in five reaction cycles) in the phosphate buffer, was investigated. The activity of supernatant was examined by its incubation in 1.1 mM substrate solution. No supernatant activity was detected, indicating that there was no desorption of the enzyme from the support, leading to a conclusion that desorption of the enzyme did not cause a decrease in the activity of the composite material, as it was the case with immobilization of enzyme into mesoporous silica layer by physical adsorption.

The results have shown that, since the immobilized lipase has good recyclability, materials with mesoporous silica layer assembled onto silica core/ferrite shell particles are suitable for covalent immobilization of lipase.

Temperature affects the stability and activity of free and immobilized enzymes. Figure 17 shows relative activities of the native and immobilized enzymes, determined after thermal treatment of the enzyme at temperatures of 45, 60 and 75 °C, respectively. The immobilized enzymes, retaining up to 30% activity at 75°C, showed better thermal stability than the native ones, which lost all activity after thermal treatment at 75 °C. The relative activities of the immobilized enzymes, obtained after thermal treatment at 45 and 60°C, were 88 and 56%, respectively, while the relative activities of native enzymes were 71 and 24%, respectively. The native enzyme was completely inactivated after a treatment carried out at 75 °C, while the relative activity of immobilized enzyme was 30% at the same temperature.

These results show that the thermal stability of lipase covalently immobilized inside the external mesoporous silica layer is higher than the stability of the free enzyme. Enhanced thermal stability of immobilized enzyme might be
related to the presence of more covalent cross-links between glutaraldehyde bound onto the amino functionalized silica layer and enzyme molecules, which result in conformational stabilization of lipase in support.

As a result of all this, the use of spherical silica particles as templates for deposition of cobalt ferrite allowed formation of homogenous and uniform cobalt ferrite shell.

Furthermore, the thickness of ferrite shell increases with the increase of deposition number, while the deposition of silica layer onto prepared silica core/cobalt ferrite shell particles results in formation of magnetic mesoporous composite spheres used for enzyme immobilization. Functionalized external mesoporous silica layer allowed enzyme immobilization inside mesopores providing the enzyme stability. On the other hand, internal magnetic shell provides recovery of immobilized enzymes by magnetic separation.

4. Conclusion

Four-step deposition of ferrite nanoparticles (obtained by co-precipitation of Co\(^{2+}\) and Fe\(^{3+}\) ions at pH=11) performed on PDDA functionalized silica core and previously deposited ferrite layer, respectively, resulted in formation of uniform four-layered ferrite shell with average shell thickness of about 50 nm. However, co-precipitation of Co\(^{2+}\) and Fe\(^{3+}\) ions at pH=15 resulted in self-aggregation of core-shell particles. X-ray diffraction patterns of as-synthesized silica core/cobalt ferrite shell particles showed amorphous structure, while thermal treatment at 600°C resulted in formation of crystalline regions having diffraction peaks characteristic to cobalt ferrite with a cubic spinel structure. Heat treatment of silica core/cobalt ferrite shell particles up to 420°C improved magnetization by 21.4% relative to the initial magnetization obtained before heating. Silica core/four-layered ferrite shell particles were used as a template for deposition of mesoporous silica layer, which was used as a host for covalent immobilization of lipase. The immobilized enzyme was not prone to leaching and had a decrease in relative activity by only 20% after ten reaction cycles. The immobilized lipase showed better thermal stability than the native one, retaining up to 30% activity at 75°C, while the native enzyme completely lost its activity at this temperature. The immobilized lipase also showed a broader pH activity profile compared to the native lipase retaining more than 90% relative activity in the pH range 7.0–10.0.

The future investigations will be focused on improving the magnetization of core/shell composite particles, as well as improving the activity of immobilized lipase by increasing the silica shell pore size.

5. Acknowledgments

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